

Araștırma Makalesi

**Research Article** 

# EFFECT OF SINTERING TEMPERATURE TO MICROSTRUCTURE IN METAL BINDER JETTING

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Keywords	Abstract
Metal binder jetting,	Metal Binder Jetting emerged and it has been developing an additive manufacturing
Additive manufacturing,	method. Binder type, binder ratio, sintering process, and atmosphere couldn't be
Resin,	transparently determined due to its development stage. In this study, effect of
Sintering.	sintering temperature was investigated parts' microstructure which manufactured
	by Binder Jetting Metal Additive Manufacturing. In manufacturing, SS316L metal
	powder used as a raw material, ultraviolet light cured and solid+fluid catalyzed
	cured resins used as binder. For solid+liquid catalyzed cured resin, furfuryl alcohol
	was selected as liquid binder, para toluene sulfonic acid was selected as solid
	catalyst. Isopropyl alcohol was used to dilute the ultraviolet light cured resin. The
	composition ratios and electron diffraction patterns of the samples were analyzed
	using optical and scanning electron microscopy.

# BAĞLAYICI PÜSKÜRTMELİ METAL EKLEMELİ İMALATTA SİNTERLEME SICAKLIĞININ MİKRO İÇYAPIYA ETKİSİ

Anahtar Kelimeler	Öz		
Bağlayıcı püskürtmeli metal	Bağlayıcı Püskürtmeli Metal Eklemeli İmalat yakın zamanda ortaya çıkmış ve hala		
eklemeli imalat,	gelişmekte olan bir eklemeli imalat yöntemidir. Gelişme aşamasında olmasından		
Eklemeli imalat,	dolayı net bir bağlayıcı türü, oranı, sinterleme süreci ve ortamı belirlenememiştir.		
Reçine,	Bu çalışmada Bağlayıcı Püskürtmeli Metal Eklemeli İmalat yöntemi ile imal edilmiş		
Sinterleme.	eklemeli imalat parçalarında, sinterleme sıcaklığının mikroiçyapıya etkisi incelenmiştir. İmalat için ham madde olarak SS316L metal tozu, bağlayıcı olarak ultraviyole ışıkla ve katı+sıvı reaktiflerle kürlenen reçineler kullanılmıştır. Katı+sıvı kürlenen reçine olarak furfuril alkol sıvı reçine, para toluen sülfonik asit katı katalizör olarak seçilmiştir. Ultraviyole ışıkla kürlenen reçine izopropil alkol ile seyreltilmiştir. Optik ve taramalı elektron mikroskobuyla yapılan incelemelerle kompozisyon oranları ve numunelerin elektron kırınım desenlerinin analizleri de yapılmıştır.		

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# 1. Introduction

In these days, with the technological advances, alternative ways have begun to be sought for parts that can't be manufactured with traditional manufacturing methods. One of these non-traditional manufacturing processes is Additive Manufacturing (AM) (Gürgen and Kayacan, 2020). Unlike traditional manufacturing methods, Additive Manufacturing does not work by reducing mass or mass forming the part to be manufactured from raw material. It is a system that manufactures cross-sectional areas of parts from powder or molten raw material in layers

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according to a properly made design (Yalçın and Ergene, 2017). It can be classified according to the material type, the way the layers are combined and the energy source. It can be defined as the production method in which the part is manufactured as desired. This technology, which has taken names such as rapid prototyping and 3D printing, is now known as Additive Manufacturing (AM) in the literature.

AM technologies have started to come to the fore as rapid prototyping systems and has shown great development since the last half of the 20th century. AM methods including, a part is produced by adding the cross-sectional areas of the designed model on top of each other in layers, starting from a digital design file. The designed 3D digital solid model is not sent directly to the AM device. The model is first converted to the Standard Tessellation Language (STL) formatted file, which is the standard interface of the AM (Duman and Kayacan, 2016). After that, the manufacturing file is transferred to the AM device, and the manufacturing process is started. This manufacturing technique, which can be considered new, brings many benefits. The most important advantages are that it can be designed without considering any restrictions on production and thus complex parts can be created, and that there is minimal waste material, no cutting tools and forces.

In Binder Jetting (BJ), which is one of the AM types; firstly, a roller is used to spread powder material throughout the build platform. The binder material is then sprayed onto the layer, similar to inkjet printers. As a result of the spraying process, the desired geometry is obtained in two dimensions. Then the same process is repeated by passing to the next layer three-dimensional part is obtained. After the manufacturing process is finished, the part is raw form. Most of the time, post processes (coating, infiltration, etc.) are done to increase the mechanical properties. Adhesive agent (in ceramics) as coating material and bronze (in metals) as infiltration material are examples of materials used for post processes (Redwood, 2019).

