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Research Article

Experimental Investigation on the Tensile Behavior of MWCNT – Nano Silica Epoxy Hybrid Nanocomposites

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ARTICLE INFO ABSTRACT

Article history: Received 22 January 2024 Accepted 27 May 2024 Published 20 August 2024 *Keywords:* Carbon nanotube, Epoxy Hybridization Nanocomposite Nano silica Tensile behavior

This study aimed to investigate the mechanical properties of epoxy nanocomposites filled with single and binary combination of multiwalled carbon nanotube (MWCNT)-nano silica (NS) nanoparticles. The epoxy nanocomposites were produced using the mold casting method, and different filler ratios were employed to create nanocomposite samples. For the single particle-filled samples, filler ratios of 0.1, 0.3, 0.5, and 0.7 wt.% for MWCNT and 0.5, 1.0, and 1.5 wt.% for NS were used. Additionally, hybrid samples were produced using filler ratios of 1:1, 1:2, and 1:3 (MWCNT: NS). The tensile test results indicated notable enhancements in the tensile behavior of the nanocomposite, attributed to the integration of nanoparticles into the epoxy matrix. Particularly, tensile strength values improved by 28.35% and 21.25% in C3 (0.5 wt.% MWCNT) and S2 (1.0 wt.% NS) composite samples compared to the pure sample, respectively. Additionally, the hybrid nanoparticle-filled composite samples introduced a synergistic effect on the tensile behavior of the nanocomposite. Especially, the hybrid sample H1 (1:1) showed the maximum enhancement in tensile strength by 44.26%. Significant improvements were also observed in tensile strain values. Compared to the control sample, the maximum improvement was recorded as 143% in the H2 hybrid sample (1:2).

1. Introduction

Epoxy, a prominent thermosetting polymer, has garnered considerable attention in recent years owing to its exceptional properties, including elevated tensile strength, minimal shrinkage during the curing process, commendable chemical and thermal resistance, and superior adhesion. Despite these favorable attributes, the inherent brittleness of epoxy polymers poses a significant drawback, characterized by susceptibility to crack initiation and propagation [1–3]. This brittleness is caused by the presence of microcracks, and micro voids induced by polymerization, which limits their effectiveness in high-performance applications, especially in aerospace structures [4–6].

Over the past decade, there has been a growing significance attributed to epoxy nanocomposites, wherein epoxies are augmented with micro/nano-sized particles, owing to their exceptional properties. The large surface area of the incorporated nanoparticles confers new macroscopic properties such as improved mechanical hardness and enhanced fracture toughness [7–11].

Seloğlu et al. [12] conducted a comparative study on the

effect of the individual addition of multi-walled carbon nanotube (MWCNT), nano silica (NS) and nano-ZnO (NZ) on the mechanical properties of geopolymer mortar composites. In comparison to the control samples, the study reports that the mechanical strength of all samples was found to be significantly enhanced by the addition of individual nanomaterials. Among the various nanomaterials tested, the samples containing MWCNT exhibited the highest levels of compressive and flexural strength. In addition, geopolymer mortar samples containing NS and NZ also showed significant improvements in strength, with samples containing NS performing better than those containing NZ. In another study, Zappalorto et al. [13] examined the mechanical response of an epoxy/silica nanocomposite system. They conducted tensile and fracture tests and analyzed the results. Additionally, the researchers investigated the impact of the curing procedure on the mechanical properties of the nanocomposites, using two different curing conditions. Their results show that the fracture toughness epoxy is enhanced by the introduction of nanoparticles. Depending on the curing process, the

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strength values of the samples either increased or decreased. Kumar et al. [14], compared the tensile and flexural properties of nano SiO_2 epoxy polymer nanocomposites. They carried out tensile and 3-point bending tests on nanocomposites featuring varying SiO₂ contents of 2, 4, 6, and 8 wt.%. Their findings indicated that the mechanical behaviors of $SiO₂/epoxy$ nanocomposites exhibited improvement with the dispersion of fillers, particularly up to 4 wt.% of $SiO₂$ nanoparticles. However, beyond the 4 wt.%, there was a decline in mechanical properties. This deterioration was attributed to a substantial increase in the agglomeration and sedimentation of $SiO₂$ nanoparticles during the extended curing time. Specifically, a 4 wt.% $SiO₂$ dispersion led to increases in tensile strength of 30.57% increase, flexural strength of 17%, and flexural modulus of a remarkable 76%. Kaybal et al. [15] explored the distribution of nano $SiO₂$ and its impact on mechanical properties. They produced nanocomposites with varying weight percentages of $SiO₂$, ranging up to 5% by weight. The test results revealed that the optimal tensile and bending strength values were achieved when incorporating 3% by weight of nano SiO₂ into epoxy nanocomposites. It has been reported that beyond this concentration particle agglomeration commenced, leading to an adverse on the mechanical properties of the nanocomposite. In a related investigation, Chen et al. [16]examined the mechanical behaviors of epoxy composites incorporating nano silica with minimal agglomeration. They observed substantial enhancements in tensile modulus and fracture toughness for nanocomposites featuring filler content below 10%.

Because of their exceptional mechanical strength, high heat conductivity, distinctive microstructure, and resistance to corrosion, carbon nanotubes (CNT) have been the subject of several studies [17–19]. Bansal et al. [18] conducted a comprehensive study to investigate the mechanical effects associated with the introduction of very small concentrations of CNT as reinforcement. Remarkably, even with a minimal CNT reinforcement of 0.25 wt.%, the epoxy-CNT composite exhibited significant improvements in both elastic modulus and hardness values. Specifically, the elastic modulus increased from 2.87 GPa to 3.93 GPa, and hardness values rose from 0.208 GPa to 0.242 GPa. The outcomes of this study underscore the positive influence of even modest CNT reinforcements on the mechanical properties of the composite material. Gantayat et al [19] conducted an analysis of the mechanical properties of an epoxy resin polymer matrix through the incorporation of varying weight percentages (0.4, 0.6, and 1.0 wt.%) of MWCNT. Preceding their inclusion, MWCNT underwent a chemical treatment involving mixed acid for functionalization. This functionalization process effectively disrupted the atomic forces between the nanotubes, mitigating their tendency to

agglomerate. The tensile strength and modulus of the epoxy experienced noteworthy enhancements attributed to the good dispersion of functionalized MWCNT within the epoxy polymer and the establishment of robust interfacial adhesion between the epoxy matrix and MWCNT. The strength gradually increased and reached its maximum enhancement of 27% at 0.6 wt.% of functionalized MWCNT inclusion. Similarly, tensile modulus displayed a maximum enhancement of 14% under the same filler content. Montazeri et al. [20] studied the mechanical properties of a composite, examining the impact of both untreated and acid-treated MWCNT. The study revealed that the incorporation of MWCNT at specific weight ratios led to notable enhancements in the mechanical properties of the composite. Furthermore, their another study [21] concentrated on both the mechanical and viscoelastic effects resulting from the addition of MWCNT to epoxy resin at varying weight ratios (0, 0.1, 0.5, 1, and 2 wt.%), demonstrating an overall improvement in the viscoelastic properties of the nanocomposite. Specifically, the investigation highlighted that the concentration of 0.5 wt.% MWCNT in epoxy resin yielded the most significant enhancement in viscoelastic properties when compared to both the pristine epoxy and other nanocomposite formulations. Salman et al. [22] fabricated nanocomposites through the casting method, incorporating single-walled carbon nanotubes (SWCNT) into resin matrices at varying weight concentrations (0.1, 0.3, 0.5, and 1 wt.%). Numerous mechanical tests, including tensile, flexure, and hardness tests, were conducted on the nanocomposites. Additionally, scanning electron microscope (SEM) micrographs illustrated the homogeneous dispersion of SWCNT within the epoxy. Furthermore, Raman spectroscopy and x-ray diffraction (XRD) results corroborated the findings from SEM. The research elucidated enhancements in mechanical properties attributable to the uniform distribution of SWCNT within the epoxy matrix. Ultimately, there were notable increases in elastic modulus, tensile strength, flexural strength, and hardness values. Gojny et al. [23] examined the mechanical properties of epoxy nanocomposites featuring distinct types of CNT at filler ratios of 0.1, 0.3, and 0.5 wt.%. The investigation encompassed SWCNT, Double-Walled CNT (DWCNT), MWCNT, and amino-functionalized DWCNT. The findings of the study assert that the incorporation of CNT leads to enhancements in the mechanical properties of the nanocomposites.

Researchers have also shown significant interest in hybridized nanoparticle filled composites due to their exceptional mechanical, electrical, and thermal attributes. A study conducted by Ismail et al. [24] investigated the impact of MWCNT and $SiO₂$ particles on the mechanical properties of natural rubber nanocomposites. The study

found that MWCNT/SiO₂ hybrids showed improvements in the mechanical performance of the nanocomposite, such as increased tensile strength, and tensile modulus. This observation underscores the advantageous effects of combining diverse nanoparticle materials. A comprehensive comparative analysis of all mixtures elucidated that hybrid matrices outperformed their single filler counterparts across various parameters, including tensile strength, fracture mechanics, electrical conductivity, and thermo-mechanical properties [25].

Based on the literature review, although the effect of single and binary combinations of MWCNT and NS nanofillers on the tensile properties of rubber nanocomposites has been studied, it appears that there is no detailed experimental study on its effect on the mechanical performance of commercially widely used epoxy nanocomposites. This study aims to assess the effect of utilizing binary (hybrid) and single-phase MWCNT and NS particles on the tensile characteristics of composites. For this purpose, the epoxy resin was mixture with various contents of hybrid and single-phase nanoparticles. Then, the composite samples underwent tensile tests to thoroughly examine their tensile properties.

2. Material and Method

2.1 Materials

Nano silica particles from Graphene Chemical Industries Co., Turkey, and multiwalled carbon nanotubes from Nanografi Nanotechnology Co., Turkey, were utilized as a reinforcement phase. Epoxy matrix used in the production of nanocomposite in this study was purchased from Dost Chemical Co., Turkey. To create the thermosetting epoxy resin system, Momentive MGS L160 and an amino-acid-based hardener MGS H160 were utilized. The resin system was then mixed at a 100:35 mass mixing ratio. The physical properties of the materials used in the study are presented in Table 1 [26].

2.2 Preparation of the Nanocomposite Samples

The effect of MWCNT and NS on the mechanical behaviors of composite was systematically investigated by producing composites with different nanofiller configurations, both single and in hybrid combinations, at specific weight ratios. The study involved the incorporation of 0.1, 0.3, 0.5, and 0.7 wt.% MWCNT and, 0.5, 1.0, and 1.5 wt.% NS for single-filled configurations. In accordance with the findings from Rahmanian et al., it was identified that the presence of nanoparticles exceeding 1 wt.% led to a disturbance in the distribution morphology of combined fillers. Considering this observation, hybrid nano-filled (binary combination of MWCNT/NS) mixtures were carefully formulated at a concentration of precisely 1% by weight within the epoxy matrix. This strategic approach was undertaken to ensure optimal

compatibility and nano filler distribution, thereby maintaining the integrity and performance of the epoxy matrix [27]. Thus, hybrid nano-filled configurations were produced with nanofiller additives at hybridization ratios of 1:1, 1:2, and 1:3 (MWCNT: NS). A total of eleven different mixture types of nanocomposite samples were produced. A naming convention detailed in Table 2 was used to determine the nanoparticle contents and weight ratios of the samples.

To improve the nanofiller dispersion and reduce the viscosity of the epoxy resin, 99% pure acetone was introduced to nanoparticle- epoxy mixture as an organic solvent. Previous studies in the literature have indicated that the inclusion of acetone does not trigger chemical alterations in the epoxy chain [13,22]. The mixture was then homogenized at 8000 rpm until the acetone had completely evaporated from the matrix. Once this was confirmed by weighing, a hardener was added to the mixture and homogenized again for 15 minutes at room temperature. After the mixtures were prepared, they were poured into silicone molds, and air bubbles in the mixtures were removed using a heat gun. The mixture was left in the mold to cure at room temperature for 24 hours. Lastly, the nanocomposite samples were post-cured for 4 hours at 60°C. The summary of production steps is illustrated in Figure 1.

2.3 Tensile Test Procedure

Tensile tests were performed at room temperature using the SHIMADZU AG-X series universal testing machine, as shown in Figure 2, following the guidelines of ASTM D638. Dog bone-shaped test samples were used in the test, and the crosshead speed was set to 2 mm/min. the findings, at least five samples were examined, and average values were compared with one another.

Material	Specification	Values	
MWCNT	Purity $(\%)$ Density (g/cm^3) Surface Area (m^2/g) Inner diameter (nm) Outer diameter (nm)	>96 2.4 >210 $5-10$ $8 - 18$	
	Length (μm)	$10 - 35$	
NS	Purity $(\%)$ Density (g/cm^3) Surface Area (m^2/g) Average particle size (nm)	99.5 0.05 300 15-35	
MGS L160	Density (g/cm^3)	$1.13 - 1.17$	
Resin	Viscosity (mPa.s)	700 - 900	
MGS H160	Density (g/cm^3)	$0.96 - 1.00$	
Hardener	Viscosity $(mPa.s)$	$10 - 50$	

Table 1. The physical properties of MWCNT, NS particles and epoxy resin system [26].

Figure 1. Nanocomposite production steps.

Table 2. Composition of nanocomposite samples

Nanoparticle Type	Designation	Filler Ratio (wt%)
	Pure	0
	C1	0.1
MWCNT	C ₂	0.3
	C ₃	0.5
	C ₄	0.7
	S1	0.5
NS	S ₂	1.0
	S ₃	1.5
	H1	0.5:0.5
MWCNT : NS	H ₂	0.33:0.66
	H3	0.25:0.75

Figure 2. (a) Tensile test set-up, (b) boundary conditions, (c) test samples -C3, C4, S2, S3, H1 and Pure-, (d) fracture test samples.

3. Results and Discussion

Tensile testing at room temperature, in accordance with ASTM D638, was conducted on pure epoxy, as well as single and hybrid nanoparticle-filled epoxy composites. Figure 3 depicts the engineering stress-strain curves of both nanoparticle-filled and unfilled samples. The samples exhibit a linear trend up to the ultimate tensile strength before experiencing sudden drops, indicating catastrophic failure in the composite samples.

The average results of the tensile tests are presented in Table 3 and Figures 4a- b. Notably, an enhancement in tensile strength was observed across all samples in comparison to the pure epoxy sample. Specifically, when focusing on single nanoparticle-filled epoxy composites, the maximum improvements in tensile strength were found as 28.35% and 21.25% for the C3 and S2 samples containing 0.5 wt.% MWCN and 1 wt.% NS, respectively, as compared to the pure epoxy sample. In the context of other nanocomposite types, increases in tensile strength values were observed, measuring 3.77%, 11%, and 25.89% for C1, C2, and C4 samples, respectively, incorporating 0.1, 0.3, and 0.7 wt.% MWCNT. Slight enhancements were computed at 4.37% and 3.7% for S1 and S3 samples, respectively, including 0.5 and 1.5 wt.% NS. This may be due to the morphology of MWCNT in tube form and NS in spherical form. MWCNT forms an interconnected structure with a complex tube shape, resulting in greater load transfer between nanotubes and epoxy [28,29]. The noteworthy aspect of the data presented in Table 3 and Figure 4a is the consistent and substantial improvement observed in the tensile strength of all hybrid nanoparticle-filled samples.

Figure 3. Engineering stress-strain curves of nanocomposites

		Standard		Standard	
Sample	Tensile Strength	Deviation in	Tensile Strain	Deviation	Tensile Modulus
	(MPa)	Tensile Stress	(%)	in Tensile Strain	(GPa)
		(MPa)		(%)	
P	31.32	0.44	2.14	0.32	1.53
C ₁	32.50	1.95	3.90	1.37	1.40
C ₂	34.77	1.78	4.02	1.19	1.14
C ₃	40.20	0.75	3.46	0.24	1.16
C ₄	39.43	2.57	2.24	0.12	2.61
S1	32.69	2.03	1.45	0.20	2.70
S ₂	37.98	5.45	4.19	0.50	4.37
S ₃	32.50	1.99	2.80	1.07	1.30
H1	45.18	1.92	4.01	0.60	2.13
H2	42.57	2.25	5.20	1.27	2.27
H ₃	41.53	3.19	3.78	0.39	2.25

Table 3. Improvements in tensile behavior of the nanocomposites

This observation strongly implies a positive hybrid effect, where the combination of different nanoparticle types leads to enhanced tensile properties compared to single nanoparticle-filled samples or the pure epoxy sample. As can be seen from the test results presented in Table 3, an excellent improvement in tensile strength was recorded as 44.26% in H1 hybrid sample with 1:1 nanoparticle ratio. Also, in H2 and H3 samples, these improvements were found as 35.91% and 32.60, respectively. The enhancement in tensile strength attributed to the introduction of nanofillers stems from the expanded interfacial region, thereby mitigating crack energy dissipation.

