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Application of submerged arc welding at different amperages in the manufacture of storage tanks and examination of the weld zone

Depolama tanklarının imalatında farklı amperlerde tozaltı kaynağının uygulanması ve kaynak bölgesinin incelenmesi

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Application of Submerged Arc Welding at Different Amperages in The Manufacture of Storage Tanks and Examination of the Weld Zone

Highlights

- Submerged arc welding application using different amperage values of ASTM A36 steels.
- Investigating the effect of different amperage values with destructive and non-destructive testing methods.

Graphical Abstract

ASTM A36 steel plates are joined by submerged arc welding method using different amperage values.



Figure. Macrostructure images and tensile/elongation graph

Aim

Investigation of welding of ASTM A36 steels.

Design & Methodology

Joining ASTM A36 steels by submerged arc welding using values of 450 A, 475 A, 500 A, 525 A and 550 A.

Originality

Both non-destructive and destructive tests were performed on ASTM A36 steels joined by submerged arc welding using different amperage values.

Findings

As a result of destructive and non-destructive tests, it was determined that there were various welding defects in welded joints at 450 A and 475 A.

Conclusion

It has been determined that ASTM A36 steels can be joined by submerged arc welding using values of 500 A, 525 A and 550 A.

Declaration of Ethical Standards

The author(s) of this article declare that the materials and methods used in this study do not require ethical committee permission and/or legal-special permission.

Application of Submerged arc Welding at Different Amperages in the Manufacture of Storage Tanks and Examination of the Weld Zone

Araştırma Makalesi / Research Article

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ABSTRACT

In the study, ASTM A36 steel materials were combined with submerged arc welding using different amperages. Non-destructive magnetic particle (MT), liquid penetrant (SP), radiographic (RT), and ultrasonic (UT) examinations were performed on the joints. In addition, optical microscope, microhardness, bending, tensile, and notch impact tests were carried out on the welds. As a result of the RT and UT examinations, a lack of root penetration was found in the welds made at 450 A and 475 A. In the optical microscope examinations, the areas formed by the HAZ-weld metal transition were found to have a similar appearance for each amperage. In the microhardness studies, the hardness values are listed as weld metal, HAZ, and base material from high to low. From the notch impact tests, it was found that increasing the temperature increased the toughness value. From the bending tests performed on the joints where 450A and 475A were used, it was found that cracks and tears occurred. In addition, a rupture occurred in the weld metal during the tensile tests conducted on the joints made at 450 A and 475 A. For the joints made at other amperage values, it occurred in the base material.

Keywords: ASTM A36 steel, different amperage values, submerged arc welding, mechanical test.

Depolama Tanklarının İmalatında Farklı Amperlerde Tozaltı Kaynağının Uygulanması ve Kaynak Bölgesinin İncelenmesi