In Metal Binder Jetting (MBJ), it is possible to manufacture parts by using advanced technology metals such as 316, 316L, 420, Inconel 625/718, cobalt/chromium and tungsten carbide (Exone, 2022). These materials are generally preferred in areas that require advanced engineering materials such as automotive, aerospace and defense industries. In addition to these, MBJ is preferred in special areas that require high mechanical properties and complex geometry (Mouritz, 2012).

As with other existing metal AM methods, the manufactured parts do not directly become functional in the MBJ devices. In MBJ, the first step is to manufacture low-density parts with raw strength. The mechanical properties of the part are brought to the desired level by applying various post processes. (curing, sintering, infiltration, etc.) Infiltration, which is one of these steps, is the process of infilling a metal with bronze, copper, aluminum etc. that melts at a lower temperature than the main part material to the part during sintering (Yin et al., 2007; Myers et al., 2015a; Myers et al., 2015b; Tuffile et al., 2016).

Working steps of MBJ method;

- Filling the powder feed chamber and binding material,
- Taking the powder from the feeding unit and spreading it smoothly on the build platform with the spreader mechanism,
- Spraying the binder on the necessary parts (according to the solid model geometry) of the powders,
- Curing process according to the characteristics of the sprayed binder (binders that harden in air, CO<sub>2</sub>, ultraviolet light, temperature, etc.),
- Repeating the spreading, binder spraying and curing processes in each layer until the production is finished,
- After the manufacturing process is completed, cleaning the manufacturing parts in from unbound powders,
- By taking the manufacturing part to the furnace process; burn out the binder and sintering metal powders,
- Infiltration in the gaps remaining from the binder evaporated from the part,
- If necessary, pressure sintering [usually Hot Isostatic Pressing] as a result of processes such as the manufacture of the functional part.

# 2. Material and Method

# 2.1. Material

Materials are;

- SS316L powder,
- Photopolymer resin,

- Isopropyl alcohol, -
- Para toluene sulfonic acid,
- Furfuryl alcohol

used in this study.

# 2.1.1. SS316 powder

Before beginning the sample manufacturing, an XRD examination of the 316L powder (with 20 – 53 µm grain sizes) was carried out and compared to the literature. The diffraction pattern of the powder analyzed in Figure 1. and in the literature in Figure 2. is given (Rokosz et al., 2015).



Figure 2. XRD of literature (Rokosz et al., 2015)

In Table 1., the percentages of materials are shown in comparison with the literature in terms of the powder material used. When both results were examined, it was seen that the powders were quite similar to each other, but the amount of oxygen in the powder used in the study was slightly higher.

 Table 1 Composition Percentage of \$\$316L Powder (FOS 2022: Registraw 2022)

Elements	Powder Used in Study	EOS	Renishaw
Si	0,85	0-0,75	0-1
Мо	1,5	2,25-3	2-3
Cr	17,55	17-19	16-18
Mn	2,24	0-2	0-2
Ni	11,16	13-15	10-14
С	0	0-0,03	0-0,03
S	0	0-0,01	0-0,03
Р	0	0-0,025	0-0,045
N	0	0-0,1	0-0,1
Cu	0	0-0,5	0
0	1,52	0	0-0,1
Fe	Balanced	Balanced	Balanced

# 2.1.2. Photopolymer Resin

Photopolymer (SLA) Resin becomes polymer when light binds molecular chains together. In this way, the layers are added on top of each other and the production is made. The compositions of the resins used in the field of AM vary according to their usage areas. Standard resins can be used for visual productions, and more durable resins with ABS properties can be used for mechanical prototypes. In some model works, semi-flexible resins are preferred (3D Durak, 2022). The resin type commonly used in the SLA method was tested for the first time in the MBJ method in this study.

# 2.1.3. Isopropyl alcohol

Isopropyl alcohol (IPA), known by the names propan-2-ol, isopropanol or 2-propanol, is a colorless, flammable chemical with a strong odor. Its chemical formula is C<sub>3</sub>H<sub>8</sub>O. It dissolves IPA, ethyl cellulose, polyvinyl butyral, many oils, alkaloids and natural resins (Wikipedia, 2022a). Within the scope of the study, IPA was used to dilute the SLA resin.