Also, the improved tensile strength observed in composites filled with either single or hybrid MWCNT and NS is a result of the better wettability of these nanofillers within the epoxy matrix. This improved wettability is a result of the effective and uniform distribution, and improved bonding strength established between the epoxy resin and the nanofillers [30,31]. This phenomenon significantly contributes to the overall enhancement in mechanical properties observed in the composite material. On the other hand, up to a certain nano-particle content, the tensile strength values begin to drop. The reason for this can be

related to as the nanoparticle filler ratio is increased, the van der Waals attractive forces between the particles increase, thus resulting in very high surface energies, and the particles tend to aggregate to reduce this energy [26,32]. Thus, these agglomerations cause nonuniform local stress concentrations.

Strain plays an important role as a fundamental parameter, offering valuable insights into the ductility of materials. As presented in Table 3 and Figure 4b, tensile strain values were improved except for the S1 sample compared to the pure epoxy sample. Specifically, the addition of MWCNT and NS as single nanofillers resulted in substantial improvements of about 87.85% and 46.26% for the C2 and S2 samples containing 0.5 wt.% MWCNT and 1 wt.% NS, respectively, as compared to the pure epoxy sample. This can be attributed to the inhomogeneous dispersion and agglomeration of the nanofillers and their embedded into the epoxy. Examining the tensile strain values of hybrid nanofillers, H1, H2, and H3 showed impressive improvements of approximately 87%, 143%, and 76%, respectively. This underscores that the combined application of MWCNT and NS significantly enhances the ductility of the epoxy matrix, mitigating embrittlement.

3.1 Fracture Characterization

The fracture surfaces of the samples were examined to better understand the behavior of the samples subjected to tensile loads and to examine the fracture modes. The surfaces of the samples were imaged with high-resolution microscopes (LEICA EZ4 HD and LEICA DM750). Figures 5 and 6 represent the surface images of the test samples. In the presented figures, the left column images were taken at 35x magnification, while those in the right column were taken at 63x magnification.

When the surfaces of the unmodified samples are examined, the fracture surface is smooth and glassy, and no significant plastic deformation occurred during fracture. In Figure 5, the surfaces of nanoparticle-filled samples are rougher and more complex than pure samples. The greater the roughness, the more complex the fracture mechanisms and the more energy absorption during crack propagation, leading to an enhancement of the matrix's mechanical characteristics [26,29]. When the fracture surfaces are compared, the surface of MWCNT samples is rougher and thus the strength values of these samples are higher. This is due to the entangled tube form of MWCNT particles, which increases the load transfer by making more interfacial bonds with the matrix.

However, as the nanoparticle content exceeded a specific weight percentage (0.5 wt.% for MWCNT and 1 wt.% for NS), stress concentrations, in the form of starshaped fracture patterns, indicating fracture origin sites on the surfaces, became evident. This ultimately led to the initiation of cracks. In the literature, these stress concentrations are often attributed to nanoparticle agglomeration [33,34].

As seen Figure 6 when the surfaces of the hybrid nanoparticle samples were examined, the surface of the H1 samples was rougher, and this roughness caused more energy absorption and showed better mechanical improvement than the other hybrid samples. This may be due to the morphological structure of MWCNT as mentioned above. It has been noted that large tensile stresses arise in matrix structures reinforced by high aspect ratio additions, like nanotubes because the filler merges with the matrix. High energy absorption and a rise in fracture energy were caused by the frictional forces attempting to keep the nanotubes from separating from the matrix and the shear force created by the separation zone expanding along the nanotube length [35].

As a result, the rougher the fracture surfaces, the more complex the fracture modes will be, resulting in higher energy absorption and higher energy fracture, increasing the mechanical properties.

4. Conclusions

This study investigated the influence of incorporating

nanoparticles into the epoxy matrix on the load-carrying capacity of the material. The study assesses different weight contents of MWCNT and NS particles. Results demonstrate that the nanofillers incorporation into the matrix significantly impacts the load-carrying capacity. The results obtained are as follows.

Figure 4. Tensile test results, (a) tensile strength, (b) tensile strain deviation for nanocomposites

Figure 5. High-resolution microscope images of single nanoparticle filled samples

Figure 6. High-resolution microscope images of hybrid nanoparticle filled samples

- Tensile strength demonstrated improvements in all nanocomposites containing single nanoparticles. Among these nanocomposites, the maximum tensile strengths were 40.20 MPa and 37.98 MPa in C3 and S2 samples, respectively, and an improvement of 28.35% and 21.25% was achieved.
- Significant enhancements were observed in the tensile strength values of all composite samples that contained hybrid nanofillers. The H1 sample, which had a 1:1 hybrid ratio, showed the highest improvement in tensile strength value, with a 44.26% increase compared to the pure epoxy sample. The H2 and H3 samples also showed improvements in strength values, with increases of 35.91% and 32.60%, respectively.
- An improvement in the ductility of the material was observed in almost all nanocomposite samples, as indicated by the increase in tensile strain values. Maximum improvement in strain value was recorded as about 143% in H2 hybrid sample.
- From the test results, we can conclude that by opting for more cost-effective materials, we can produce composite materials with enhanced mechanical properties. Although the unit cost of NS is much less than that of MWCNT, it is possible to achieve superior mechanical properties in a composite material by using equal or higher amounts of NS in the form of a hybrid nanofiller.
- Further research could also be conducted to determine the effectiveness of using a hybrid nanofiller combined with a nanoparticle having tube shape morphology (CNT) and different types of morphology such as plate-like nanoparticles graphene nanoplatelets (GNP).

Declaration

The author declared no potential conflicts of interest with respect to the research, authorship, and/or publication of this article. The author also declared that this article is original and was prepared in accordance with international publication and research ethics, and ethical committee permission or any special permission is not required.

Author Contributions

A. Kayalar and N.F. Dogan have contributed to defining and managing the conceptual and design processes of the study, data collection, data analysis and interpretation, drafting the article, critically examining the conceptual content, and contributing to the final approval and full responsibility stages.

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Research Article

Experimental investigation on energy absorption capability of 3D-printed lattice structures: Effect of strut orientation

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1. Introduction

The use of cellular structures in engineering applications has become increasingly popular due to their unique properties. However, the irregular and unpredictable shapes of such structures present a challenge in determining their mechanical properties. To overcome this obstacle, lattice structures made up of interconnected 3D cells have emerged as a viable alternative. These lattice structures offer consistent and adjustable mechanical properties, making them the preferred choice for constructing lightweight cellular structures in different sectors, such as aerospace, aviation and automotive. Moreover, the interconnected cells in these lattice structures can be repeated, allowing for easier fabrication and assembly. This has made lattice structures a go-to solution for engineering applications that require the use of cellular structures with predictable and adjustable mechanical properties while also being lightweight and easy to manufacture [1–10]. Lattice structures are threedimensional structures composed of interconnected struts that form a repeating unit cell. These structures possess mechanical properties that are influenced by various

factors, such as load type, material used, unit cell dimensions, topology, cell edge radius-to-length ratio, and density. The density of the structure influences its stiffness, strength, and energy absorption capabilities. The cell edge radius-to-length ratio plays a vital role in determining the mechanical properties of the structure. A higher radius-to-length ratio results in a structure that is more flexible and has a lower strength. On the other hand, a lower radius-to-length ratio results in a more rigid structure with a higher strength [11]. Lattice structures are ideal for applications that require high stiffness, energy absorption, and strength, such as aerospace, automotive, and biomedical industries. These structures provide a high strength-to-weight ratio, allowing them to withstand high loads while being lightweight. The introduction of additive manufacturing (AM) has revolutionized the production of lattice structures. This technique enables precise control of the geometry of cellular structures with intricate architecture at all scales, ranging from nano to macro. AM can be used to fabricate virtually any open-cell lattice architecture, including complex shapes and patterns. The design possibilities are endless, and the structures can be

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customized to meet specific requirements. Moreover, AM enables the production of structures with reduced material waste and increased productivity, making it a sustainable and cost-effective manufacturing process [12–18].

Mechanical analysis is a common approach to studying the mechanical properties of lattice structures. Experimental studies, as documented in sources [19–23], have been found to produce highly precise results. However, when dealing with more complex cellular geometries, numerical examination, as described in sources [24–28], is usually preferred due to the challenges associated with manufacturing such structures. Sun et al. [29] carried out an experimental and numerical investigation to examine the impact of relative density on the absorbed energy in lattice structures created using ABS material through the AM technique. The outcome of this research highlights a distinct association between the energy absorbed and relative density. The results indicate that an enhancement in relative density is directly proportional to an increase in energy absorption. A recent study has revealed a noteworthy finding related to the relationship between specific energy absorption and relative density. The research indicates that as the relative density increases, there is a steady rise in the specific energy absorption, implying a reliable and consistent bond between the two variables. This discovery can have significant implications for industries that rely on highenergy absorption materials, such as the automotive and aerospace sectors [30,31].

In their study, Wang et al. [32] explored the behavior of polyurethane 3D-printed lattice structures subjected to compression loads, specifically investigating the influence of lattice wall thickness on deformation force. Their findings offer valuable insights into optimizing lattice structures to enhance their capacity to withstand compression, thereby enhancing their applicability in realworld scenarios. The researchers observed a notable distinction in the behavior of lattice structures with variable wall thickness compared to those with fixed wall thickness. Specifically, they noted that as plastic deformation initiated, the force value increased for structures with variable wall thickness, while structures with fixed wall thickness maintained a constant force value throughout the process. This observation suggests that the material's response to deformation is influenced by the variability in wall thickness within the lattice structure. These findings carry significant implications for the design and manufacturing of products requiring specific levels of force resistance and structural integrity. By understanding how variations in lattice wall thickness affect deformation force, designers can tailor lattice structures to meet the desired performance requirements for various applications. Such insights pave the way for developing

more robust and efficient lattice-based components and products across various industries.

The research study was focused on examining the orientation of struts in cube, octet, and body-centeredcubic (BCC) lattice structures with a relative density of 30%. The fused deposition modeling (FDM) method was used to manufacture the samples, and nylon filament with chopped carbon fiber was utilized as the printing material. The lattice structures had three units in each direction of x and y, while in the z-direction, they had either one or three units. The test speed was set as quasi-static, which means the test was conducted at a very low speed (2 mm/min), and the test results were examined to determine the energy absorption values of the samples. The absorbed energy values were evaluated by weighing the samples to obtain the specific energy absorption values. This research provides valuable insights into the strut orientation in lattice structures and highlights the potential of Additive Manufacturing (AM) in producing complex and efficient cellular structures. The findings of this study can be used to optimize the design of structures with improved energy absorption capabilities, which can be utilized in various applications such as automotive components, aerospace structures, and sports equipment.

2. Materials and Method

The current study involved the fabrication of various lattice samples using a nylon filament infused with chopped carbon fiber. The mechanical properties of this composite filament were rigorously quantified, yielding a Young's modulus of 528 MPa and a yield strength of 22.6 MPa, as reported in reference [7]. The lattice structures were manufactured through the Fused Deposition Modeling (FDM) technique. This method was chosen for its precision and reliability in producing complex geometries. All lattice types were designed to maintain a consistent relative density of 30%, ensuring structural integrity and performance uniformity across the different samples.

Table 1 provides a comprehensive list of the printing parameters used during the 3D printing process. These parameters include but are not limited to, nozzle temperature, bed temperature, printing speed, layer height, and infill pattern. Such detailed documentation of the printing settings is crucial for reproducibility and for understanding the influence of these parameters on the mechanical properties of the final printed structures. Each unit cell of the lattice was meticulously designed with dimensions of 10 mm along the x, y, and z axes. For experimental purposes, the lattice structures were printed in arrays consisting of three units in the x-direction, three units in the y-direction, and either one or three units in the z-direction. This specific configuration was selected to explore the effects of varying structural height on the

mechanical performance of the lattice. This detailed dimensional and configurational information is essential as it provides insights into the design considerations and scalability of the lattice structures. Understanding the geometric parameters is critical for correlating the observed mechanical properties with the structural design, thereby enabling the optimization of lattice structures for various engineering applications.

The selection of three lattice types — cube, octet, and body-centered cubic (BCC) — was based on their distinctive strut orientations. Figure 1 illustrates the 3Dprinted unit lattices for each of these structures, providing a visual representation of their geometrical configurations. All lattice types were fabricated with struts having circular cross-sections, ensuring consistency in cross-sectional geometry. However, the orientation of the struts varies significantly among the different lattice types, which contributes to their unique mechanical properties and structural behavior. A detailed examination of Figure 1 reveals that the cube lattice is characterized by exclusively vertical (highlighted by a blue dashed box) and horizontal struts (highlighted by a red dashed box). This simple arrangement results in a straightforward, orthogonal lattice structure. In contrast, the octet lattice features a more complex arrangement with both horizontal struts and struts angled at 45 degrees (highlighted by a green dashed box). This configuration enhances the lattice's ability to distribute loads more evenly and efficiently. The BCC lattice, on the other hand, incorporates a combination of all the strut orientations found in the cube and octet lattices. This includes vertical, horizontal, and 45-degree angled struts, creating a more intricate and potentially more resilient structure. The diversity in strut orientation within the BCC lattice contributes to its superior mechanical properties, as it can better resist different types of mechanical stresses. The varied strut orientations among these lattice types are critical to understanding their mechanical behavior and performance. This selection allows for a comparative analysis of how different geometrical configurations impact the overall structural integrity and mechanical properties of the 3D-printed lattices.

Figure 1. The unit lattice structures and the strut orientations of a) cube, b) octet, c) BCC

2.1 Crashworthiness Parameters

In order to discover the most efficient lattice type in terms of energy absorption, it was necessary to analyze the data collected from the experiments using specific crashworthiness parameters. To achieve this, the densification displacement needs to be specified to calculate the energy absorption values of the lattice samples. For this purpose, the crushing displacement efficiency parameter (η) defined in the literature [33,34] is used. The highest value of the calculated η represents the densification starting displacement value. The related equation of the crushing displacement efficiency parameter is given in Equation 1:

$$
\eta(\varepsilon) = \frac{1}{F(\varepsilon)} \int_0^{\varepsilon} F(\varepsilon) d\varepsilon \tag{1}
$$

where the η , F and ε refer to the displacement efficiency, force and displacement values, respectively. The EA (energy absorption) value of the samples is determined through Equation 2, as shown below. It is worth noting that the displacement value considered as the final displacement was the one obtained at the highest value of the crushing displacement efficiency parameter.

$$
EA = \int_0^{\varepsilon_D} F(\varepsilon) d\varepsilon \tag{2}
$$

The computation of Specific Energy Absorption (SEA) is a crucial process that involves determining the amount of energy absorbed per unit mass. To derive the SEA value accurately, the following equation can be utilized, which takes into account the mass of the lattice structure. It is pertinent to note that, similar to EA, the calculation of SEA also extends up to the densification displacement. The m in Equation 3 refers to the total weight of the lattice sample and the ρ refers to the density of the bulk material.

$$
SEA = \frac{EA}{m} = \frac{\int_0^{\varepsilon} F(\varepsilon) d\varepsilon}{\rho' \rho}
$$
 (3)

3. Results and Discussion

It is worth noting that all experiments were meticulously repeated thrice to ensure accuracy and consistency. A single curve with the average value was presented to minimize visual clutter in the graphs. The study began by examining the test results of single-layer lattice structures and then proceeded to investigate the test results of multi-layer lattice structures.

3.1. Single-layer lattice

The results of axial compression tests of various lattice structures, including cube, octet and BCC, are presented in Figure 5. The study investigates the force-displacement and compression efficiency-displacement curves of the samples and highlights their characteristics in the graphs. The force curve of all three lattice structures shows a similar behavior with a rapid increase at the beginning of the experiments, followed by a plateau region, and then a further increase. The plateau values are found to be approximately 6 kN, 5 kN and 2.5 kN for cube, octet and BCC lattice structures, respectively. The cube structure exhibits the highest plateau value, which can be attributed to the largest strut diameter in this structure. It is noteworthy that the octet structure, despite having the smallest strut diameter, shows a higher plateau value than the BCC structure.

The compressive force efficiencies were calculated to determine the energy values absorbed by the samples. The efficiency curves calculated with the expression given in Equation 1 highlight the displacement value corresponding to the highest value reached by the compressive force efficiency during the entire test period, which gives the value corresponding to the densification of the sample. Densification initial displacement values were calculated to be approximately 4.5 mm in cube and BCC lattice structures, and 5 mm in the octet structure. The energy values absorbed by the samples were calculated according to these displacement values (as the area under the force curves up to the initial values of densification).