ÖΖ

Bu çalışmada, 10 mm kalınlığındaki ASTM A36 çeliği tozaltı ark kaynak yöntemiyle birleştirilmiştir. Farklı kaynak akımlarının kaynak bölgesine etkisi, tahribatsız ve tahribatlı muayene yöntemleriyle incelenmiştir. Kaynaklı birleştirmelerin tahribatsız incelemelerinde sıvı penetrant (SP), manyetik parçacık (MT), ultrasonik (UT) ve radyografik (RT) muayene yöntemleri kullanılmıştır. Kaynaklı birleştirmelerin makro-mikroyapı ve mekanik özelliklerini belirlemek için ise tahribatlı muayene yöntemlerinden optik mikroskop, mikrosertlik çalışmaları, çekme, eğme ve çentik darbe testleri uygulanmıştır. Tahribatsız muayene yöntemleri sonucunda; SP ve MT yöntemlerinde kaynak yüzeyinde herhangi bir süreksizliğe rastlanmamıştır. UT ve RT incelemelerinde ise 450 A ve 475 A kaynak akımında birleştirilen levhalarda kök nüfuziyet eksikliği gözlemlenmiştir. Tahribatlı muayene yöntemleri sonucunda; makroyapı incelemelerinde, 450 A ve 475 A kaynak akımlarında birleştirilen levhaların kök kaynaklarında eksik nüfuziyet gözlemlenmiştir. Mikroyapı incelemelerinde, kaynak metali-ITAB geçiş bölgelerinin birbirlerine benzer görüntüler sergilediği belirlenmiştir. Sertlik testlerinde sonucunda en yüksek sertlik değerleri, kaynak metalinden elde edilirken onu sırasıyla ITAB ve ana malzeme takip etmiştir. Çentik darbe test sonuçları incelendiğinde, sıcaklık yükseldikçe tokluk değerlerinin arttığı tespit edilmiştir. Eğme testleri sonucunda, 450 A ve 475 A kaynak akımlarında birleştirilen levhalarda yırtılma ve çatlak tespit edilmiştir. Çekme testleri sonucunda, 450 A ve 475 A kaynak akımlarında birleştirilen levhalarda kopma kaynak metalinde elekiştirilen levhalarda kopma kaynak metalinde gerçekleşmiş diğer kaynak akımlarında ise kopma ana malzemede gerçekleşmiştir.

Anahtar Kelimeler: ASTM A36 çeliği, farklı amper değerleri, tozaltı ark kaynağı, mekanik test.

1. INTRODUCTION

Steel consumption in the manufacture of storage tanks is quite high. In particular, tanks are manufactured from flat or bent plates. In recent years, the use of high-strength finegrained steels and light alloys has increased in order to extend the service life of storage tanks. However, the production costs of many materials and the need for low carbon steel for mass production should not be ignored [1]. ASTM A36 steels; It is widely used in construction machinery manufacturing, construction equipment manufacturing, general structural plates, various machine parts manufacturing, railway and land vehicle manufacturing [2-4]. This frequency of use has highlighted the need to join ASTM A36 steels by various methods. One of the most common joining methods is welding. Welding is a reliable and

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efficient metal joining process that is widely used in the infrastructure and heavy equipment industries, such as steel bridge construction, shipbuilding and the installation of large pipelines [3-6]. High joint efficiency, ease of installation and low manufacturing costs are the advantages of this joining process [7-11].

Submerged arc welding is a fusion welding process used to produce machine parts and tools with basic surface properties such as corrosion resistance, wear resistance and pressure tightness [12]. This process has a high material deposition rate and is often suitable for working in a horizontal position. In addition, the main difference between submerged arc welding and other welding methods is that the arc is completely immersed in granular flux and is not visible. This process minimises heat loss and can achieve thermal efficiencies in excess of 90%. Weld seams produced by submerged arc welding have high strength and ductility with low hydrogen and nitrogen content. Differences in submerged arc welding parameters (welding current, voltage, wire feed speed, type of flux used and welding conditions, etc.) affect the weld quality and the heat affected zone [13].

In this study, ASTM A36 steel was joined by submerged arc welding using welding currents of 450A, 475A, 500A, 525A and 550A. The effect of different current values on the weld

zones was investigated. In experimental studies, the weld zones were examined using liquid penetrant (SP), magnetic particle (MT), ultrasonic (UT), radiographic (RT), optical microscope, microhardness studies, tensile, bending and notch impact tests.

2. EXPERIMENTAL STUDIES

Experimental studies were carried out on ASTM A36 steel plates with dimensions of 400x150x10 mm. The chemical content of ASTM A36 sheets, welding consumables and powders used in the welding process are given in Table 1. The mechanical properties of ASTM A36 steel are given in Table 2.

In accordance with EN ISO 15609-1, a 30° V weld groove was opened on ASTM A36 plates and fix from the rear, leaving a 1 mm gap. Prior to welding, the plates were turned upside down and the root pass was made using the electrode arc welding method. The plates, which had been cleaned, were then joined in a single pass using submerged arc welding at different current levels. The parameters used in the welding processes are given in Table 3. The sample image after welding is shown in Figure 1.