### 2.1.4. Para toluene sulfonic acid

Para toluene sulfonic acid (p-TSA); it is a white solid chemical that is soluble in water, alcohols and other polar organic solvents. It is a strong organic acid like other sulfonic acids (Wikipedia, 2022b). In this study, p-TSA was used to act as a catalyst for the hardening of furfuryl alcohol. p-TSA powder was mixed with SS316L powder and spread on the substrate layer by layer.

### 2.1.5. Furfuryl alcohol

Furfuryl alcohol (FA) is a furan-containing organic compound. It cures into a resin when treated with acids, heat or catalysts. Its primary use is to be synthesized the furan resin. FA is used in cements, adhesives, coatings and casting resins (Wikipedia, 2022c). FA was used to bind the metal powders in the layers by cure with the p-TSA in this study.

# 2.2. Method

The methodology of the study is given at Figure 3.



Figure 3. Diagram of methodology of the study

#### 2.2.1. Preparing powder

Powders for solid + liquid cured manufacturing p-TSA and SS316L were mixed and made ready for manufacturing.

# 2.2.2. Preparing binder

SLA resin has a wettability problem when sprayed onto powders. It can't wet the powders because of the surface tension. To prevent this, SLA resin was diluted with IPA.

# 2.2.3. Test part manufacturing

Within the study, the samples whose mechanical properties will be examined were modeled as  $\emptyset 10x10$  mm cylinders as in Figure 4.



Figure 4. Test part dimensions

The powder mixture of p-TSA and 316L was laid on the fabrication table and FA was sprayed onto the layer with the spray head.

The sample in Figure 4. was also produced with SLA resin on the same bench. After SS316L powder was laid in the manufacturing area, SLA resin was sprayed according to the relevant layer geometry. The UV light source was held on the layer to cure the sprayed resin. 3D part was produced by repeating these processes.

# 2.2.3. Heat curing

UV curing wasn't enough for SLA resin. Because of that, test parts manufactured with SLA resin were heat cured at 175<sup>o</sup>C for 120 minutes. The large volume of the furnace required a longer curing process than literature in order to distribute the temperature homogeneously throughout the furnace (Taormina et al., 2018; Binnion et al., 2016).

# 2.2.4. Sintering

The produced samples were sintered at three different temperatures (1100, 1150 and 1200<sup>o</sup>C).

In the way of the literature;

- Inside of the furnace made nitrogen atmosphere,
- Temperature increase to 450°C with 3°C/min heating rate,
- Waited at 450°C for 90 minutes,
- For make furnace vacuum atmosphere, vacuum pump was started,
- Temperature increase to sintering temperature (1100,1150,1200°C) with 10°C/min heating rate,
- Waited at sintering temperature 120 minutes,
- Waited the decreasing temperature to 150°C (by itself)

steps selected as furnace process (Vangapally et al., 2017; Johnston et al., 2004; Gülsoy, 2008; Do et al., 2017; Ren et al., 2017; Mostafaei et al., 2017; Enneti et al., 2018). Temperature-time graph during sintering process is given at Figure 5.



The initial heating rate is selected low for the best burn-out of the binder. With the same reason, waited 90 minutes at that temperature. The reason for choosing the nitrogen atmosphere is to prevent oxidation in the parts at high temperatures and to remove the burning resin vapor from the environment with nitrogen gas. With the start of the vacuum process, it is aimed to remove the resin vapor and nitrogen from the furnace atmosphere.

While the process was planned to proceed in this way, at the initial stage of the furnace process of the samples, which were planned to be sintered at 1150°C, the furnace suddenly rose to 300-350°C, then decreased to its normal values again. However, this rise burned some of the resin and prevented the samples from being sintered as desired. Another problem is that the furnace process, which is required to continue in a vacuum environment, could not create the desired furnace environment due to the lack of good vacuuming, and the resin could not be evaporated from the part.

# **3. Experimental Results**

The internal structures of the test parts were examined in the images examined under the microscope. The structure of the powder grains and their sintering necks are shown in Figure 6. The sinter necks formed show that the process was successful and the powder grains are made neck each other, although not at the desired level.



Figure 6. Microstructure of sintered sample

Figure 7. shows micro and macro pores, ferrite and austenite structures. Micro pores can be caused by the structure of the powder or for unknown reasons. Macro pores may be caused by the evaporation of the resin and the poor compaction of the powders.



Figure 7. Etched microstructure of sintered samples

In Figure 8, ferrite and austenite structure images are given from a study in the literature (Zietala et al., 2016). Austenite cells and ferrite structures between cells can be seen in the microscope image, which is given more closely in the study.