The energy absorption values for the cube, octet, and body-centered cubic (BCC) lattice structures were measured to be 22.7 J, 24.8 J, and 12.8 J, respectively. This indicates that the cube and octet structures absorb 77% and 94% more energy compared to the BCC structure. Additionally, the octet structure absorbs 9% more energy than the cube structure. These observed differences in energy absorption are particularly noteworthy, considering that the cube lattice structure has the largest strut diameter among the three. The vertical orientation of the struts in the cube structure is likely the primary reason for the observed decrease in the force curve beyond a displacement of 3 mm. This behavior can be attributed to buckling occurring in the vertical struts of the cube lattice. If this buckling-induced decrease in force did not occur, the energy absorption value of the cube structure would likely be comparable to that of the octet structure. Therefore, the orientation and configuration of the struts play a significant role in the mechanical performance and energy absorption efficiency of lattice structures. These findings provide important insights into the relationship between strut geometry and mechanical behavior. The superior performance of the octet structure, despite its smaller strut diameter, suggests that optimizing strut orientation and preventing buckling can lead to significant improvements in energy absorption capabilities. This information is valuable for the further development and refinement of lattice structures for engineering applications that demand high energy absorption and mechanical stability.

Figure 2. The force-displacement and efficiency-displacement curves of single-layer a) cube, b) octet, c) BCC lattice structures

3.2. Three-layer Lattice

The force-displacement curves of three-layer lattice structures under axial load are presented in Figure 3, along with the compressive force efficiencies shown with dashed lines of the same color as the force curves. Although the force values are similar for octet and BCC lattice structures, they have lower values. This happens because the struts that are forced between two rigid plates in single-layer lattices are forced to be crushed between a rigid plate and the lattice below in three-layer structures. Therefore, the decrease in

plateau values is due to this reason. The sudden decrease in the force curve in the cube structure is another important issue. Due to the short vertical struts of the single-layer cube lattice, it was observed that a deformation in the form of barreling occurred in these structures instead of buckling. The struts of the three-layer cube structure are subjected to buckling between rigid plates under compressive force. It can be said that this buckling behavior is directly related to the decrease in the force value of the truss structure.

When examining the compressive force efficiency values of the samples, it is observed that the displacement values corresponding to densification are calculated as 11 mm, 15 mm, and 13 mm for the cube, octet, and BCC structures, respectively. The octet structure exhibits the highest displacement value, consistent with its behavior in singlelayer structures. Although the single-layer samples show similar densification values, the lower displacement value of the cube lattice structure in the three-layer configuration can be attributed to the buckling behavior previously discussed. By calculating the absorbed energies of the cube, octet, and BCC lattice structures using the determined densification displacement values, it was found that the cube structure absorbed 26 J, the octet structure absorbed 34.7 J, and the BCC structure absorbed 27.3 J. In single-layer lattice structures, the cube structure absorbs 77% more energy than the BCC structure, though it remains at overall lower energy levels. Additionally, the octet structure absorbs 94% more energy than the BCC structure in a single layer, but this advantage decreases to 27% in the three-layer configuration. These findings highlight the critical influence of structural geometry and configuration on the energy absorption capabilities of lattice structures. The octet structure's superior performance in both single and three-layer configurations underscores its potential for applications requiring high energy absorption. In contrast, the cube structure's reduced efficiency in multi-layer setups due to buckling behavior suggests the need for design modifications to enhance its stability. These results provide valuable insights for the further research and development of optimized lattice structures for various engineering applications.

3.3. Crashworthiness

Table 2 displays the results of the axial crushing tests that were carried out on single and three-layer lattice structures. The tests were analyzed based on the energy absorption efficiency parameters explained in Section 2.1. It is seen that the samples had similar weights regardless of their geometry since the relative density was chosen at 30%. Choosing the same relative density for all lattices resulted in different strut diameters. The diameters of the lattices were measured as 2.05 mm, 0.87 mm, and 1.13 mm, respectively. The changes in strut diameters are mainly because of the varying strut numbers. Since the cube lattice has only four struts in a single unit, it has the highest strut diameter and vice versa.

Figure 3. The force-displacement and efficiency-displacement curves of three-layer a) cube, b) octet, c) BCC lattice structures

The sample weights of the single-layer and three-layer structures were determined to be 3.8 grams and 9.8 grams, respectively. Analysis revealed that the octet structure exhibited the highest energy absorption values among the lattice types for both single-layer and three-layer configurations. However, the energy absorption values for the three-layer samples were lower than anticipated. This discrepancy is primarily attributed to the structural configuration of the three-layer lattices.

Lattice type		Weight (g)	Densif. displ. (mm)	Energy absorbed (J)	SEA (J/g)
	Cube	3.8	4.5	22.70	5.97
3x3x1	Octet	3.8	5.0	24.80	6.53
	BCC	3.8	4.4	12.80	3.37
3x3x3	Cube	9.8	11	26.00	2.65
	Octet	9.8	15	34.70	3.54
	BCC	9.8	13	27.30	2.79

Table 2. Crashworthiness parameters of single-layer and three-layer lattices

In these configurations, the struts were in contact with the rigid plate on only one side. In contrast, the struts in the single-layer lattices were compressed between two rigid plates, leading to more effective energy absorption. Moreover, it was observed that the cube lattice displayed buckling behavior in the three-layer configuration, a phenomenon that is not present in the single-layer structure. This indicates that the additional layers in the cube lattice introduced instability, resulting in buckling under compressive loads. These findings highlight the impact of layer configuration and boundary conditions on the mechanical performance of lattice structures. The difference in energy absorption and the occurrence of buckling provides valuable insights into the structural behavior of multi-layered lattices under load, informing future design and optimization of such structures for enhanced mechanical performance.

4. Conclusion

The effect of the strut orientation on the energy absorption capability of the lattice structures was experimentally investigated. Three different lattices of cube, octet, and BCC were chosen since the first two of them have vertical and angled struts, respectively, while the BCC has a combination of vertical and angled struts. The samples were manufactured using the FDM printing method. A nylon-based filament, that contains chopped carbon fiber, was chosen to print the samples. The lattice samples were printed at single and three-layer heights to observe the effect of the strut length on the deformation behavior of the samples.

The results of this study have shown that the orientation of the struts in 3D-printed lattice structures plays a critical role in their ability to absorb energy. It was seen that the 45˚ angled strut lattice resulted in the best in terms of the absorbed energy and the specific energy absorption parameters. The study found that vertical struts are prone to buckling when their length is increased, resulting in lower energy absorption. However, lattice structures that incorporate both vertical and angled struts, such as the three-layer BCC lattice, could demonstrate significantly higher energy absorption rates. Interestingly, the cube lattice with only vertical struts showed insufficient energy

absorption due to buckling issues, as observed in both single and three-layer configurations. Therefore, it can be concluded that adding angled struts is a crucial factor in enhancing the energy absorption and crashworthiness of lattice structures. Also, the layer numbers of the lattices significantly affect the deformation behavior and energy absorption capability of the samples. Especially for the vertical struts, the increase in layer number increases the slenderness of the vertical struts, which causes buckling. These findings should be considered when designing 3Dprinted lattice structures for applications in aerospace, automotive, and biomedical engineering.

Declaration

The author declared no potential conflicts of interest with respect to the research, authorship, and/or publication of this article. The author also declared that this article is original and was prepared in accordance with international publication and research ethics, and ethical committee permission or any special permission is not required.

Author Contributions

Muhammet Muaz Yalçın developed the methodology, performed the experiments, and wrote the manuscript.

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Research Article

Influences of post-weld artificial aging on microstructural and tensile properties of friction stir-welded Al-Zn-Mg-Si-Cu aluminum alloy joints

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1. Introduction

Aluminum and its alloys, with their low specific gravity, high strength and improved corrosion resistance, are now widely used in areas requiring high mobility, such as automotive, aerospace and defense [1–7]. Similar and dissimilar welded aluminum joints and its alloys are needed in these areas. However, the weldability of the most widely used age-hardenable Al-Cu, Al-Cu-Mg and Al-Zn-Mg-Cu alloys is poor with fusion welding methods such as gas tungsten arc welding [8–11]. This is because during the re-solidification of the molten zone after welding, defects such as microstructural changes and solidification cracking occur, which negatively affect the mechanical properties [12]. In addition, lower distortions and residual stresses occur with lower heat input [13]. For this purpose, solid-state welding methods are used for agehardenable 2XXX and 7XXX series aluminum alloys. Friction stir welding (FSW) is one of the solid-state welding methods used for welding aluminum and its alloys [10]. The Weld Institute of Cambridge in the UK developed and patented this technique in 1991 [12,14]. In the FSW technique, two plates to be joined are joined by dynamically stirring the two parts together using a tool that rotates on the joint surface [15]. In addition to preventing solidification cracking with this method without melting,

significant grain refinement occurs due to dynamic recrystallization in the nugget zone [11,12].

Artificial aging increases the strength of aluminum alloys and welded joints. If heat treatment is applied before welding, the precipitates in fusion welds may dissolve again due to the high temperature. In FSW, although the melting temperature is not reached, microstructural distortions may occur due to high temperature and high plastic deformation in the weld zone [16]. In addition, many studies have reported that intermetallic precipitates are re-dissolved by the FSW process after aging [17–19]. For this reason, the post-welding heat treatment (PWHT) of FSWed aluminum alloys is a strength enhancement method. In the literature, there are several studies investigating the effect of post-welding heat treatment on the mechanical properties of aluminum alloys such as AA2014 [20], AA2024 [21], AA 6061 [22,23], AA7039 [24], AA7075 [13] etc., joined by friction stir welding. Most of the studies in the literature have focused on postweld heat treatment [25–28]. This is because friction stir welding of peak-aged aluminum alloy reduces tool life due to higher resistance to deformation [29].

Yadav et al. subjected AA2024 aluminum alloy to an artificial aging process after FSW. It was reported that the yield and tensile strengths, as well as the ductility of the

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specimens, increased with PWHT [21]. When Feng et al. applied PWHT to FSWed AA2219-O, it was reported that the strength of the welded joint increased, although grain coarsening occurred in the stir zone (SZ) [30]. Sabari et al. performed tests with untreated AA2519, welded, artificially aged, solution treated and artificially aged FSWed AA2519 specimens and obtained the best mechanical properties in the solution-treated and artificially aged FSWed AA2519 specimen [31]. Pabandi et al. investigated the effects of precipitation hardening solution treatment and artificial aging heat treatment on the mechanical and microstructural properties of FSWed AA6061-T6-AA2024-T6. They found that PWHT increased the weld strength [32]. Sharma et al. reported that the mechanical properties of FSWed Al-Zn-Mg aluminum alloy specimens increased with natural aging, while only the solution-treated specimens showed a decrease in mechanical properties [24]. As seen in many studies in the literature, the increased tensile strength will be achieved by obtaining a precipitated and fine-grained microstructure with a homogeneous distribution in the weld zone.[33]. Most of the studies in the literature have focused on post-weld heat treatment. This is because friction stir welding of peak-aged aluminum alloy reduces tool life due to higher resistance to deformation.

This study selected the Al-Zn-Mg-Si-Cu alloy, widely used in the automotive, aerospace and defence industries but not recommended for fusion welding. The effects of annealing and artificial aging on this alloy's microstructural and mechanical properties after friction stir welding were investigated.

2. Materials and Methods

In this study, wrought aluminum alloy with dimensions of 500x80x5 mm sheets were used. The chemical composition of the aluminum alloy used is presented in Table 1. The samples were welded using the friction stir welding (FSW) machine (Beijing FSW Technology Co.) with an FSW-LM-BM 16-2D Gantry Model. The FSW process was performed using a rotational speed and a traverse speed of 1250 rpm and 40 mm·min-1 , respectively. The images of the 4 mm long pin and shoulder made of H13 tool steel used are given in Figure 1. After the FSW process, 1 mm chips were removed from the surfaces of the joints by milling. To determine the mechanical properties of the heat treatment after FSW, some welded specimens (T0) were annealed at 415 °C for 3 hours. The remaining specimens were subjected to artificial aging heat treatment at 450 °C for 2 hours after solution and quench cooling. These specimens were coded as aging temperatureduration (190 °C-2 h, 190 °C-4 h, 205 °C-2 h, 205 °C-4 h).

The samples to be used for microstructural and mechanical investigations were removed from the welded plate by electrical discharge machining. The samples for microstructural investigations were sanded with 100-1200 grit SiC abrasive papers in a Metkon Forcipol 1V polishing machine and then polished using 3 and 1μ diamond solution, respectively. After polishing, the sample surfaces were etched using Keller's Reagent (1.0 mL HF, 1.5 mL HCl, 2.5 mL HNO₃ and 95 mL water). After the etching process, the microstructures of the samples were examined with a Nicon Eclipse LV150N optical microscope (OM) and Clemex image analysis system. Grain sizes were measured at 1000x magnification with 10 measurements from different regions and averaged. Microhardness measurements were conducted with a load and dwell time of 100 gf and 10 s, respectively, 1 mm below the surface. The measurements were conducted using the Future-Tech FM-700 Vickers micro-hardness tester (Future-Tech Corp., JAPAN). The tensile tests were carried out according to ASTM-E8-04 standard using the Shimadzu Autograph AG-IS 100kN universal testing machine. The tests were repeated five times for each specimen at room temperature with a 1 mm·sec⁻¹ strain rate. Fracture surface examinations were performed using a ZEISS GeminiSEM 500 scanning electron microscope (SEM) with a secondary electron (SE) detector from ZEISS, Oberkochen, Germany.

Table 1. Chemical composition of the aluminum alloy

$(wt.\%)$				
Elements	Amount (wt. %)			
Al	93.700			
Zn	2.335			
Si	0.912			
Mg	1.022			
Cu	0.912			
Fe	0.485			
Mn	0.196			
Cr	0.129			
Other	0.309			

Figure 1. Photograph of FSW tool with threaded conical pin

3. Results and Discussion

Figure 2 shows the cross-sectional macrostructure of the samples annealed, aged at 190 °C for 2 and 4 hours and at 205 °C for 2 and 4 hours, respectively. The cross-sectional image clearly shows the different regions in the welded specimens. These regions are thermally and mechanically affected base metal (BM) at the edges, friction stir zone (FSZ) in the middle, shoulder deformation zone (SDZ) at the top and thermo-mechanically affected zone (TMAZ) between BM and FSZ. In addition, the retreating side (RS) and the advancing side (AS) were seen in the SDZ of all joints. It was found that the TMAZ was wider in RS in all samples. In addition, the heat-affected zone (HAZ) was also relatively wider in RS due to the higher temperature. Voids may form in the AS, especially when material transfer, plastic flow and temperature are insufficient [34,35]. However, as shown in Figure 2, no void, tunnel or cracking-like weld defects were detected in any region of the specimens. Onion ring-like flow bands, also detected in different studies [35–37], were observed in the FSZ region. While some of the studies explained the "onion ring" shaped band formations with the temperature differences occurring during the FSW process [38,39], some studies have found that there is a difference in precipitate and hard particle density between these bright and dark bands [40,41]. In addition, it has been reported that band formation is evident in artificially aged specimens before welding, while it loses its visibility in annealed specimens [35]. In this study, the band formation was not visible in the T0 sample due to the dissolution of precipitates during annealing after the friction stir welding (FSW) process (Fig. 2a).

In Figure 3, micrographs of the specimens taken from the FSW region are given respectively. Different microstructures in different regions of the welded joint can be detected in these micrographs. In addition to FSZ and TMAZ, HAZ can be seen at the bottom right of the micrographs. The TMAZ and HAZ are relatively narrow in these pictures taken from the AS area. In all samples, apparent grain growth was detected in the HAZ, while in the TMAZ, generally small and oriented grains were observed. Smaller but almost coaxial grains were observed in the FSZ. The average size of the FSZ grains was 3-5 µm. Similarly, Sajadifar et al. reported the average grain size in the nugget zone of AA 7075 alloy specimens aged after welding as $2 \mu m$ [42]. In addition, changes occurred in the microstructure of FSZ with artificial aging. Some increase in the size of α aluminum grains in the FSZ of artificially aged samples was detected.

The hardness changes of FSWed specimens at the weld line after heat treatment are given in Figure 4. Post-weld heat treatment improved the microhardness of all areas.

Figure 2. Cross-sectional macrostructure images of the samples: a) T0, b) 190 °C-2 h, c) 190 °C-4 h, d) 205 °C-2 h and e) 205 $^{\circ}$ C-4 h.

Figure 3. Cross-sectional micrographs of the samples: a) T0, b) 190 °C-2 h, c) 190 °C-4 h, d) 205 °C-2 h and e) 205 °C-4 h.

Hardness increases, especially in the HAZ, where dramatic hardness increases were observed after the welding process, were also seen in previous studies where heat treatment was performed [43]. It can be seen that the highest hardness values in all specimens are in the FSZ, and the hardness decreases from the weld center to the BM. Many studies have reported that in aluminum alloys, the combination of high plastic deformation and frictional heating effect during FSW, always in the FSZ, results in a recrystallized microstructure [1,44–48]. As is well known, grain size affects the strength of metals. The relationship between grain size (d) and yield stress (σ_v) is described by the Hall-Petch equation [48]:

$$
\sigma_y = \sigma_0 + k_y d^{-1/2} \tag{1}
$$

Here, σ_0 is the friction stress, and k_y is the material constant. As can be seen from this equation, the yield strength of metallic material decreases as the grain size increases. This equation takes the following form using the $H_v \cong 3\sigma_v$ equation between the material hardness H_v and $\sigma_{\rm v}$:

$$
H_v = H_0 + k_H d^{-1/2}
$$
 (2)

As can be seen, a decrease in grain size increases the hardness of the metallic material. For this reason, a hardness decrease from FSZ to BM was observed in all samples. Microstructural investigations revealed a slight increase in grain size in the FSZ of the artificially aged specimens compared to the annealed specimen (Figure 3). The effective mechanism for the hardness increases along the weld line in all aged specimens is precipitation hardening. The hard intermetallic precipitates formed due to artificial aging provide Orowan hardening by preventing dislocation movements. Figure 5 shows the elemental microanalysis results obtained by SEM-EDS for the 190 °C-4 h sample. The region given as Spectrum 2 is α aluminum. It can be seen that the bright precipitate given as Spectrum 1 is rich in Fe, Si, and Cu. The presence of AlFeSi intermetallic precipitate has been reported in similar studies [49,50].