Table 1. Chemical content of base material, filler metal and powder (% by weight)

Alloying Element (%)	С	Mn	Р	S	Si	Fe
ASTM A36	0,25	1	0,03	0,03	-	Balance
Filler Metal	0,04	1,3	0,025	0,02	0,25	Balance
Magma weld SF124 (powder)	0,05	1,3	0,025	0,02	0,25	Balance

Base Material	Yield Strength	Tensile Strength	Elongation
	(N/mm ²)	(N/mm ²)	(%)
ASTM A36	250	400-550	22

Table 3. The welding paramete	rs
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Filler Metal Diameter	Current	Current	Volt	Wire Speed	Heat Input
(mm)	Туре	$\begin{array}{c c c c c c c c c c c c c c c c c c c $	(cm/min)	(kJ/mm)	
		450	29	28	2,6
		$\begin{array}{c c c c c c c c c c c c c c c c c c c $	2,8		
3.2	DC +	500	30	35	3
		525	30,5	38	3,2
		550	31	40	3,4



Figure 1. Sample image after welding

The plates joined by submerged arc welding were first subjected to non-destructive testing (NDT) by an NDT level 2 expert. BETA brand BT-68 penetrant, BT-70 developer and BT-69 cleaner were used for liquid penetrant testing. Magnetic particle testing was carried out using a Magnaflux magnetic testing device. Ultrasonic testing of the specimens was performed using a Krautkramer USM36 ultrasonic testing device with a probe at 60° and 70° angles. Radiographic testing of the plates was carried out on a Detay Quality radiographic testing machine.

Following the application of non-destructive testing methods, the welded plates were prepared for destructive testing. Specimens were taken from the plates using a water-cooled band saw for macro/micro imaging, hardness, tensile, bending and notch impact studies. In addition, these samples were taken parallel to the rolling direction. The specimens used for macro/micro imaging were ground with sandpaper in accordance with standard metallographic specimen preparation procedures. It was polished with 3 micron diamond paste and then etched with 2% Nital solution. The microstructural studies were carried out using an Inverted Brand optical microscope. Three samples were used for each of the mechanical tests (tensile, hardness, flexural and notch) and the results were averaged over the three samples. Hardness testing was carried out using a Hardway model HV10 AP fully pneumatic variable load instrument. Vickers hardness values were used for hardness testing and a 10 kg load was applied during measurements. Tensile tests were carried out at room temperature at a tensile speed of 5 mm/min. Bending test specimens were prepared with dimensions of 130x40 mm and the mandrel diameter was determined to be 20 mm. The bending tests were performed using a Hardway/WAW 600D tensile tester at a bending speed of 10 mm/min. For the notch impact test, a 55x10x5 mm specimen of weld metal and HAZ was prepared. The notch impact tests were performed by the Charpy method using a Hardway/JB 300 B tester with a capacity of 300 joules. Tests were performed at temperatures of -20, 0 and 20 °C.

3. RESULTS and DISCUSSION 3.1. Non-destructive Testing

3.1.1 Penetrant liquid inspection

The tests were carried out by a liquid penetrant expert (Figure 2) and reported in accordance with EN ISO 23277 (Figure 3).



Figure 2. Liquid penetrant testing application image

When the samples were examined after the liquid penetrant test, no surface defects such as cracks, voids, pores and undercut were found on the weld surfaces. The expert's approved report stated that there were no surface defects in any of the welding currents used and that the welds were suitable. Özkan [14] stated that no discontinuity was found on the weld surface by liquid penetrant testing of structural steels after welding.

3.1.2. Magnetic particle inspection

It was applied by a magnetic particle inspection test expert in accordance with the TS EN ISO 9934 standard (Figure 4) and reported in accordance with the EN ISO 23277 standard (Figure 5). Upon examination of the magnetic particle test results, no surface defects such as cracks, pores, or combustion grooves were found on the weld surfaces. In the report approved by the expert, it was determined that there were no superficial defects in all welding currents used and that the welds were in compliance with the standards. Özkan [14] reported that no cracks, pores, etc. were detected on the weld surface as a result of magnetic particle tests performed after the welding of structural steels.