Figure 8. Etched microstructure from literature (Zietala et al., 2016)

In the atmosphere-controlled furnace environment, the nitrogen environment at first prevents the oxidation of the part. When the furnace reaches  $450^{\circ}$ C, the resin starts to burn out and mixes with nitrogen gas. After this moment, the vacuum pump is operated and the resin vapor mixed with nitrogen is removed from the environment. In cases where the vacuum cannot be done well, resin residues may remain in the part as seen in Figure 9.



Figure 9. Porosity and binder residue at microstructure

In Figure 10., 11. and 12., the consequence of a rapid temperature rises during the heating process of the samples sintered at  $1150^{\circ}$ C is seen. While micro cracks are not visible in the samples at 1100 and 1200°C; It is quite abundant in the sample sintered at  $1150^{\circ}$ C as seen in Figure 11.



Figure 10. Microstructure at 1100°C sintered part



Figure 11. Microstructure at 1150°C sintered part



Figure 12. Microstructure at 1200°C sintered part

Grain size values according to sintering temperature and resin type are given in Table 2. The average of the measurements taken from ten grains chosen randomly was used to calculate the grain sizes.

Number Of Measuring	Ø at 1100°C Sintering Temp.	Ø at 1150°C Sintering	Ø at 1200°C Sintering Temp. (um)
in an internet of the assuming	μm)	Temp. (μm)	p at 1200 0 cintoring romp. (pm)
1	53.573	84.384	89.602
2	56.060	89.899	94.081
3	57.825	91.178	124.981
4	77.907	103.355	125.331
5	79.232	107.946	148.681
6	80.769	110.147	154.090
7	84.929	121.392	159.860
8	105.820	125.276	173.710
9	111.786	127.692	174.281
10	117.648	130.605	174.991

In Figure 13. Graph of the grain diameter of the different sintering temperature.



#### Grain Diameters of The Sintered Parts

Figure 13. Graph of the grain diameters of the sintered parts

In the obtained values, it was observed that the increase in the sintering temperature increased the grain sizes, similar to the literature (Stawarczyk et al., 2013; Gupta, 1972).

After sintering situation, the composition of the parts was found. At Table 3. shown the percentage of the powder which before the AM process, SLA+IPA used parts and FA+p-TSA used parts.

Elements	Percentage of Powder	Percentage of SLA+IPA	Percentage of FA+p-TSA
Silicon	0,85	0,81	0,87
Molybdenum	1,5	1,33	2,98
Chromium	17,55	15,43	10,31
Manganese	2,24	2,31	0,57
Nickel	11,16	6,68	11,71
Carbon	0	12,67	5,41
Oxygen	1,52	17,05	2,95
Ferrite	Balanced	Balanced	Balanced

Table 3. EDS analysi	s of the sintered parts
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When the values in the table are examined, it is seen that the oxygen and carbon ratios increase as a result of sintering. The main reason for this is the non-pyrolysis binder. As a result of the burning of the binder, which cannot be removed from the part properly, at the sintering temperature, it causes the formation of oxygen and carbon waste structures in the part.

The ratio of C and O<sub>2</sub> among the samples produced with SLA resin is higher than those produced with p-TSA. This shows less evaporation of the resin in the part and burning residue inside the structure. The strength values of the samples decrease as the residue in the part increases.

#### 4. Result and Discussion

Within the study, for SS316L produced by the MBJ method; the parameters related to resin type and sintering temperature were examined. It was evaluated with which parameters it gave better results in microstructure analysis.

The rapid temperature changes in heating and cooling experienced during the furnace process at  $1150^{\circ}$ C caused the formation of micro cracks in the samples. In addition, as a result of the weak vacuum process in the furnace, it causes the binder to remain in the part. This binder remaining in the part burns during sintering and causes residue.

Looking at the microstructure, it was seen that sintering between the powder grains was successful, and larger grains were formed at high temperatures. Ferrite and austenite structures in the grain were obtained. Macro and micro pores were determined. Micro pores may have been caused by a defect in the powder structure, an unknown reason. Macro pores may be the result of holes remaining from the burn out of the resins or not compacting the powders well while spreading.

For future works,

- Working with lower layer thicknesses,
- Making productions by mixing powders in different powder diameter ranges with each other,
- Sintering of samples at various temperatures,
- Trial of low resin ratios,
- Using a sintering furnace with better vacuum conditions,
- Measurement of values such as strength, density and hardness by performing HIP or infiltration on the samples,

can be recommended.

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### **Conflict of Interest**

No conflict of interest was declared by the authors.

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