The influence of post-weld heat treatment on the aluminum alloy samples' ultimate tensile strength and maximum strain are summarized in Figure 6. The improvement in mechanical properties by heat treatments has also been reported in previous studies [51–53]. In an earlier study, the tensile strength of a naturally aged sample with the same chemical composition was reported as 238.25 N·mm⁻² and the elongation to fracture as 6.67% after FSW treatment with the same parameters [1]. In this study, the tensile strength of the aluminum alloy plate decreased to $156.3 \text{ N}\cdot\text{mm}^{-2}$, and the elongation at break decreased to 4.90% with the annealing process performed after the FSW process. For the artificially aged specimens, the highest elongation at break was 6.15 % for the specimen aged for 2 h at 190 °C. The lowest elongation at break was 3.19% for the specimen artificially aged at the same temperature for 4 h.

Figure 4. Variation of microhardness of the post-weld heat treated FSWed samples.

Figure 5. EDAX results for 190 °C-4 h sample.

This specimen had the highest tensile strength, and its ultimate tensile strength was 210.8 N·mm⁻². Therefore, artificial aging at 190 °C for 4 h improved the tensile strength by ~34.8% compared to the annealed specimen. İpekoğlu and Çam also found that the mechanical properties of the specimens improved after heat treatment, but the ductility decreased [54]. After artificial aging at 205 °C for 2 and 4 h, the mechanical properties of the specimens relatively decreased and tensile strengths of 187.3 and 189.9 N·mm-2 were measured after 2 and 4 h aging, respectively. Tensile test data and 190 °C and 4 h were determined as peak aging parameters for the artificial aging process of this aluminum alloy. Figure 7 shows the fractographs of annealed and artificially aged specimens after the tensile test. For all heat treatment parameters after welding, populated dimples of uniform size were observed on the fracture surfaces of all specimens, elongated in the load direction. However, it can be seen in Figure 7a that the size of the dimples decreased with artificial aging, with the largest dimples formed on the fracture surface of the annealed specimen. Course dimples and low hardness are essential indicators of low tensile strength [50]. In addition, microvoids were also observed in the course dimples in the annealed sample, where crack formation started. The size of the dimples decreased with artificial aging. Specimens artificially aged at 190 and 205 °C for 2 h have larger dimple sizes than those artificially aged for 4

h at the same temperatures. In the 190 °C-4 h (Fig. 7c) specimen, in addition to fine dimples, flat surfaces were observed, while less flat surfaces were detected in the 205 °C-2 h (Fig. 7d) and 205 °C-4 h (Fig. 7e) specimens. Fine dimples are a characteristic indicator of ductile fracture. Therefore, this specimen is the joint with the highest ductility. Fine-size dimples are also indicative of highstrength [56]. In addition to dimple formation, several tear ridges were observed on the fracture surfaces of 190 °C-4 h (Fig. 7c), 205 °C-2 h (Fig. 7d) and 205 °C-4 h (Fig. 7e) specimens.

Figure 6. The effect of the of post-weld heat treatment on ultimate tensile strength and total elongation of FSWed aluminum alloy plates.

Figure 7. SEM fractographs of the samples a) T0, b) 190 °C-2 h, c) 190 °C-4 h, d) 205 °C-2 h and e) 205 °C-4 h.

4. Conclusions

This study investigated the effects of post-weld heat treatment on the microstructural and mechanical properties of age-hardenable Al-Zn-Mg-Si-Cu aluminium alloy plates welded by friction stir welding (FSW) method. The main results obtained are as follows:

- No failure was detected in the weld zone after FSW, but significant onion ring formation was observed, especially in artificially aged specimens.
- As a result of grain refinement in the weld zone with FSW, hardening with grain boundary strengthening occurred.
- AlFeSi precipitates were detected in the artificially aged specimens, which showed an increase in microhardness over the entire welded section, including the friction stir zone, due to Orowan hardening.
- These specimens reached a peak aging parameter of 190 \degree C and 4 hours of soaking time, and the ultimate tensile strength was 210.8 N·mm-2 .
- Ductile fracture mode was observed in all annealed and artificially aged specimens.

Declaration

The authors declared no potential conflicts of interest with respect to the research, authorship, and/or publication of this article. The author) also declared that this article is original, was prepared in accordance with international publication and research ethics, and ethical committee permission or any special permission is not required.

Author Contributions

D.Arslan and S.I.Ayvaz developed the methodology together. D.Arslan and S.I.Ayvaz performed the analysis together. S.I. Ayvaz supervised and improved the study. D.Arslan and S.I.Ayvaz wrote the manuscript together.

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Antagonistic interaction of HSP90 inhibitor XL-888 and 5-FU combination treatment in breast cancer cells

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1. Introduction

Breast cancer is an important and universal health problem for which new treatment strategies must be continually sought to improve survival rates[1,2]. The commonly used drug in the treatment of breast cancer is 5- FU. However, its clinical usage is limited due to side effects and dosage limits. A popular strategy for cancer treatment, called combination therapy, is being explored to overcome these limitations. Combination therapy involves using drug molecules to increase the effectiveness of existing treatments while relieving unwanted side effects. HSP90 has recently come into play as a key target in combination treatments. HSP90 is associated with many proteins within the cell and plays an important role in facilitating the maturation of client proteins (CPs), which are crucial for cellular survival, differentiation, and growth [3]. HSP90 appears to be an interesting therapeutic target due to its association with critical client proteins such as estrogen receptor (ER), human epidermal growth factor receptor 2 (HER2), and progesterone receptor (PR), which are prominent in breast cancer [4]. Many HSP90 inhibitors

that exhibit high selectivity and significant anti-cancer potency despite different chemical structures have been discovered. Among these inhibitors, XL-888, an advanced oral drug designed to target HSP90, has demonstrated the capacity to suppress HSP90 activity significantly without affecting other kinases [5]. XL-888 inhibits HSP90 activity, which may impair the survival of cancer cells under stress conditions and may, therefore, increase the effectiveness of chemotherapy drugs when used combined with chemotherapy drugs [6-7]. Ongoing preclinical trials have investigated the efficacy of XL-888 in melanoma and advanced pancreatic/colorectal cancer, lung cancer, and breast cancer [8–10].

Breast cancer is classified into two molecular subtypes: hormone receptor-positive and triple-negative. The majority, about 80%, fall into the hormone receptor-positive category, while the remaining 20% are classified as triple negative. Combining different drugs in the treatment of breast cancer, for example, using two different chemotherapy drugs or a targeted drug and a chemotherapy drug together to target different biological properties of cancer cells, is one of the effective treatment strategies [11-13].

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This research aimed to investigate how 5-FU and XL-888 combination therapy affected MCF-7 and MDA-MB-231 cell lines. When administered alone or in combination, the cytotoxic potential of these agents was evaluated at 24-hour and 48-hour periods. Evaluation of drug interactions was done using the Chou and Talalay methodology.

2. Materials and Methods

2.1 Materials

 MTT (3- (4,5- dimethylthiazol -2 - yl)-2,5 diphenyltetrazolium bromide) was provided by Serva. XL-888 was purchased from AdooQ® Bioscience. 5-FU was obtained by Gold Biotechnology*.* Dulbecco's Modified Eagle's medium High Glucose (DMEM), fetal bovine serum heat-inactivated (FBS), penicillin-streptomycin solution and other necessary chemicals (L-glutamine, phosphate buffer saline (PBS), and trypsin–EDTA) for use in cell culture were purchased from Biological Industries. The breast cancer cell lines, MDA-MB-231 and MCF-7 were purveyed from American Type Culture Collection.

2.2 Cell Culture

To culture the MDA-MB-231 and MCF-7 cancer cell lines, we employed DMEM High Glucose medium supplemented with 10% FBS. The cancer cell lines were maintained in an incubator at a temperature of 37°C within a humidified environment containing 5% CO₂.

2.3 Cell Viability Assay

The cell proliferation assay, generally preferred and known as MTT, was chosen to evaluate the in vitro cytotoxic impact of XL-888 and 5-FU. Culturing the cancer cell lines was performed in 96-well plates with a density of $5x10^4$ cells per well. Subsequently, the cells were exposed to varying concentrations of XL-888 (100 nM -1.5625 nM) and 5-FU (10 μ M-0.156 μ M) for 24 h and 48 h. Following the incubation period, a solution of MTT (5 mg/mL) was introduced to each well and then allowed to incubate at 37°C for 3 hours. After the formed formazan product was dissolved in dimethyl sulfoxide, the absorbance of each well was measured at 570 nm, and the viability of the cancer cells was calculated as a percentage relative to the control group. Moreover, the initial concentrations for cytotoxicity analyses of the drugs, 100 nM for XL-888 and 10μM for 5-FU, were determined by preliminary testing following literature research. The combination ratio was based on these initial concentrations, and serial dilution was studied by keeping the ratio of 100 nM: 10μ M, i.e. 1:100, constant.

2.5 Combination Index

In drug combination studies, the Chou and Talalay method, a type of software, generally stands out in determining combination effectiveness. This approach relies on the utilizing of the median effect equation as its fundamental principle. The derivation of this method from the law of mass action has made it more preferred in practical applications [14-15]. The widespread adoption of the Chou and Talalay method in practical applications began in 2005 with the introduction of CompuSyn software [16]. This program calculates effective dose-dependent combination indexes by comprehensively analyzing drug interactions using cell viabilities and combination ratios depending on drug concentrations [17]. To assess the effectiveness of the XL-888 and 5-FU combination, CompuSyn software version 1.0 was used to calculate the combination index (CI). The Chou-Talalay method was employed for CI determination.

2.4 Statistical Analysis

GraphPad Prism 8.0 program was utilized with significance regarded as $p<0.05$.

3. Results and Discussion

The MTT assay, a method for assessing cell viability, was used to evaluate the impact of XL-888 and 5-FU, individually and in combination, on MDA-MB-231 and MCF-7 cell lines. As drawn in Figures 1A, 1B, 1C and 1D, both XL-888 and 5-FU exhibited time- and dose-dependent inhibition of cell viability in MDA-MB-231 and MCF-7 cells.

According to the IC50 values calculated in Table 1, the HSP90 inhibitor XL-888 had a cytotoxic effect on the cells at low concentrations in both cell lines. Compared to the MDA-MB-231 cell line, MCF-7 cells were more sensitive to HSP90 inhibition. Since the expression of HSP90 protein is significantly higher in tumor cells, inhibition of HSP90 in tumor cells is an important strategy for cancer therapy [18]. Previous research has demonstrated that decreasing HSP90 alone or in combination inhibits the development of breast cancer cells. Ganetespib, an HSP90 inhibitor, acted as an inhibitor of oncogenic signaling in MDA-MB-231 cells at low doses and suppressing tumor growth by increasing the apoptotic markers Parp and Bim [18]. Phase II/III clinical studies of the HSP90 inhibitor 17-AAG in breast cancer and other tumors showed up-and-coming [20]. Moreover, tanespimycin, a different version of 17-AAG, showed promising antitumor activity in phase II trials when combined with trastuzumab in the scope of HER2-positive metastatic breast cancer [21]. HSP90 inhibitors have significant potential to induce apoptotic cell death pathways in breast cancer [22]. In a previous study, our research focused on the HSP90 inhibitor Debio-0932, this compound was found to effectively inhibit human Hsp90 ATPase activity and reduce cell proliferation by inducing apoptotic pathways in MDA-MB-231 and MCF-7 breast cancer cell lines [23]. Proia et al. evaluated the apoptotic potential of the combination of ganetespib with doxetaxel, DOX and paclitaxel. The combination of DOX and ganetespib caused an increase in Parp expression in MDA-MB-231 cells.

Figure 1. The antiproliferative effects of XL-888 (A) and 5-FU (B) were examined in the MCF-7 cell line, while XL-888 (C) and 5-FU (D) were assessed in the MDA-MB-231 cell line at both 24 h and 48 h.

Further, ganetespib treatment increased the expression of Parp and Caspase-7 in BT-20 cells [24]. Another combination therapy study by Muhammedyan et al. showed that the effect of NVP-AUY922, an HSP90 inhibitor, combined with DOX on the MCF-7 cell line was stronger than the effects of the individual drugs [25].These findings emphasize the prospective therapeutic effects of HSP90 inhibitors in the treatment of breast cancer by inducing apoptosis and inhibiting metastasis.

In this research, 5-FU had a strong cytotoxic impact on MDA-MB-231 and MCF-7 cells after 24 and 48 hours. 5-FU is a pyrimidine analog antimetabolite that inhibits thymidylate synthase activity. This molecule has the ability to arrest the cell cycle at the S phase and induce apoptosis [26]. In first- and second-line breast cancer treatment, 5-FU is used in combination with other drugs such as doxorubicin, methotrexate, and cyclophosphamide [27]. However, 5-FU treatment is restricted by drug resistance caused by breast cancer resistance proteins and dangerous side effects. As a result, finding novel selective breast cancer drugs that may be employed as a single dosage or in combination with other cytotoxic treatments is critical [28].

The clinical efficacy of 5-Fluorouracil (5-FU)-based chemotherapy is limited by issues of multidrug resistance and dose-dependent cytotoxicity. To address these issues, studies are being conducted using 5-FU and additional anticancer drugs as a new combination that interacts with cells.

In this study, the combined use of XL-888 and 5-FU revealed an antagonistic effect on MDA-MB-231 and MCF-7 cell lines. These results show that contrary to the synergistic effect usually sought, the combined use of these two drugs completely reverses the expected effect (Figures 2A and 2B). Analysis results show that the antiproliferative effects of the two agents on breast cancer cells are higher when used alone compared to their combined use. To confirm this antagonism, the combination of two drugs prepared at a fixed ratio was serially diluted and treated to the cells, and then CompuSyn software, designed with the methodology introduced by Chou and Talalay, was used to calculate the combination index (CI) [29]. This analysis showed the following combination effects: CI > 1 demonstrates antagonism, CI <1 indicates synergism, and CI $= 1$ suggests additivity [30]. The combined treatment of XL-888 and 5-FU exhibited an antagonistic effect in both MCF-7 (Figure 2C) and MDA-MB-231 (Figure 2D) cell lines.

In previous studies, combining Hsp90 inhibitors with chemotherapy agents has typically shown synergistic effects. In combination with doxorubicin, the HSP90 inhibitor AUY-922 increased caspase-3 expression, a marker of mitochondrial apoptosis, and decreased VEGF mRNA levels [25].

In a preclinical study, 17-AAG demonstrated a positive effect on breast cancer cells treated with bevacizumab

(VEGF inhibitor) [31]. In another study, colorectal cancer cells treated with the HSP90 inhibitor AUY-922 were shown to be more sensitive to 5-FU-based chemotherapy in vitro and animal models. This suggests that the coadministration of 5-FU with AUY-922 may be a valid treatment strategy [32]. In the other study, Liu and colleagues investigated the effect of the combined use of 5-FU and the HSP90 inhibitor SNX-2112 in esophageal cancer. Unexpectedly, the combined use of these two agents resulted in antagonistic results in the cells. Further investigation into the molecular mechanisms behind this response revealed several plausible factors [33]. Based on these possible factors, the combination of SNX-2112 and 5-FU could potentially lead to opposite results by countering G2/M cell cycle arrest, reducing Hsp90 client proteins and suppressing caspase-dependent apoptosis, inhibiting the initial reduction of MMP (Mitochondrial Membrane Potential).

In a study investigating the combined effect of the chemotherapeutic drug DOX and the HSP90 inhibitor XL-888 on liver cancer cell lines, it was shown that the simultaneous use of both drugs strongly triggered apoptosis [34]. In addition, the combination of DOX and the HSP90 inhibitor MPC-3100 was found to be more effective in triggering apoptosis in breast cancer cells than either drug alone and showed a synergistic effect [35]. In a study examining the combined effect of 5-FU and the HSP90 inhibitor MPC-3100 on liver cancer cell lines HepG2 and HUH-7 [36]. It was observed that both drugs individually showed dose- and time-dependent cytotoxic effects. However, particularly the combined use of the two drugs resulted in an antagonistic effect rather than the expected additive or synergistic effects on HepG2 and HUH-7 cell lines.