					Report No.	8	PT-001		
	LIQUID PENETRANT INSPEC				Sheet/Page	0	1/1		
					Date		16.3.2020		
Surface Temp	erature : T<10	10≤	Г≤38 ⊻	T>38	<u> </u>		MACHINED		
Surface Cond	ition : AS IT IS 🗹	S	AND BLASTED	GRO	DUND		MACHINED		
Welding Proc	cess : GTAW	SMAW [GMAW	SAW					
Joint Type :	FILLET	V	~	x 🗆	К		0		
Penetrant Bra	and: BETA PROSES BT68		Dwell Time:	20 mins.	in the second	UV- Light Inter	nsity: 500 lux		
Developer Brand: BETA PROSES BT70		Developer Time:	20 mins.		and and the second second	and have approved			
Cleaner Bran	d: BETA PROSES BT69			a state and a state	112.100	1 - The AV	The second second		
Sr. No	Description / Joint no	Thickness/	Material Grade	Observation/ Type of Defect	Defect Location	Remarks	Result		
1	450 A	10mm	ASTM A36	and the second		1.640	A		
2	475 A	10mm	ASTM A36				A		
3	500 A	10mm	ASTM A36				A		
4	525 A	10mm	ASTM A36				А		
5	5 <u>50 A</u>	10mm	ASTM A36			X	А		
LEGEND	for RESULT : A : Accepted	R: Repair/ to	be corrected			1	0 (
LEGEND	OF TYPE OF SURFACE DEFE	CTS: LC: Long	itudinal Crack TC:	Transverse Crack C	C: Crater Cra	ack MSP: Micro	Surface Porosity		
	PREPARED BY (H	AZIRLAYAN)		and the second sec	APPROVE	D BY (ONAYLA	YAN)		
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Figure 3. Liquid penetrant inspection report



Figure 4. Magnetic particle test application image

ма	GNETIC PARTIC	LE TEST	REPO	RT	Report No. Sheet/Page	MT-001	
1					Date	16.3.2020	
MATERIAL THK.: 10	mm						
Magnetic force appl	ication: AC	DC 🔄	CO	NSTANTLY	HALF WAY	OTH	IER
Magnetic Particle Di	rection :	CIRCUL	AR 🗌			LONGITUDNAL	7
EQUIPMENT TYPE: 1	MAGMA FLUX					and the second second	a state
Test Temperature:	23° C		Light Inten	sity: 500 lux	and the second		
Surface type:	GROUND	SMOOTH	2		COARSE		
Surface condition :	WET	DRY	2	FLO		CONTRAST	
Welding Process: SA	AW	20.020 C	Heat Treat	ment:	Before	After 🗌	No
Joint Type: BW		1	Demagneti	zasyon:	Applied	Not Applie	ed 🔽
Item No.	Weld/ Part No.	Weld/te	st Length	Deffect Type	Rem	arks	Result
1	450 A	400	mm		Charles Ve	- Shares	A
2	475 A	400	mm			Station of the second	A
3	500 A	400	mm	1. 1. 1. 1. 1. 1.	100		A
4	525 A	400	mm	The state	6	51	A
5	550 A	400	mm	1 Stands		ALC BAR	A
Legend A: Ac	cepted R: Repaired / to be	corrected					
	PREPARED BY (HAZIRI	LAYAN)			APPROVED BY	(ONAYLAYAN)	
	ATILGAN			NAME: Korom	CANAKCI	(annual annual)	
VAME: Yiğitcan				TAMINIC. NEICHI	LANANCI		

Figure 5. Magnetic particle test report

3.1.3. Ultrasonic examination

To detect subsurface/section defects, an ultrasonic test was applied by an expert according to TS EN ISO 23279 standard (Figure 6) and reported (Figure 7).