Table 1**.** IC₅₀ values of XL-888, 5-FU, and combination form in MDA-MB-231 and MCF-7 cell lines at 24 h and 48 h.

Figure 2**.** Survival graphs of MCF-7 (A) and MDA-MB-231 (B) XL-888 and 5-FU combination-treated cancer cell lines; graphs were generated based on combination index data plotted against fraction affected (Fa) values obtained from Chou-Talalay median effects analysis for MCF-7 (C) and MDA-MB-231 (D) cell lines.

* CI: combination index, Fa: affected fraction; ED50, ED75, and ED90 refer to the amount of dose required to inhibit 50%, 75%, and 90% of cells, respectively, and reveal the potential of a drug or treatment.

Furthermore, in a study investigating the effectiveness of combined therapy of Hsp90 inhibitors and anticancer agents in pancreatic cancer cell cultures, it was found that although both inhibitors have the same mechanism of action, the use of different Hsp90 inhibitors with the same anticancer agent may lead to different effects. It has been suggested that these differences may be due to the different structures and activities of both compounds [37].In addition, another study, pathway analyses of the anticancer activities of Hsp90 inhibitors XL-888 and Debio0932 on the neuroblastoma cancer cell line SH-SY5Y showed that these inhibitors are important in regulating many cancer-related pathways (such as invasion, metastasis, angiogenesis, and apoptosis) [38]. In a study conducted to overcome the resistance to trastuzumab treatment in HER2-positive breast cancer, the Hsp90

4. Conclusions

In this study, the potential of combined use of the conventional chemotherapy agent 5-FU and HSP90 inhibitor XL-888 in the treatment of breast cancer was investigated. The findings separately revealed the doseand time-related cytotoxic impacts of each agent on MDA-MB-231 and MCF-7 breast cancer cell lines. However, data from the combined use of the two drugs showed surprising effects. Co-administration of 5-FU and XL-888 had an antagonistic rather than predicted additive or synergistic effect on MDA-MB-231 and MCF-7 cell lines. This unexpected result underscores the complex interaction between different drugs and demonstrates the importance of comprehensive evaluation of all variables during combination therapy development. Although combining XL-888 and 5-FU was not effective in increasing cytotoxicity in breast cancer cells, information gained from studies such as ours will contribute to the creation of more targeted and effective treatments against breast cancer and other malignancies.

Declaration

The author affirms that the article is original and was prepared in compliance with international publication and research ethics standards. No potential conflicts of interest are declared by the author regarding the research, authorship, and publication of this article. Additionally, the author states that no special permission or ethical committee approval was necessary for this study.

Author Contributions

Nazan GÖKŞEN TOSUN: Investigation, Methodology, Formal analysis, Writing-review & editing, Visualization, Methodology, Resources.

inhibitor HVH-2930 was reported to prevent angiogenesis and tumor growth in trastuzumab-resistant xenograft mice in vivo. In addition, in this study, it was reported that the combination of the Hsp90 inhibitor HVH-2930 with paclitaxel exhibited a synergistic effect in JIMT-1 xenografts [39].

In our research, possible explanations for the antagonistic effects of combining 5-FU with an HSP90 inhibitor in breast cancer cells may overlap with these findings. In a broader context, the data obtained in our study should further investigate the antagonistic interaction between HSP90 inhibitor and 5-FU across various types of cancer. It is recommended that the combination of 5-FU and HSP90 inhibitors be considered when planning clinical applications.

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A Review on hydrogen embrittlement behavior of steel structures and

measurement methods

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1. Introduction

 Hydrogen gas is usually trapped in metal by a number of ways. These result from hydrogen embrittlement, interstitial diffusion [1], chemical reaction, and absorption. Hydrogen atoms can become trapped in high-stress areas of the metal structure, including grain boundaries or fracture tips, in the event of hydrogen embrittlement [2]. Even at lower stress levels than those that would typically cause fracture in the absence of hydrogen, the buildup of hydrogen at these locations can cause localized embrittlement and encourage the spread of cracks [3]. A number of variables, including the metal's crystal structure, hydrogen solubility, diffusion kinetics, and the specific environmental circumstances in which the metal is exposed to hydrogen, affect the exact mechanism of hydrogen trapping [4]. A phenomenon known as hydrogen embrittlement, in which hydrogen gases become trapped on the inside of the liquid metal during the solidification

process and penetrate the base material, can cause steel to become brittle. This can lead to a loss of ductility and toughness, making the material more susceptible to brittle fracture, even under relatively low stress conditions [5]. Fossil fuels could eventually be replaced with hydrogen, which reducing the environmental impact. However, hydrogen absorbs and permeates through metals, equipment exposed to hydrogen is susceptible to harmful consequences [6]. ASTM F2078 defines HE as "a permanent loss of ductility in a metal or alloy caused by hydrogen in combination with stress, either internal residual stress or stress applied externally." [7].

As hydrogen builds up in the crystal lattice and generates stress, Figure 1 illustrates the initial effect of hydrogen interaction, which is the development of HE [5].

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Figure 1 Diagrammatic representation of the interactions between hydrogen atoms and a metal crystal structure [5].

Figure 2. Hydrogen interaction and HE in AHSS [11]

 The second effect is called a hydrogen attack and is caused by hydrogen building up in the crystal lattice, which reacts and interacts with other crystal lattice components to alter the stress and composition. Hydrogen has several advantages, such the possibility of being clean and renewable, but there are also major safety concerns. Not only is hydrogen flammable [8] and explosive, but it can also penetrate and erode metallic surfaces especially from high strength steels surfaces, which is a serious safety concern when handling and storing hydrogen [9]. The majority of metals and alloys have the capacity to absorb hydrogen, and its build-up near internal defects (such as vacancies, grain boundaries, dislocations, precipitates, and inclusions) poses a significant risk to iron, steel, nickel and titanium-based alloys, and numerous other materials that are typically used in industrial settings [6].

1.1 Interaction Between H and AHSS

 Following loading distribution, trapping, and migration; the little H atoms' ability to interact with practically all metal flaws is well known. Because of this, its local distribution, atomic entrapment, and migration upon loading are essential for the activation of particular HE processes and, consequently, the local and global HE resistance. A lot more when it comes to AHSS than singlephase model alloys. Both in its non-distorted and deformed stages, complex microstructure often consists of many phases and lattice faults of various sorts [10]. Furthermore, the local thermodynamics and kinetics of H migration and redistribution are significantly impacted by the very complex evolution of microstructure, local stress states, and defects that is caused by deformation.

Within AHSS, there are typically significant differences across phases with regard to mechanical properties, the evolution of defects under load, H solubility, and difusivity. Different HE mechanisms are triggered by such fundamental differences, which in turn cause different H interactions with different phases. Both H trapping and migration as well as the local mechanical driving force for damage creation would be significantly changed by the local strain/stress states and how they evolved upon deformation [12].

Over the years, a great deal of research has been done on the hydrogen-induced degradation of metals; yet, hydrogen embrittlement, remains the cause of many industrial failures and the consequent catastrophic releases of dangerous compounds into the environment [13]. The aim of this review paper is to summarize the recently published papers specifically on the definition of HE, the events that lead to HE, the method for charging hydrogen, the technique for measuring hydrogen concentration, and the safety measures that limit the passage of hydrogen to steel.

2. Hydrogen Embrittlement in Steel

 For advanced high strength steels (AHSS), hydrogen embrittlement becomes more significant when strength levels rise above 1000 MPa [6]. AHSSs have favorable mechanical properties, but hydrogen-induced processes have the potential to compromise their integrity [7]. When exposed to hydrogen, these materials lose some of their ductility and eventually experience HE [1-5]. This limitation could arise from exposure to the service environment, during the manufacturing process, during product assembly and termination, or both. The risk of hydrogen embrittlement can be raised by a number of variables the most common of them are shown in Figure 3.

 Environmental Factors: The creation of hydrogen is facilitated by corrosive environments, which raises the possibility of embrittlement. Hydrogen production during corrosion processes can be accelerated in acidic environments (Silva et al., 2021). Although cathodic protection systems are useful in preventing corrosion, they may cause embrittlement by introducing hydrogen onto the metal surface. And also at higher temperatures, hydrogen embrittlement often reveals itself more strongly.

But it can also happen at room temperature, particularly with some materials [2].

 Material Susceptibility: High strength materials, like certain alloys and high-strength steels, are more prone to hydrogen embrittlement [8]. The alloy's susceptibility to hydrogen embrittlement may vary depending on its composition. Certain alloying materials may reduce or increase the effects. Compared to other microstructures, steels with martensitic microstructures are more vulnerable to hydrogen embrittlement [9]. Embrittlement may result from hydrogen present at grain boundaries. Some microstructures might be more vulnerable than others, such as those with finer grains. Hydrogen can be introduced into the material through heat treatment and welding procedures [15]. Attentive regulation of these procedures is necessary to reduce the possibility of embrittlement.

 Load (Stress): High levels of tensile stresses can make a material more vulnerable to hydrogen embrittlement [16]. This is especially relevant when materials are being mechanically loaded during manufacturing or during use [3]. Figure 3 shows a graphic representation of the key parameters for HE. Environmentally assisted cracking (EAC) can be observed with either a static or dynamic applied load, and the damage form is known by several names such as corrosion fatigue, stress corrosion cracking, sulfide stress cracking, and so on. One or more harm mechanisms (which may work concurrently) have the ability to initiate and/or assist each of the aforementioned damage kinds.

 It is notable that the degree of hydrogen embrittlement varies depending on the material and application and is dependent on the interaction of these factors [17]. The implementation of preventive measures in corrosive settings, control over processing conditions, and suitable material selection are examples of prevention and mitigation strategies [10].

 The development of successive generations of AHSS has resulted in a considerable improvement in the characteristics of steel during the past few decades [18]. Given their increasing usage in automobiles, these materials are especially intriguing to the transportation sector, such as the automotive industry, which is looking to minimize weight and consumption. However, because of their susceptibility to cracking, especially when hydrogen is present and causes HE, the potential of these novel steels is not being fully utilized. The local diffusible hydrogen content is the crucial parameter that determines the HE, which also depends on the steel's ability to trap hydrogen in an irreversible or reversible manner [11].

Figure 3. Factors that increase the risk of hydrogen embrittlement [3]

3. Mechanisms Causing HE

Complex processes that vary based on the particular material and ambient circumstances are responsible for hydrogen embrittlement. A few of the main mechanisms involved are as follows:

Adsorption-Induced Dislocation Emission (AIDE), Stress-Induced Hydrogen Embrittlement, Hydrogen-Enhanced Strain Localization, Hydride-Induced Embrittlement (HIE), Hydrogen-Enhanced Decohesion Mechanism (HEDE), Hydrogen-Enhanced Local Plasticity Model (HELP), and Hydrogen Changed Microfracture Mode (HAM) [9-12].

3.1 Hydrogen Enhanced Decohesion (HEDE)

 According to the HEDE mechanism, hydrogen weakens an alloy's cohesive strength [19]. Pfeil [20] first suggested the decohesion process in 1926, stating that "The cohesiveness between grain boundaries and cubic cleavage planes was reduced by hydrogen." A more refined version suggested that the cohesive strength of lattice planes or interface boundaries is decreased in the presence of hydrogen [21]. The underlying mechanism of HEDE is that an electron from a hydrogen atom dissolved in steel enters the unfilled three-dimensional shell of an atom (such as a Fe atom) that constitutes the steel. The interatomic repulsive forces are increased by these additional electrons in the d-shell, lowering cohesive strength. [12].

Figure 4 displays the HEDE mechanism schematic. The atomic bonds at the crack's tip weaken as a result of the crystal lattice expanding due to the admission of hydrogen. This leads to a reduction in the energy needed to facilitate the propagation of cracks, culminating in a macroscopic brittle fracture. Involving the tensile separation of atoms caused by the following: (i) hydrogen in the lattice; (ii) adsorbed hydrogen; and (iii) hydrogen at particle–matrix contacts, which weakens interatomic connections [12].

Figure 4. Schematic of the HEDE mechanism [12]

Figure 5. Schematic of the HELP mechanism [15]

3.2 Hydride-Induced Embrittlement (HIE)

 This mechanism, which has strong experimental and theoretical backing, is one of the well-known mechanisms of HE. Hydrides originally formed in the crack's stress field, and they expanded to enormous sizes by forming new hydrides in the other hydrides' stress fields rather than by expanding from individual hydrides [14]. They demonstrated how the smaller hydrides developed into the larger hydrides through growth. Brittleness of the resulting hydride nucleation and development along with the autocatalytic process appears to be the primary cause of embrittlement of the typical hydride former element [15].

3.3 Hydrogen Enhanced Localized Plasticity (HELP)

 The hydrogen atom is accumulated in close to the crack tip. Moreover, it lessens the opposition to dislocation motion. As a result, dislocation becomes more maneuverable and functions in a metal lattice as a carrier of plastic deformation [22]. It might be clear that it depends on the material's microstructure, stress intensity, or hydrogen clustering. Fractographic examination was carried out to verify the material's microstructure characteristics. HELP contains a wide variety of structures, such as Face-Centered Cubic (FCC), Body-Centered Cubic (BCC), and Hexagonal Close-Packed (HCP) type structures. The HELP mechanism will cause a brittle fracture surface with tear ridges, dimples, and slip $[15]$.

Figure 6. Diagrammatic representation of the hydrogen-induced dislocation emission from the crack tip in the AIDE mechanism model [9].

3.4 Adsorption Induced Dislocation Emission (AIDE)

This represents the combination of HEDE and HELP. Adsorption of the solute hydrogen atoms occurs close to the fracture tip. Because of the solute hydrogen atom dislocation that forms close to the fracture tip, hydrogen adsorption at the crack tip weakens the cohesive strength and interatomic bonding of materials via the HEDE process. Dislocation promotes the formation of micro voids through the HELP mechanism and the slip-induced crack development [21]. The process known as "adsorption-induce dislocation emission mechanism" occurs when hydrogen is adsorbed on a surface, which further enhances or inhibits the dislocation nucleation at the surface and has a significant impact on surface energy. In crystalline solids, dislocation nucleation, or emission from the fracture point, plays a crucial role in the ductilebrittle transition. A quantitative model was developed, as shown in figure 6, to capture the hydrogen-affected dislocation emission from the crack tip and its effect on hydrogen embrittlement. [23].

3.5 Hydrogen Assisted Micro-Fracture Mode (HAM)

 Hydrogen causes a shift in the material's microfracture mode, causing the ductile to become brittle. Hydrogen charge decreased the material's ductility and caused the ultimate tensile strength fracture mode to shift from a cupand-cone to a brittle shear fracture mode. This is also due to the shear fracture mode being amplified by the high concentration of hydrogen at the dislocation. HAM is the term used to describe the shift in micro fracture mode caused by the influence of hydrogen [5].

4. Measurement of Hydrogen Concentration in Steel

 It's critical to measure the hydrogen concentration in steel, particularly in situations where hydrogen presence may compromise the integrity of the material. Hydrogen has a capacity to embrittle steel, which can alter its

mechanical characteristics and perhaps result in structural failure [24]. Several well used techniques exist for determining the hydrogen content of steel. The development of the currently available AHSS, which reduce their vulnerability to HE in the presence of low hydrogen concentrations, requires precise monitoring of the concentration of hydrogen and trapping in the materials [25]. Methods for measuring concentrations and analyzing hydrogen trapping in metallic materials are: hydrogen microprint technique (HMT), thermal desorption spectroscopy (TDS), hydrogen permeation test (HPT), linearly increasing stress test (LIST) and gas chromatography (GC). The most widely used experimental methods for measuring concentrations and analyzing hydrogen trapping in metallic materials are now TDS and GC with thermal conductivity detectors (TCD) [26].

4.1 Hydrogen Microprint Technique (HMT)

 This technique has been used to calculate the hydrogen's diffusion through metal. Knowing these pathways allows for the identification of their unique microstructure and the determination of the hydrogen's effect. This highresolution, very accurate HMT approach is relatively straightforward and distinct [27]. The hydrogen distribution on the stress field, for example in notched and deformed steel, was found using the HMT technique. The HMT process can be used on a variety of materials, including austenitic stainless steel, high strength steel, and low carbon steel [28]. During the procedure, a thin layer of AgBr gel is placed on the face of the hydrogen-charged substance. Hydrogen interacts with a silver salt as it breaks free of the metal. Following the reaction, silver ions take on a metallic form and leave an area where hydrogen contact occurred. The area has silver particles, and the extra unreacted gel is being removed from it. The sample is examined using a scanning electron microscopy (SEM), and when the sample is examined, hydrogen exits the areas where silver is present [29].

4.2 Thermal Desorption Spectroscopy (TDS)

Certain features of the TDS spectra are associated with hydrogen trapping at dislocations, vacancies, vacancy complexes, grain boundaries, and interfaces of nonmetallic inclusions (NMI), which enables the measurement of the hydrogen trapping activation energies. The TDS technique measures the amount of desorbed hydrogen by employing a limited and controlled heating procedure. There are traps in steel, and it's these traps that lead to the buildup of hydrogen. Hydrogen is absorbed by thermal energy when steel is heated and is released when the absorbed energy reaches a critical threshold, which is the same as the activation energy of desorption. Consequently, the temperature at which hydrogen atoms

are released is known as the desorption temperature. With quadrupole mass spectrometry, the amount of desorbed hydrogen is quantified. [11].

4.3 Hydrogen Permeation Test (HPT)

 The easiest method of measurement and the amount of diffusible hydrogen in steel are also determined by permeation testing. It is possible to determine and access steel's HE susceptibility if the quantity of diffusible hydrogen is known. This permeation test has been used in conjunction with other testing techniques to successfully test steel [30]. In essence, this permeation test uses a twocell system, with an entering cell (also called a charge cell) and an oxidation cell (also called an exit cell) in each chamber. A steel membrane divides these two chambers [31]. The process of electrochemistry has been employed for hydrogen charging. After entering the cell to charge it, the hydrogen moves to the oxidation cell with the assistance of membrane [15].

Figure 7. schematic diagram of HMT [29]

Figure 8. Schematic view of the TDS apparatus [11]

Figure 9 Schematic illustration of the electrochemical hydrogen permeation test [15]

4.4 Linearly Increasing Stress Test (LIST)

 This mechanical testing technique is used to find the HE in various material types. In this example, a sample was loaded and the applied stress was steadily increased until failure occurred. A load is applied by weight movement, and a motor regulates the load's rate of motion. Figure 10 shows the LIST method schematic. The specimen will fail whenever the threshold stress is reached in this load control HE measurement method, confirming the completion of the test. Using SEM, the fracture surfaces can be assessed after the LIST test [32]. Thus the impact of hydrogen on the material's interior microstructure can be determined. The most recommended, flexible, and userfriendly method for it is SEM. Additionally, SEM offers a three-dimensional image and shows fracture features like fisheyes, dimples, and micro voids [33 -34].

5. Preventive Measures of HE

 The interactions between the hydrogen concentration and stress gradient cause hydrogen to diffuse toward and accumulate in the stress accumulation region. Fracture failure happens as the local hydrogen concentration approaches the critical value. Internal and external hydrogen are the two categories into which hydrogen sources are often divided. While hydrogen generated during service is external hydrogen caused by corrosion, hydrogen gas, and H2S gas acid environments, internal hydrogen is produced during material preparation procedures like melting, welding, pickling, and plating [35]. Based on the literatures researchers proposed and recommended so many preventive measures that restrict hydrogen diffusion in to the steel and any other metals.

 In addition to proper material design, notches, abrupt and irregular variations, and residual stresses should be eliminated prior to processing [36]. Hydrogen, which is absorbed and could cause failure or damage, has been separated from the material by baking. Baking temperature varies depending on the substance being baked, as it is essentially a heat treatment technique. Using an acidic solution, the pickling process was used to remove some scale and oxide compounds from the material. Since this acid is what causes hydrogen to diffuse, mechanical methods like sandblasting, vapor blasting and grit blasting, will be used for reducing it [28].

 Moreover, HE can be avoided by coating the base material with a protective layer and incorporating metal alloys into it. Mechanical plating, vacuum deposited coating, and organic coating are a few of the coating methods. Using effective inhibitors is also essential. When titanium is readily available in large amounts, its presence can lessen a hot stamped boron steel's HE susceptibility by producing titanium carbide within the material [21]. Certain authors claim that alloying aluminum can also aid in reducing the HE impact. Graphene and niobium coatings can also prevent material from HE. Steel is treated with cadmium to prevent hydrogen from leaking through. Diffusion of hydrogen becomes blocked when nickel is coated on steel. In order to prevent hydrogen from penetrating, a variety of coatings, including WC, TiC, TiN, TiO2, alumina, BN, and Cr2O3, have been employed [37].

 Generally, there are two methods that can be used to prevent HE. First, surface treatments including coatings and modification treatments are applied. These methods are used to halt HE from the outside. The second tactic entails altering the material's microstructure by refining the alloy's microstructure and introducing or eliminating the appropriate alloy components [35].

5.1 Surface Coating

 Steels that are prone to hydrogen embrittlement, particularly lightweight, high-strength steels or low-cost alloy steels, should be able to be used in hydrogen-based economies thanks to hydrogen barrier coatings. Hydrogen barrier coatings are protective layers that have the ability to slow down, stop, or prevent hydrogen permeability [38]. They are made of materials with low intrinsic hydrogen diffusivity and solubility. When a coating is applied to a metal surface, hydrogen entrance into the alloy is inhibited, leading to high hydrogen resistance in the alloys [39]. When steels are treated to improve their resistance to air corrosion, a 1-3 μm thick oxide layer is formed on their metal surface. This process is known as surface blackening. Furthermore, alloys' susceptibility to HE can be decreased and hydrogen infusion effectively suppressed by surface coatings of Ni, Cd, Al, and Al–Ni complex coatings [35]. Moreover, as figure 10 illustrates, hard coatings like TiC, Al2O3, and Si3N4 are capable of significantly reducing hydrogen diffusion behavior [35]. Found that after coating the surface of stainless steel with a 1 μm-thick TiN film, the metal's hydrogen diffusion coefficient decreased by five orders of magnitude.

Figure 10 Schematic representation of the LIST apparatus [32]

Figure 11 Temperature dependence of the hydrogen diffusion coefficient in different films [35]

 The large difference in the rate of hydrogen diffusion between the austenite phase and the TiC and Al2O3 coatings suggests that these films may be a good barrier to enhance the HE resistance of austenite steels. When nickel is coated on steel, it prevents hydrogen from diffusing through. Many coatings have been used as a barrier to stop hydrogen permeability, such as WC, TiC, TiN, TiO2, alumina, BN, and Cr2O3. Coatings consisting of Pt, Cu, Cd, Ag, Al, and Au can help reduce the amount of hydrogen that migrates inside steel [33] [38]. Another way to prevent HE is to coat the base material with a protective layer and incorporate some metal alloys into it [40]. Mechanical plating, organic coating, and vacuum deposited coating are a few of the coating methods. Using strong inhibitors is also crucial. When titanium is widely accessible, its presence can lower the HE susceptibility of a hot stamped boron steel by forming titanium carbide inside the alloy [41]. According to some authors, alloying aluminum may also aid in reducing the HE effect. Coatings using niobium and graphene can also shield materials from HE [42].

5.2 Modifying Microstructure of Material

 Since the basic principles underlying the HE process are still poorly understood, developing microstructure design methods to reduce HE is difficult [43]. Nevertheless, there have been some recent advancements in this subject. Based on different literatures: Surface treatment, solute segregation and heterogeneity, grain boundary engineering, second-phase entrapment, and grain refining are some of these techniques [41, 44]. With rising Ni, Al, and Mo elemental concentrations and falling C, Si, P, and S elemental concentrations, HE will fall. For instance, it was found that the HE of the Mn–B steel rises with an increase in the C element level. Nevertheless, with C concentrations higher than 0.3%, the sensitivity to HE does not vary [17]. According to Xinfeng Li et al. [35], once the P element content was decreased, the threshold stress intensity factor of 4340 steels increased by a factor of five. P segregation may be hindered by a decrease in the concentration of Cr, Mn, Si, or an increase in the content of Mo and Ti. First-principles calculations [46] indicate that Al inhibits hydrogen diffusion more than Si in BCC iron. As a result, low HE sensitivity is found in bainitic steel with a high Al element content. The Fe–Mn–C steel's HE resistance is increased by Cu and Al components, which also boost the stacking fault energy and reduce the stress concentration at the grain boundaries. Furthermore, regular combinations of elements such as Mo, V, and Ti with C result in the formation of carbides, which improve the alloy's resistance to HE and act as permanent hydrogen traps [47].

 Steels' HE is dependent on their microstructures. In particular, the highest HE susceptibility is found in the martensitic structure, which is followed in order by bainite, pearlite, and austenite [48]. HE susceptibility was lower in fastener steels having a pearlitic microstructure than in bainitic steel [35]. Although martensitic and pearlitic steels have the same strength level, pearlitic steel is known to have a greater HE resistance.

6. Result and discussion

 The phenomenon of HE in steel and other materials is now well established. This study addresses materials that are prone to hydrogen penetration or diffusion, as well as the mechanisms and causes that lead to mechanical property degradation and hydrogen-related failures like HE. Preventive measures to get rid of or lessen hydrogen penetration in the material are talked about concurrently. These preventative measures can lower the likelihood of HE by obstructing the diffusion of hydrogen. The table below summarizes the relevant prior related studies in summary form. Following investigations into measurement strategies to determine the hydrogen content of steel, a variety of techniques have been employed, including the HMT, TDS, HPT, LIST, and GC. With the distinct benefits of each technique, researchers can analyze diffusion behavior through HPT, exact quantification with TDS, surface mapping with HMT, and many other facets of hydrogen activity in steel. Parallel research on preventive measures against HE revealed a multidisciplinary strategy that includes diffusion barrier coatings, modification of microstructure, new materials development with improved hydrogen-trapping capabilities, coating selection and design, and mechanical design optimization. By decreasing hydrogen absorption into the material, increasing hydrogen trapping capacity, improving material characteristics, or altering mechanical design features to lessen stress concentrations and minimize embrittlement, these preventive approaches seek

to reduce HE. Researchers can create complete methods for effectively understanding, mitigating, and preventing hydrogen embrittlement in steel structures and components by combining insights from measuring techniques with preventative measures. By working

together, it can be possible to improve testing methods and preventive measures, which leads to a better knowledge of hydrogen embrittlement mechanisms and useful solutions to protect steel materials from its harmful consequen

$Ref.$ #	Author	Types of Mechanisms	Results and conclusions	
		Causing HE		
$[19]$	Djukic, M. et al.	HEDE	Results indicate a simultaneous action of the hydrogen-enhanced decohesion (HEDE) and hydrogen-enhanced localized plasticity (HELP) mechanisms of HE, depending on the local concentration of hydrogen in investigated steel.	
$[21]$	Kappes, M.et al.		The cohesive strength of lattice planes or interface boundaries is decreased in the presence of hydrogen.	
$[15]$	Zheng, W. et al		The atomic bonds at the crack's tip weaken due to the accumulation of hydrogen at the crack tip.	
$[17]$	Koyama, M. et al.	HIE	Hydrides first appeared in the stress field of the fracture, and instead of growing from individual hydrides, they generated new hydrides in the stress fields of the other hydrides to reach vast sizes.	
$[18]$	Pérez, F. et al.	HELP	Investigated the HELP mechanism in high-strength steels. \bullet Found that hydrogen-induced softening and localized plasticity contribute to ٠ embrittlement, especially at stress concentrations.	
$[22]$	Martin, M. et al.		Developed a multiscale model to elucidate the HELP mechanism. \bullet Identified hydrogen trapping and dislocation interactions as key factors \bullet contributing to embrittlement.	
$[21]$	Kappes, M .et al.	AIDE	Studied AIDE mechanism in high-strength alloys using atomistic \bullet simulations. Demonstrated that hydrogen-induced dislocation emission significantly \bullet reduces the material's ductility.	
$[5]$	Pradhan, A .et al.	HAM	Used molecular dynamics simulations to investigate the role of hydrogen in \bullet micro-fracture initiation. Found that hydrogen accumulates at grain boundaries, facilitating micro- fracture.	

Table 1 studies related with Mechanisms Causing HE

$Ref.$ #	Author	Preventive Measures	Results and conclusions		
$[40]$	Fan, Y. et al.	Coating selection and design	Investigated the effectiveness of various coatings in preventing hydrogen embrittlement. Found that certain coatings reduce hydrogen diffusion into \bullet the material, mitigating embrittlement.		
$[36]$	Ćwiek, J. et al.	Mechanical design optimization	Investigated the role of mechanical design features, such as notch geometry and stress concentration reduction, in mitigating HE. Recommended design modifications to reduce \bullet embrittlement.		
$[35]$	$Li, X.$ et al.	Modifying Microstructure of Material	altering the material's microstructure by refining the alloy's microstructure and introducing or eliminating the appropriate alloy components		

Table 3. Studies related with Preventive Measures of HE

6. Conclusion

 This work addresses the causes, mechanisms, measurement techniques, and hydrogen-related failures, such as HE, in steel material which is susceptible to hydrogen penetration or diffusion. Preventive measures intended to eliminate or minimize hydrogen penetration in steel are discussed about at the same time. By preventing hydrogen diffusion, these preventative measures may reduce the probability of hydrogen catastrophes. The conclusions drawn from the review of academic literature that gives a general overview of hydrogen embrittlement in steel are as follows.

- When the diffusible hydrogen content above the critical hydrogen concentration, HE occurs in the material, and the degree of HE depends on the hydrogen concentration.
- Steel loses its ductility and toughness as its carbon content rises or its strength levels exceed 1000 MPa, making it more susceptible to fracture even in situations with comparatively little stress.
- The main causes that raise the possibility of hydrogen embrittlement in steel are environmental factors, material susceptibility, and load (stress) conditions.
- There is currently no standard HE mechanism for all materials; all of the current HE mechanisms are only relevant to certain materials in particular applications.
- The most widely used experimental methods for measuring concentrations and analyzing hydrogen trapping in steel are now TDS and GC with TCD.
- There are two ways to prevent HE from occurring. First, surface treatments including coatings and modification treatments are applied. These methods are used to prevent HE from the outside. The second method involves modifying the material's microstructure by optimizing the alloy's

microstructure and introducing or eliminating the appropriate alloy elements.

Steel's HE will drop if the amounts of the elements C, Si, P, and S are decreased or the amounts of Ni, Al, and Mo are increased. Additionally, elements like Mo, V, and Ti typically combine with C to form carbides, which strengthen the alloy's resistance to HE by functioning as long-term hydrogen traps.

Declaration

 There are no possible conflicts of interest that the authors have disclosed about the research, writing, or publication of this article. Additionally, the authors declared that no specific authorization or ethical committee approval was needed for this piece, which was written entirely on its own and in compliance with international publication and research ethics.

Author Contributions

 Biniyam Ayele Abebe: Original draft writing, conceptualization, methodology, and data collection. Ekrem Altuncu: Teaching and supervision, structure – assessment & editing.

Nomenclature

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Review Article

A review on integration of carbon fiber and polymer matrix composites in 3D printing technology

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1. Introduction

In recent years, additive manufacturing (AM) technologies have been playing an important role in the rapid and cost-effective production of components, especially those with complex geometries [1,2]. Innovations in this field support a wide spectrum of uses, especially from prototype production to end-use parts and offer alternative solutions to traditional production methods [3,4]. AM methods such as Fused Deposition Modeling (FDM) can be used on a wide range of materials, from plastics to metals and ceramics, thus offering new opportunities for industrial designers and engineers [5,6].

However, there are some limitations in the use of these methods. Traditional 3D printing materials generally have limited mechanical properties. This situation creates serious limitations, especially in industrial and structural applications that require high strength, hardness, and wear resistance [7,8]. For example, plastic-based filaments used in typical FDM printing fail rapidly under high loads or harsh environmental conditions [9,10]. In order to expand the application areas of such materials by increasing their strength performance, composite materials need to be

further researched and adapted to 3D printing technology [11,12].

Composite materials are hybrid materials created by combining one or more different materials. This combination generally enables the materials to exhibit superior properties than those they have alone [13,14]. The use of composites in 3D printing technologies offers a wider range of industrial applications by increasing the strength, stiffness, and thermal resistance of materials [15- 25]. Composite filaments, especially those containing reinforcements such as carbon fiber, glass fiber, and metallic fillers, are considered to be in the category of materials closest to the potential to provide the durability required for structural components [26,27].

In this context, the use of composite materials stands out as an important strategy to expand the application areas of 3D printing technology and increase the mechanical performance of materials [28,29]. Thus, 3D printing can be used more effectively in many fields, from automotive to aviation, from medical devices to structural engineering, and the acceptance rate of these technologies will increase.

This study focused on the production of composite filaments containing various reinforcement materials

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(carbon fiber, glass fiber, metal oxides, etc.) and their usability in 3D printing processes. Studies on improving the mechanical properties of composite filaments, such as tensile strength, modulus of elasticity, and creep resistance, are mentioned. In addition, filament production techniques, optimization of 3D printing parameters, and the suitability of materials for industrial applications were also examined. The development and characterization of composite filaments have been supported by various analytical methods. These include Differential Scanning Calorimetry (DSC), Scanning Electron Microscopy (SEM), Dispersive X-ray Spectroscopy (EDS), and various mechanical tests. These methods examine in detail the internal structure of composites and the distribution of reinforcement materials in the matrix and determine the effects of the materials on mechanical properties. In conclusion, this study demonstrates the potential of composite materials in 3D printing processes and aims to increase the suitability of these materials for industrial applications. Thus, the production of more durable, functional, and economical 3Dprinted parts will become possible. These developments continue to push the boundaries of 3D printing applications by bringing together innovations in materials science, engineering, and manufacturing technologies.