When the ultrasonic examination results were examined, a lack of penetration at the root (insufficient penetration) was observed in processes using 450 A and 475 A welding currents. As sufficient heat input cannot be achieved at these current values (Table 3), it is thought that full penetration between the root pass and the weld (Figure 10) cannot be achieved. No discontinuities (undercut, residue, pores, cracks, etc.) were observed in the plates joined at 500 A, 525 A and 550 A. In the report approved by the expert, it was determined that the plates combined with 450 A and 475 A welding currents needed to be repaired, while the plates combined with 500 A, 525 A, and 550 A welding currents were suitable.



Figure 6. Ultrasonic inspection application image

As a result of ultrasonic tests applied to various steel materials joined by the submerged arc welding method, cracks, lack of penetration, gas gaps, etc. have been detected in the weld seam zone. It has been stated that no volumetric errors were encountered [14-17].

				18.4			Report No.	UT-001			
	ULTRAS	SONIC INS	PECTI		EPORT		Sheet/Page	1/1			
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						22.4					
/eld Process		SMAW(111) SAW (1:	21) 🗹 GT.	AW (141) 🗌	GMAW (131/	135) Other		VXU	к	_	
leld Joint Typ	pe	Butt Weld 🔽	Fillet Weld]	Branch Weld] T-joint 🗌	Joint Design		0 0	ther	
est Standard	1	EN ISO 17640 🔽 Technique 1	ASME S	ECV 🗆	Diğer Other					-	
Evaluation St	tandard	EN ISO 11666 🗹	ASME	. 🗆	Diğer Other		To all				
Heat Treatme	ent Condition	Before	After 🗌		N/A 🗹	Tes	t Temperature	23 * C	-		
Material Th	nickness 10		, and		Materia	al AS	TM A36	The second second	- Alton	100	
			31715		and a second	1-2-2-1	2 CHANNE	1071			
INSTRUMENT USED			MAKE			MODEL/TYP	E	-			
HATA DEDEKTÖRÜ KRAU			UTKRAMER USM 36					-			
OLÇO: 7° (Inch) KAPASİTE:				8 GB, SD-Kart							
ÇÖZÜNÜRLÜK: 800 x 400 piksel		DARBE: Eksen başına 1000 çevrim							24		
HIZ:		250 16,000 m/s	KORUMA: 1966 / IEC 60529								
PROB BAĞLAN	TILARI:	2 x LEMO - 1 veya 2 x BNC	ÇALIŞMA SICAKLIČ	ŝł:	10	The Contractory	and the second second	110000	The Are		
ITEM	WELD No.	Lo	cation / Dimen	sion of Defe	ct	Test	DITE OF DUDE	DEFECT	Ch (ALLIA)		
No.	(Part No.)	11 (Q1) mm	12 (Q2) mm	HVLmn	b (t) mm	Lenght (mm)	DEFECT TYPE	LOCATION	EVALUAT	ION	
1	450 A			1			Db	0-30	R		
2	475	4	1 million			1	c	0-30	R		
3	500 /	4					с	0-30	A		
4	525 A	1	1			and the	С	0-30	A		
5	550 A	4	1.200	1	O King	- B.W.	с	0-30	A		
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Figure 7. Ultrasonic inspection report

3.1.4. Radiographic examination

It was applied (Figure 8) and reported (Figure 9) by the radiographic examination specialist according to TS EN ISO 17636 standard.

When the radiographic examination results were examined, a lack of penetration into the root was observed in processes using 450 A and 475 A welding currents. No defects were found in the plates joined at 500 A, 525 A and 550 A. In the report approved by the expert, it was stated that the plates joined at 450 A and 475 A welding currents needed to be repaired. However, it has been stated that plates combined with welding

currents of 500 A, 525 A, and 550 A are suitable. Canli [18] carried out radiographic examinations on welded connections in the production of storage tanks. Several welded joints in the storage tank exhibited pores in different directions (horizontal and vertical), slag residues in filler material, cracks caused by shrinkage in the root, and areas of insufficient penetration in some of the weld seams. It was also noted that undercut and cracks were present in the root areas of a few of the weld seams.