The aim of this study is to examine the integration of carbon fiber and polymer matrix composites in 3D printing technology and to show how the desired mechanical properties and lightness of these materials can be transferred to 3D printing applications. This has the potential to revolutionize the production of highperformance and durable components and can lead to important applications in various industries (aerospace, automotive, and sports equipment). In addition, the study addresses technical challenges and solutions related to the use of carbon fiber and polymer composites in 3D printing, providing a resource to guide research and development activities in this area. Thus, it supports innovations in materials science and manufacturing technologies by providing new knowledge and ideas for the development and expansion of 3D printing technologies.

2. Literature Surway

In their study, Çanti et al. [30] carried out the production and characterization of composite filaments for 3D printing technology. The aim of the research is to increase the mechanical properties of filament materials used for 3D printers and thus provide a wider range of usage in industrial applications. In the study, Acrylonitrile Butadiene Styrene (ABS) was used as the main matrix material, and composite filaments were produced by adding various micro and nano-sized particles (MWCNTs, $SiO₂, ZrB₂, Al)$ to this matrix. The production process was carried out using a twin screw extruder, as shown

schematically in Figure 1, and the resulting filaments were characterized by methods such as DSC, SEM, EDS, tensile test, and surface roughness tests. Research results showed that micro and nanoparticles added to the ABS matrix significantly improved the Breaking Stress (UTS) and tensile strain of the composites. In particular, ZrB_2 and Al particles reinforced with microparticles significantly improved the mechanical strength of the material, increasing the tensile strain by 17.8% and 40%, respectively. Additionally, filaments reinforced with nanoparticles showed good performance overall, although they experienced local condensation and internal void problems. These findings show that composite filaments can be used in commercial FDM devices and that these new-generation materials can enable the production of more durable and functional 3D-printed parts thanks to their improved mechanical properties. As a result, this study can be considered as a step towards increasing the performance of the materials used in 3D printing technology and making them more suitable for industrial applications.

In the studies of Sezer et al. [31], the usability of carbon fiber-reinforced ABS composite filaments in the 3D printing process was investigated. For use in FDM 3D printers, carbon fibers were mixed with ABS granules using a twin-screw extruder and then formed into filaments with a single-screw extruder (Figure 2). In the research, the effects on the mechanical properties of the filaments obtained by adding 6 mm long carbon fibers to the ABS matrix were examined. Moreover, the printing pattern was found to have a significant effect on the mechanical properties. The filaments were used to produce standard ASTM D412 A tensile test patterns containing various weight percentages of carbon fiber. Tensile test results showed that as the carbon fiber ratio increased, the breaking strength of the parts increased, but ductility and toughness decreased. As a result, it has been found that extrusion temperature and speed have a significant impact on filament quality, and carbon fiber reinforcement, especially at low rates, significantly improves the mechanical properties of ABS. However, it has been observed that high carbon fiber ratios have negative effects on mechanical properties by increasing porosity. This study makes a significant contribution to the development of 3D printing materials and reveals the potential advantages of using FDM technology in industrial applications.

In the studies of Çelik and Gür [32], the effects of the parameters used in the production of ABS and carbon fiber-reinforced ABS composites with a 3D printer on the mechanical properties were examined.

Figure 1. The production steps of nano-micro polymer composite filaments [30]

Figure 2. Composite filament preparation and 3D printing [31]

Figure 3. Tensile test sample dimensions in ISO-527-2-type-1A standard [32]

The work includes the production and testing of tensile test samples in accordance with the ISO 527-2 Type-1A standard (Figure 3) using various printing directions and filling angles. The results showed that regular ABS exhibited better mechanical properties than carbon fiberreinforced ABS. Tensile test results revealed that printing parameters, especially the way the samples were placed on the production table and the internal structure filling angles, had a significant effect on the mechanical properties. Significant differences in mechanical properties were observed between samples with different printing directions and internal structure filling angles. The study emphasizes that optimizing 3D printing parameters is important, especially in terms of strengthening interlayer connections and reducing material defects. These optimizations can increase the suitability of 3D-printed parts for industrial applications and enable the production of parts with a wide range of uses.

In the studies of Urtekin et al. [33], the mechanical properties of polyester resin matrix and unidirectional

carbon fiber reinforced composite materials produced by the hand-laying method at room temperature were examined. The study was carried out by placing unidirectional carbon fibers in one to three layers into a polyester matrix and then applying various mechanical tests (tensile, bending, and low-speed impact tests). According to the tensile test results, two-layer composites reached the highest tensile strength values. In bending tests, three-layer composite materials have the highest elasticity values when the fiber direction angle is 0°. In impact tests, single-layer composites showed the highest deformation values. These results show that the number of layers has a significant effect on the mechanical properties of composite materials. The higher tensile strength of twolayer composites emphasizes the effect of layers on mechanical performance. In addition, the high elasticity values of three-layer composites in bending tests indicate the potential use of these structures in applications requiring high flexibility. As a result, this study reveals in detail how the methods and materials used in the production of unidirectional carbon fiber-reinforced polyester composite materials with different numbers of layers affect the mechanical properties of the final product. This information can be taken into account when designing and manufacturing composite materials, especially in sectors such as aerospace, automotive, and defense industries.

In the work of Güneş and Çayıroğlu [34], the mechanical behavior of 3D printed parts with continuous steel wire reinforcement was examined (Figure 4). In the research, production with steel wire-reinforced polymer nylon material provided an approximately 5.58 times higher strength increase compared to unreinforced polymer nylon material. This result shows that continuous steel wire reinforcement can significantly increase the strength of 3D-printed parts. Additionally, the effects of steel wire reinforcement and different printing patterns on strength were also investigated. The strength performances of the parts produced using different printing patterns were compared, and the patterns with the highest strength values were determined. This study provides important information about innovations and potential applications in steel wire-reinforced 3D printing technology. In particular, the potential of using this technology in the production of parts requiring high strength in industrial applications and engineering fields is emphasized.

In the studies of Nergün et al. [35], the infrared heating method was used to increase the mechanical properties of continuous carbon fiber-reinforced thermoplastic composites (CFRTP). Thermoplastic filaments were produced using polyamide and continuous carbon fibers, and the aim was to increase the mechanical properties with an infrared heat source during the printing process, as seen in Figure 5.

Figure 4. The extruder mechanism developed for composite printing [34]

Figure 5. Schematic image of the infrared heater [35]

In experiments using three-point bending tests, significant increases in bending modulus and bending strength were observed when infrared heaters were used at low printing speeds. The highest flexural strength was measured as ~420 MPa, and the highest flexural modulus was measured as ~52 GPa. These results show that infrared heating is an effective method to improve the mechanical properties of CFRTP composites. By using infrared heating at low printing speeds, it was possible to increase interlayer bonding and thus improve mechanical properties. These findings increase the potential for CFRTP use, especially in industries such as aerospace, automotive, and medical.

In the study of Kurban et al. [36], the use of carbonbased filament yarns in different forms in the design of textile-reinforced concrete structures was examined. The study includes textile components used in the form of raw yarn and hybrid yarn coated with various polymers. Particular emphasis was placed on the knitting technique, which is one of the many hybrid yarn production methods used in the textile industry. In the research, samples were produced by placing two different textile surfaces produced using three different yarn structures into concrete in three different positions. Compared to the use of raw filament, a 23% increase in bending strength was observed with the use of hybrid yarn and a 167% increase

in bending strength with the use of epoxy-coated filament. This study provides important information for more effective reinforcement of carbon fiber-reinforced concrete structures and provides methods on how to improve the mechanical properties of textile-reinforced concrete using epoxy coating or hybrid yarn technologies. These findings have important implications, especially for structural applications requiring high strength, and have the potential for the development of textile-reinforced concrete technologies.

In Tanabi's study [37], the temperature effect on the mechanical properties of composite materials produced through 3D printing was examined. The research covers short glass fiber reinforced polyamide 6 (GFPA6) composites and unreinforced ABS polymer at various temperatures (from -20 \degree C to 60 \degree C). The experiments were carried out on samples produced according to the ASTM D638 standard, and the samples were subjected to tensile loading at various temperatures. As a result of the tensile tests, it was determined that the GFPA6 material showed up to 56% higher hardness and up to 59% higher strength than ABS, as seen in Figure 6. It has been observed that as the temperature increases, both materials undergo significant deformation, and their tensile strength decreases. Additionally, by microscopic analysis of the fracture surfaces, fiber extrusion was determined to be the dominant fracture mechanism for GFPA6 and filament fracture for ABS. Research results indicate the potential for the use of these materials in various industries, such as aerospace and automotive, and contribute to the understanding of their structural performance, especially under different temperature conditions.

In the research of Seçgin et al. [38], optimization of the surface roughness of parts produced using carbon fiber filament with 3D printers was discussed. In the research, the effects of different printing parameters (nozzle temperature, layer thickness, filling ratio) on surface roughness were examined. Signal/Noise ratio analysis was performed using the Taguchi methodology, and the parameters that provided the best surface quality were determined. According to the results of the analysis, the optimum levels for nozzle temperature, layer thickness, and filling ratio were determined to be 240°C, 0.1 mm, and 20%, respectively. The study provides valuable information on how material surface quality can be improved in 3D printing processes and highlights the importance of optimizing printing parameters when using special filaments such as carbon fiber. These findings may be particularly useful in industrial applications and production requiring high quality.

In Yıldız's study [39], the creep behavior of carbon fiber-reinforced polylactic acid (PLA) samples produced using 3D printing technology was examined. Samples were prepared by adding carbon fiber to PLA polymer at

Figure 6. Stress vs strain diagram of a) ABS and b) GFPA6 [37]

different rates (0%, 15%, 20%), and experiments were carried out at various filling rates (70%, 80%, 90%, 100%). Creep tests were carried out specifically at 37 °C and under loads ranging from 20-80 Newtons. Study results indicate lower "exponentially related creep" rates in 15% and 20% carbon fiber reinforced samples compared to unreinforced samples. This shows that 15% of CF-reinforced composites are more stable and perform well. Additionally, the fracture areas of the samples were examined in detail with SEM analysis, and changes in the material structure and damage mechanisms were observed. These findings indicate that the addition of carbon fiber to the PLA matrix improves the mechanical properties of the material by affecting the creep resistance, thereby increasing the suitability of these composites for industrial applications.

In the study of Taleb [40], the production and mechanical tests of continuous wire-reinforced PLA matrix filament for 3D printers were carried out. It is aimed to increase the mechanical properties of the filaments used in 3D printers and thus improve the durability of the printed parts. In the study, continuous wire-reinforced composite PLA filament was produced using a single screw extruder and a designed mold. Tensile test samples were printed using the produced filament, and their

mechanical strength was determined. Experimental results revealed that samples printed with composite filament showed significantly higher strengths compared to samples printed with pure PLA filament. This shows that continuous wire reinforcement can significantly improve the mechanical performance of PLA matrix composites. The results indicate that such composite filaments can expand the scope of 3D printing applications, especially by using them in the production of structural components.

In the work of Çelik et al. [41], a composite filament extruder machine was developed for 3D printers. The aim is to evaluate whether these filaments meet industrial requirements by preparing functional composite test samples. The study focuses on various composite materials and covers the design and manufacturing of an extrusion system to ensure the processability of these materials in the filament geometric structure by the molten filament fabrication method. Thanks to technological developments, AM techniques have become popular and have an important place in industrial production. These innovations have encouraged various studies to expand the use of AM, especially in order to save time and cost in production processes. The study focused on the processability of materials such as polymer-matrix composite and carbon fiber by extrusion method, their mechanical properties, and their suitability for the AM process. As a result, by producing composite filaments with the designed extruder machine, it has been shown that these filaments can be used for 3D printers and comply with industrial standards.

Usun et al. [42] studies focus on the CFRTP 3D printing process and investigate its potential to be used as an alternative production method, especially in sectors such as aviation, automotive, prototyping, medical applications, and space industry. Using the FDM method, the effects of printing parameters (such as nozzle temperature, printing speed, layer thickness, and heated bed temperatures) on the mechanical performance of CFRTPs were examined. CFRTP filaments were obtained using PLA and carbon fiber, based on the melt impregnation technique, and threepoint bending test samples were printed using these filaments. Test results showed 23% fiber proportion with flexural strength ranging from 108 to 224 MPa and flexural modulus ranging from 9.67 to 17.69 GPa. In conclusion, this study shows that the use of CFRTPs in 3D printing processes can be effective, especially in applications requiring high mechanical properties, and its use can become widespread in various industrial sectors.

In the studies of Yeşil and Asi [43], the water absorption properties of glass and carbon fiber-reinforced polyester and vinyl ester pultruded hybrid composite profiles were examined. The main aim of the study is to determine the water absorption properties of composites produced using various fiber contents and resin types to predict their longterm performance. The extrusion process is the process of producing composite materials with constant crosssection. In this process, fibers are immersed in a pool of resin and pulled through a heated die with the crosssectional geometry of the specific profile, allowing the composite materials to cure. In the research, four different I-section beam configurations were produced using glass fiber, carbon fiber, glass fiber continuous strand mat, and carbon woven roving mat, and polyester and vinyl ester resins were preferred as a matrix. Water absorption tests were carried out by immersing the composite samples in water according to the ASTM D 570 standard. Fiber contents were determined by the calcination method according to TS 1177 EN ISO 1172 standard. The results obtained were analyzed using the data obtained from the experiments. Glass-carbon fiber-reinforced polyester matrix composite materials tend to absorb more water than other samples, and glass-fiber-reinforced polyester matrix composite materials have the lowest water absorption percentage. This is due to the void content of the polyester matrix. It has been observed that the percentage of water absorption increases as the carbon fiber ratio increases. This situation was caused by voids formed due to poor penetration of polyester resin into carbon fibers. Glasscarbon fiber reinforced vinyl ester matrix composite materials have the lowest water absorption percentage values. Because vinyl ester has better penetration capacity than polyester, and the void content of vinyl ester composites is lower than polyester composites. For this

reason, vinyl ester resin composites are preferred, especially in marine applications. These findings show that water absorption can significantly affect the long-term behavior of composite materials and that fiber content and matrix type play a determining role in these properties. Such information is critical in determining the design and application areas of composite materials.

In the study by Yaman et al. detailed in the reference [44], carbon fibers are explored for their distinct applications and manufacturing techniques stemming from a variety of precursor materials including polyacrylonitrile (PAN), rayon, and various other organic and inorganic precursors. The results and properties of the carbon fibers significantly depend on the choice of raw materials and the processing conditions applied. Different carbon fibers exhibit varying degrees of modulus and strength, which categorizes them into types like ultra-high modulus, high modulus, intermediate modulus, and low modulus, each having specific applications ranging from aerospace to medical technologies. Significant numerical results include the production efficiency and the fiber modulus and strength. For instance, PAN-based carbon fibers, which make up about 90% of the carbon fiber market, have a yield of around 40-45% from the precursor, with the fibers undergoing intense processing conditions including stabilization, carbonization, and graphitization. These fibers achieve a modulus in the range of 300-500 GPa and tensile strength around 3 GPa.

In the studies of Yaman et al. [45], the physical and chemical properties of carbon fibers, their production methods, and their use in various industrial applications were discussed. In summary, the high strength, low weight, and high electrical and thermal conductivity properties of carbon fibers are emphasized. Additionally, the production processes of carbon fibers from various raw materials (PAN, rayon, tar, pitch) and the effects of these processes on fiber properties are detailed. The application areas of carbon fibers cover a wide range. It has been stated that it is used effectively in areas such as aviation, space, automotive, sports equipment, construction, and medicine. In addition, it has been emphasized that carbon fibers are a preferred material in a wide variety of industrial applications thanks to their properties such as high fatigue resistance, corrosion resistance, and chemical inertness. The study also provides information on innovations in carbon fiber production technologies and how these innovations affect the cost, and it is predicted that carbon fibers may have wider usage areas in the future. This information reveals the future potential of carbon fibers.

In the study of Kaygısız et al. [46], they examine the mechanical properties of materials printed on a 3D printer using carbon fiber-reinforced filaments. The research indicates that fiber-reinforced composite materials have a wide range of uses due to their properties, such as high strength, hardness, and conductivity. These materials are especially used in areas such as aircraft structural elements, wind turbine blades, automotive exterior panels, and computer enclosures. In the study, carbon fiber-filled nylon filament (ePA-CF) and PLA+ filament materials produced by ESUN were used. These filaments were printed with a 3D printer, and the mechanical properties of the samples were comparatively examined. It has been observed that the tensile strength of the material increases significantly, especially when carbon fiber-reinforced nylon filament ePA-CF is used. The results of the research point to important opportunities for 3D printing technologies to be used more effectively, especially in industrial applications and in the production of products requiring high performance.