Figure 8. Radiographic examination method images

3.2. Destructive Testing

3.2.1. Macrostructure studies

Macroscopic examination, cross-sectional macro images (Figure 10) of the plates joined by welding according to TS EN ISO 17639 standard were examined. When the macrostructure photographs were examined, it was determined that the single-pass welding of the sample joined at 450 A did not penetrate the root pass and a melting/penetration defect of approximately 2 mm in diameter occurred. Additionally, it was determined that a melting/penetration deficiency of approximately 0.5 mm in diameter occurred in the sample joined at 475 A. No welding mistakes were observed in the samples joined at other welding currents. Additionally, the weld metal and base material are clearly distinguished in the macrostructure images. In addition, it was seen from the images that single-pass welding and root pass processes were performed in all the samples combined with different welding currents. Additionally, it was determined that there was no symmetry error in the weld pass and root pass processes and no sagging occurred in the root pass. As a result of the macrostructure examinations of pressure vessel steels joined by the submerged arc welding method, it has been reported that no welding defects were encountered in the welding zone of the joints [18,19].



Figure 9. Radiographic examination report



Figure 10. Macrostructure images

3.2.2. Microstructure studies

Microstructure images of plates welded with submerged arc welding using different welding currents are given in Figures 11-15. When the microstructure images were examined, similar structures were seen at the HAZ, weld metal and melting line boundaries of the plates joined using different welding currents. It was observed that the HAZ grain sizes became slightly larger due to the increasing heat input as the welding current increased. In the images, the white grains have a ferritic structure and the black grains have a pearlitic structure. Moreover, preeutectoid ferrite and widmanstätten structures were also detected in the weld metal.



Figure 11. Microstructure of the plate combined at 450 A welding current



Figure 12. Microstructure of the plate combined at 475 A welding current



Figure 13. Microstructure of the plate combined at 500 A welding current



Figure 14. Microstructure of the plate combined at 525 A welding current



Figure 15. Microstructure of the plate combined at 550 A welding current

When the images are evaluated in general, it is observed that the grains in the HAZ become larger, whilst in the weld metal the grains have a columnar structure and extend in the direction of heat flow (towards the base material). Eroğlu and Aksoy [20] investigated the effects of heat input change on microstructure/mechanical properties in welded joints. According to microstructure examinations, it was stated that high heat input slows down cooling and solidification and causes more grain coarsening. It was also observed that the weld metal is made up of large and columnar grains, which are directed towards the centre of the weld metal. The microstructures of ship plates of different thicknesses joined by submerged arc welding method were examined. As a result of the investigations, it has been reported that grain boundary ferrite, widmanstätten ferrite, acicular ferrite, pearlite and martensite structures may form in the weld metal of low carbon and low alloy steels, depending on the cooling rate. Additionally, as a result of microstructural studies, it was stated that it consists

mainly of acicular ferrite and polygonal ferrite structures [21].

3.2.3. Hardness test

The graph obtained because of the hardness tests applied horizontally to the welding zone of the plates joined by the submerged arc welding method is given in Figure 16.

When the microhardness results of the joints made using different welding currents was explored, it was observed that the measured hardness values were close to each other, but the samples joined using 550 A welding current had higher hardness values. It is thought that the increase in hardness values due to the increase in current values is due to heat input. It was determined that the hardness values of the weld metal in all samples were higher than the HAZ and the base material. The highest hardness value was determined in the weld metal, followed by HAZ and base material. The difference in hardness values of weld metal, HAZ and base material varies depending on the heat and cooling rate they are exposed to. It is thought that the increasing Mn ratio in the chemical composition of the additional wire and welding flux used may be effective as the reason why the hardness of the weld metal is higher than that of the HAZ and the base material.

Weld metal hardness values were measured as (450 A) 200 ± 5 HV, (475 A) 202 ± 5 HV, (500 A) 203 ± 5 HV, (525 A) 205 ± 5 HV, and (550 A) 207 ± 5 HV. HAZ hardness values were measured as (450 A) 185 ± 5 HV, (475 A) 186 ± 5 HV, (500 A) 188 ± 5 HV, (525 A) 190 ± 5 HV, and

approximately 0.5 mm in the sample joined at 475 A welding current. Tensile test results confirm the results of non-destructive testing and macrostructure studies.

The tensile strengths of the welding areas of the plates joined at 500 A, 525 A and 550 A welding current were determined as $(541\pm5 \text{ N/mm2})$, $(552\pm5 \text{ N/mm2})$ and $(567\pm5 \text{ N/mm2})$, respectively. These values were found to be higher than the tensile strength of the base material (528 N/mm2). The reason why the samples joined in these welding currents show higher tensile strength can



Figure 16. Hardness test results

(550 A) 191 \pm 5 HV. The hardness value of the base material was measured as 160 \pm 5 HV. In hardness measurements of steel joints previously made by submerged arc welding [22-25], the highest value was determined the weld metal. It has been stated that there is a decrease in hardness values as one moves towards the base material.

3.2.4. Tensile test

The graph of the tensile tests applied to the sheets joined using the submerged arc welding method is given in Figure 17, and the rupture images after the tests are given in Figure 18.

When the tensile test results were examined, the rupture occurred in the welding zone in the plates joined to 450 A and 475 A, while the rupture occurred in the base material in the plates joined to 500 A, 525 A and 550 A. It was determined that the tensile strength of the sheets joined at 450 A (435±5 N/mm2) and 475 A (458±5 N/mm2) welding currents was lower than the tensile strength of the base material (528 N/mm2). Ultrasonic (Figure 7), radiographic (Figure 8-9) and macrostructure (Figure 10) results showed that 450 A and 475 A values were not appropriate. In the sample joined at 450 A welding current, it was observed that the welding performed in a single pass did not penetrate the root pass and a lack of melting/penetration occurred with a diameter of approximately 2 mm. It was observed that there was a melting/penetration deficiency of

be explained by the fact that the hardness values of the weld metal and HAZ (Figure 16) are higher than the base material due to the increasing heat input (Table 3) in parallel with the increase in the welding current. Similarly, according to the previous analysis results, the results of the non-destructive and macrostructure studies did not indicate that any welding defects were found in the welding area, and the tensile test results were also supported by these results. The samples that broke in the weld area after the tensile test broke brittle due to the welding defects mentioned above. However, the samples that fractured in the base material experienced ductile failure, and the weld zone did not show any visible damage. In the samples united at 500 A, 525 A or 550 A, the rupture zones were observed in the base material, not in the weld metal or HAZ.

In addition, samples welded at 450 A and 475 A showed 8.4 and 9.3% elongation in tensile tests due to the welding errors mentioned above. While the % elongation of the base material was 22%, the samples joined at 500 A, 525 A and 550 A exhibited % elongation of 21.2%, 19.7% and 18.6%, respectively. The reason why these samples exhibit less % elongation than the base material can be explained by the fact that the hardness values of the weld metal and HAZ are higher than the base material due to the increasing heat input in parallel with the increase in welding current (Figure 16). Similar results have been reported after tensile tests applied to steels joined by submerged arc welding method [19,26,27].



Figure 17. Tensile and elongation graph



Figure 18. Fracture images after tensile test

3.2.5. Bending test

Sample images after the 180° bending test applied to welded samples are given in Figure 19. When the sample images were examined after the bending test, cracking and tearing occurred in the welding zones of the samples joined using 450 A and 475 A. However, no welding defects were observed in the samples joined using 500 A, 525 A and 550 A. These results also coincide with the

non-destructive tests and macrostructure studies applied to the samples in the previous sections.

As a result, it has been determined that the samples joined at 450 A and 475 A using the submerged arc welding method are not suitable for use by bending, while the samples joined at 500 A, 525 A and 550 A welding current can be used by bending under service conditions. As a result of bending tests applied to different materials joined using the submerged arc welding method, it was reported that no visible damage occurred in the samples when appropriate welding parameters were used [28,29].



Figure 19. Images of the sample after the bending test

3.2.6. Notch impact test

Notch impact tests were conducted to ascertain the impact toughness of the sheets joined via the submerged arc welding method, the weld metal and HAZ at temperatures of -20 oC, 0 oC, and 20 oC. Figure 20 indicates the weld metal, while Figure 21 illustrates the HAZ's impact toughness.

When the impact toughness graphs were investigated, it was seen that the highest toughness was obtained at the test temperature of 20°C (room temperature), followed by the test temperatures of 0°C and -20°C, respectively. It was determined that as the test temperatures decreased, the impact toughness of the welded samples also decreased. It has been reported that after notch impact tests applied to steels joined by submerged arc welding method, impact toughness decreases with decreasing test temperature [15-17,28].

When weld metal impact toughness is examined, it is seen that there is a significant decrease in the samples joined at 450 A and 475 A. The reason for this decrease, as stated in the previous sections, is the lack of penetration at the root in the samples combined with these welding currents. Results inversely proportional to hardness were determined in the samples joined at 500 A, 525 A and 550 A. HAZ impact toughness was similarly found to be inversely proportional to the hardness results. When weld metal and HAZ impact toughness's are compared, it is seen that HAZ impact toughness's are higher than weld metal toughness's. When the impact toughness was compared with the hardness test results, it was determined that the toughness values decreased as the hardness increased. In a study, it was reported that as hardness values increased, toughness values decreased [30].



Figure 20. Weld metal impact toughness chart



Figure 21. HAZ impact toughness chart

4. CONCLUSIONS

- No discontinuity was observed on the weld surfaces in liquid penetrant and magnetic particle tests.
- In ultrasonic examinations, a lack of root penetration was observed in the sheets joined using 450 A and 475 A. No discontinuity was observed in the plates joined using 500 A, 525 A and 550 A.
- As a result of ultrasonic examination, lack of penetration at the root (insufficient penetration) was observed in the plates welded at 450 A and 475 A. No discontinuities (undercut, residue, pores, cracks, etc.) were observed in the plates joined at 500 A, 525 A and 550 A.
- As a result of macrostructural examinations, it was observed that there was melting/penetration insufficiency in the samples joined at 450 A and 475 A welding current. No welding defects were detected in the welding area of the samples joined using other welding currents.
- In the microstructure images, it was observed that the grains became larger in the HAZ, while the grains in the weld metal had a columnar structure and extended in the direction of heat flow (towards the base material).
- When the hardness values were examined, the highest hardness values in all samples were obtained from the weld metal, followed by HAZ and the base material, respectively.

- While the rupture occurred in the welding zone in the plates welded using 450 and 475A in the tensile tests, the rupture occurred in the base material in the samples combined with other welding currents.
- In bending tests, cracking and tearing are observed in the welding zones of the samples joined using 450 and 475A, while there is no cracking, tearing, etc. in the samples joined using other welding currents. No welding errors were found.
- As a result of the notch impact tests, it was determined that the highest toughness values were obtained at the test temperature of 20°C (room temperature), followed by the test temperatures of 0°C and -20°C, respectively.

DECLARATION OF ETHICAL STANDARDS

The authors of this article declare that the materials and methods used in this study do not require ethical committee permission and/or legal-special permission

AUTHORS' CONTRIBUTIONS

Yiğitcan ATILGAN: Perofrmed the experiments.

Mehmet Serkan YILDIRIM: Analyzed the results and wrote the manuscript.

Yakup KAYA: Analyzed the results.

CONFLICT OF INTEREST

There is no conflict of interest in this study.

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