In the studies of Öztürk et al. [47], the production of continuous fiber-reinforced composite filaments and the printability of these filaments on a 3D printer were examined. In particular, 3D printer technology has been used for the production of complex parts and offers lower cost, faster, and easier production advantages compared to traditional production methods. However, the parts produced with 3D printer technology cannot be used as final products due to their low strength properties, requiring improvement of the material. In the study, high-

strength continuous fiber-reinforced composite filament was produced with an extruder device using carbon fiber yarn and PLA. The mold was designed and produced by coating the continuous fiber with thermoplastic material. By using the FDM method, these filaments were produced and are suitable for use in 3D printers, and parts were printed in the specified dimensions. As a result, the production of continuous fiber-reinforced composite filaments was successful, but some technical difficulties and problems with material properties were identified. This study provides important information for increasing material strength in the production of complex parts with 3D printer technology and forms the basis for future research in this field.

In the studies of Hu et al. [48], the production of continuous carbon fiber (CCF) reinforced prepreg filament and the use of 3D printing were discussed. An innovative method for printing continuous carbon fiber composite parts using FDM technology is proposed. Within the scope of the research, first, CCF prepreg filament was produced, and then the extruder heads of 3D printers were modified to print this filament. The mechanical properties of the produced composite parts were examined using a threepoint bending test and Surface Response Methodology. The results showed significant improvements in the flexural strength and modulus of composites produced using certain printing parameters. Layer thickness was determined as the parameter that makes the biggest contribution to the final bending strength. These findings increase the potential of 3D printing, especially in industries such as aerospace and automotive that require materials that are lightweight and have high mechanical performance. Additionally, the study developed a detailed mathematical model for the production of CCF-reinforced composite parts by 3D printing and proved this model with validation tests.

Gahletia et al. [49] work deals with the process of reinforcing micro carbon fiber filled filament in nylon with Kevlar, Fiberglass, and HSHT Fiberglass with various layer thicknesses, reinforcement types, and filling patterns. The study suggests that the optimal composition of these components can be used to produce strong, high-quality parts, especially for use in fields such as the aerospace and automotive industries. The research states that properties such as tensile strength, wear resistance, and surface roughness were examined during the production of these composites under varying process parameters. The study stated that these properties were tested in samples in accordance with various standards (ASTM et al. IV and ASTM G99). It is described that the response surface methodology and the central composite design approach are used as well, and the MOGA-ANN approach is used to

optimize multiple response targets. The study results showed that maximum tensile strength, as well as minimum surface roughness and wear rate, were achieved by using a certain fiber layer thickness and filling pattern. These findings support the potential use of these material combinations in industrial applications and provide a basis for future research.

In the work of Xin et al. [50], the fusion bonding performances of composites reinforced with short and continuous carbon fibers were examined using fusion filament fabrication. The study used short beam shear and in-plane tensile shear tests to evaluate the effects of short fiber quantity on filament bonding properties. The results revealed that with the increase of short fiber content, there was first an increase and then a decreasing trend in the interlaminar shear strength (ILSS) and in-plane shear strength. In particular, samples containing 5% short fibers showed a significant increase in their load-carrying capacity and fusion bonding performance, outperforming the sample containing 0% fibers, with an increase of approximately 41% and 80% in ILSS and in-plane shear strength, respectively. When the damaged surfaces were examined, fiber extrusion and resin breakage were determined as the dominant bond failure models. It was found that the use of the appropriate amount of short fibers effectively improved the fusion bonding of S-CFRPCs. These results seem promising for the future design of strong and durable composite structures.

Almeshari et al. [51] examine the development process of polypropylene composite filaments reinforced with short carbon fibers for 3D printing. The research proposes a method in which short carbon fiber (SCF) content is mixed with polypropylene (PP) granules in different proportions (4% to 22% by weight) and processed in a twin-screw extruder and then made into filaments using a single-screw extruder. The produced filaments were evaluated in terms of their mechanical, physical, and morphological properties. The findings specifically indicate that the composite containing 22% SCF showed a 150% improvement in tensile strength and a 260% improvement in impact energy compared to pure PP. However, the values in the fracture time of the composites showed a linear decrease of up to 11% SCF content, after which there was a sharper decrease. Research suggests that SCF/PP composites are suitable for 3D printing applications and can be used especially in applications requiring high performance.

When the literature is evaluated, the prominent mechanical properties can be categorized when matrix material types and reinforcement elements are used, as in Table 1.

3. Discussion and Estimation

This review article examines in depth the use of composite filaments in 3D printing technology, revealing the potential of these materials in industrial applications. The results of the studies show that various reinforcements are effective in improving the mechanical properties of composite filaments. These improvements have the potential to reduce costs and speed up production processes by diversifying material selection, especially in applications requiring high performance. The discussion is categorized under the following subheadings.

3.1 Effects on Mechanical Properties

With the addition of reinforcement materials, significant improvements were achieved, especially in tensile strength

and bending strength. However, reducing effects of these reinforcements on other mechanical properties, such as material flexibility and ductility, have also been observed. For example, while carbon fiber reinforcement increased the strength of ABS and PLA matrices, it reduced the machinability and flexibility of the materials. This highlights that material choices must be carefully evaluated against application specifications. In Table 2, the effects of matrix materials and reinforcement elements on mechanical properties according to their type are evaluated.

3.2 Effects of Environmental and Operational Conditions

Studies in the literature have also tested the durability of composite materials against environmental and operational factors. In particular, temperature changes and water absorption have emerged as determining factors in material performance. The low water absorption of vinyl ester and carbon fiber reinforced composites provides advantages in applications that come into contact with water, such as marine. Against temperature changes, materials such as GFPA6 have shown that they are suitable for use in the automotive and aerospace sectors by offering hightemperature resistance. In Table 3, the affected properties of the composite material components are categorized according to their operational properties.

3.3 Sustainable Production Technologies

3D printing technology offers great opportunities for customized production and waste minimization. However,

innovative methods need to be developed to reduce the environmental impact of this technology and increase energy efficiency. Future research may develop more efficient 3D printing techniques and printing processes that optimize energy use.

3.4 Long-term Performance and Reliability Tests

Detailed studies designed to evaluate the long-term performance and reliability of composite filaments can be conducted. This is especially critical for industries such as aerospace and automotive that require structural integrity and durability. Aging processes of materials, long-term fatigue tests, and exposure to environmental factors can be examined.

3.5 Innovative Design and Manufacturing Approaches

The discovery of new material combinations and hybrid structures could revolutionize 3D printing technology. Future research may focus on multi-material printing techniques and hybrid printing systems, which may allow the production of more complex and functional parts. Additionally, Artificial Intelligence (AI)-)-AI-supported design optimization and simulation-based engineering applications can contribute to further improvement of materials and manufacturing processes. In Table 4, the aims of the research themes and suggestions/approaches are categorized.

Material Type	Conditions	Affected Features	Effect Type	Explanation	Ref.
GFPA6, ABS	Temperature changes $(-20$ °C to 60° C)	Strength, deformation	Strength decreases, deformation increases	High temperatures weaken the structure of the material and increase deformation.	$[37]$
Vinylester, Polyester	water absorption	Water absorption rate	Decrease or increase	Vinylester shows lower water absorption because it is more resistant to water. Polyester shows higher absorption.	[43]
Glass and carbon fiber reinforced polyester/vinyl ester	Water absorption continuous and exposure	long term performance	Performance decreases	Water absorption can reduce the structural integrity and mechanical properties of the material over time.	[44]
PLA	Temperature $(37 \degree C)$ and load $(20-80$ Newtons)	Creep resistance	Improvement	The creep resistance of PLA at high temperatures increases with carbon fiber reinforcement.	[39]
Various composites	High carbon fiber ratio and porosity	Mechanical properties	in Decrease features	A high carbon fiber ratio may have a negative impact on mechanical properties by increasing porosity.	$[31]$

Table 3. Properties of the material type affected by operational properties

Table 4. Purposes of research themes and suggestions

4. Potential Future Applications

The integration of 3D printing technology and composite materials has the potential to revolutionize various industries by enabling customized production, high-performance materials, smart structures, environmentally friendly manufacturing, and rapid prototyping. In the medical and healthcare sectors, 3D printing can produce tailored prosthetics, orthotics, and implants that precisely fit individual anatomies. Highperformance materials, such as those used in the aerospace and automotive industries, will benefit from lighter, stronger, and more durable composites, enhancing fuel efficiency and reducing carbon emissions. The development of smart materials

integrated with sensors will allow for the creation of intelligent structures capable of monitoring their condition and responding to environmental changes. Furthermore, 3D printing offers more sustainable production methods by reducing waste and improving energy efficiency, providing economic and environmental benefits. The rapid and cost-effective production of complex geometric structures through 3D printing accelerates innovation in engineering and design. These advancements, at the intersection of materials science, engineering, and manufacturing technologies, hold promise for substantial improvements in efficiency, performance, and sustainability across various sectors.

5. Conclusions

This review study focuses on the production of various composite filaments and their use with 3D printing technology. Research has involved the reinforcement of ABS, PLA, polyamide, polyester resin, and various other polymers with different reinforcement materials such as carbon fiber, glass fiber, metal particles, and continuous wire. The findings can be listed as follows;

- It has been shown that reinforcement elements significantly improve the mechanical properties of cocomposite materials. In particular, the improvements observed in critical mechanical properties such as tensile strength, bending strength, and creep resistance enable a wider range of industrial applications for these materials.
- GFPA6 and glass fiber-reinforced polyamide samples, which exhibit superior performance against temperature effects, offer potential application opportunities, especially in the aviation and automotive industries. In terms of water absorption properties, vinyl ester matrix composites stand out as ideal options, especially for marine applications.
- Studies have shown that optimization of 3D printing parameters can significantly affect material surface quality and mechanical integration. This provides strategic information on customizing and processing filaments, especially for industrial applications requiring high performance.

As a result, these studies provide a guide to what role composite filaments and 3D printing technology can play in industrial applications and what methods can be followed to maximize the performance of these materials. Future studies can build on these findings and focus on the development of 3D printing materials that are more durable, functional, and resistant to environmental factors. This is especially critical for the discovery of economical and efficient production methodologies that can replace high-cost materials.

Declaration

The authors declared no potential conflicts of interest with respect to the research, authorship, and/or publication of this article. The authors also declared that this article is original and was prepared in accordance with international publication and research ethics, and ethical committee permission or any special permission is not required.

Author Contributions

A. Kaptan and F. Kartal jointly supervised and improved the study, co-wrote the text, and proofread the manuscript.

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Review Article

Merger of internet of things and machine learning: The internet of everything sector projects, benefits, and future roles

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1. Introduction

The Internet of Things (IoT) is the network of physical objects such as sensor devices, appliances, vehicles, buildings, and other elements embedded with electronics and circuits. Every IoT device has software and network connectivity, enabling it to act as an object that exchanges data with other objects; hence, it is the definition of Machine-to-Machine (M2M) communication. Thus, IoT enables remote sensing and control of objects, opening up the possibility of rendering the physical world in computer-based systems. The trend towards greater IoT connectivity means collecting real-time data from more places, leading to real-time decisions and increased revenue, productivity, and efficiency [1].

The booming of smart devices due to miniaturization and the advances in technologies that connect everything from anywhere at any time has digitalized the world of human beings. The constant communication of humans with each other, as well as sensors and actuators, has brought a revolution focused on interconnection, automation, autonomy, and real-time data. Data grew so large that the name Big Data was given to describe the

huge data from which useful additional information could be extracted. Artificial Intelligence (AI) and its child Machine Learning (ML) came to the rescue in data analysis to provide early warnings, forecasts, decision support systems, and even alarm generation. Hence, the IoT and ML collaboration opened the avenue for the Internet of Everything (IoE). IoE is beyond IoT as it involves people, data, things, and processes. In IoE, the data, processes, people, and things interact, thus facilitating transactions and information flow optimization for value creation. It is crucial to remember that ML is an AI discipline that learns from experiences and paradigms to augment IoT in delivering greater wisdom to humans. The number of devices and humans interconnected continues to increase incredibly, following the technology merger trend [2].

As shown in Figure 1, the latest IoT Analytics report shows that the number of global IoT connections grew by 18% in 2022 to 14.3 billion active IoT endpoints. In 2023, IoT Analytics expects the global number of connected IoT devices to grow another 16% to 16.7 billion active endpoints.

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Figure 1. Global IoT devices forecast until 2027 [2].

While 2023 growth is forecasted to be slightly lower than in 2022, IoT device connections are expected to grow for many years. Propelled by the staggering number of IoT devices, M2M provides manufacturing companies with many new application opportunities. Hence, the progress in consumer applications improves our everyday lives. The remaining sections of our study are organized as follows:

- Section 2 discusses previous studies on the subject according to sectors.
- Section 3 provides information about the joint projects in which the three are involved by sector.
- Section 4 explains the results and discusses the future roles of joint IoT and ML projects that shape the IoE environment.
- In Section 5, we conclude.

With the above contents, our work attempts to make the following contributions:

- Provide a review and comparison of joint state-ofthe-art IoT and ML research and applications that form the IoE environment.
- Focus on the collaboration, advantages, and future of IoT and ML technologies towards IoE in sectoral applications,
- Offer recommendations on how IoT and ML technologies towards IoE can be made more effective and efficient,
- Provide a perspective on future trends and their long-term impact on sectors.

The opportunities are shown by prominent examples of projects in important sectors and by highlighting the development and progress in the related fields. Briefly, our study aims to stimulate new studies by assisting researchers in identifying the effects of the merger between IoT and ML and, thus, the new IoE environment.

2. Related Background In Different Sectors

Figure 2 shows the map that embraces the IoT and different sectors [3]. The comprehensive map by Breecham Research divides the sectors into their scientific fields of expertise. Each sector's Application groups are given, and the application types of every group are listed. Finally, the devices and their use locations are listed at the outer rim of the circular map. Some of the most important sectors are discussed in detail below.

Figure 2 demonstrates that IoT technologies contribute to environmental sustainability efforts, energy optimization, efficiency in industrial production, and reduction of all sorts of consumptions. In the industrial sector, the IoT-ML partnership helps anticipate customer needs by analyzing large amounts of data. The same collaboration helps manage production planning and perform efficient supply chain management. Below, further typical examples are given in detail in some selected sectors.

Investigating research studies on joint IoT and ML initiatives forming the IoE reveals that their association or collaboration is overlooked in many works. Singular studies on IoT or ML only curb researchers' ability to identify research gaps. The research gap is depicted when sensor networks' live data from the environment is not augmented by ML-enabled smart machines that can perform diverse tasks. For example, collaborative work can easily help to detect insect damage in agriculture or help with irrigation planning [4].

In the manufacturing sector, it is possible to track the health of machines and the state of stored products using IoT devices and ML. When used together, it is possible to detect anomalies in machines and products [4]. Machines are now smart devices that can sense the environment and make decisions based on previous data, thus forming the IoE environment. Following production, products are inspected using IoT tags and tracked via cameras or sensors. The collected data of each product is fed to an ML or another AI set-up. Analyzing live data from machines through a well-designed ML algorithm can be used to monitor, detect potential production problems, and make the right corrections instantly.

Smart algorithms created using ML can reduce power consumption in the energy sector. Accurately tracking objects and applying computer vision and sensor networks with intelligent and well-chosen AI algorithms can also achieve quality control and assurance. Using live data across individual energy application groups(Figure 2) supported by an ML approach, a highly adaptable IoT-ML design is possible as an IoE example. AI-IoT integration produces supportive decision-making, predictive maintenance, and personalized user experiences, thus forming the IoE environment [4].

The impact of 4G and 5G communication on the IoT-ML merger is worth mentioning. 5G networks' faster and more reliable connectivity provides instantaneous IoT data to MLbased analyses, thus improving their performances. In an end-to-end IoT solution, continuous and instantaneous data from hundreds of sensors creates Big Data, the term of today's popular definition. By applying deep analysis through AI-based software, Big Data is transformed into new

Figure 2. M2M and IoT Sector map according to Application Groups, location, and devices, by Beecham Research [3].

information that provides powerful new insight and intelligence. Considering the automotive industry, automotive connectivity to wearable body sensors, and autonomous cars, the magnitude of Big Data and the advantages gained from its AI analyses by the consumers and producers can be imagined.

The new hype of Digital Twins further magnifies the Big Data and its analyses. A Digital Twin replicates a physical model for remote monitoring, display, and control as a virtual representation on a computer. It is a real-time model of its physical system that adapts to changes. Based on the real-time data from various IoT sensors, a digital twin helps generate predictions using AI and ML [5]. Creating a Digital Twin in an IoE environment involves a lot of work. On topof making the virtual replica of the physical object, all the necessary sensors and actuators for tracking and controlling the physical object are needed. The virtual twin has to be prepared and

trained for the new, real-time, instantaneous data to predict the possible feature outcomes related to the physical object. A Digital Twin can be created with a heart, car, production process, or whole system. In other words, Digital Twins will appear in the Health, Manufacturing, Agriculture, Energy, Transportation, and Construction Sectors.

2.1 Health and Life Science Sector

The healthcare sector is a data-rich environment. More data from more sources means better ML results in healthcare. Main medical IoT activities are:

- Creating automatic medical records,
- Predicting disease diagnoses,
- Monitoring patients in real-time,
- Dispensing medicine.

The reasons for the rapid adoption of IoT and MLtowards an IoE environment in the healthcare sector are as follows.

Figure 3. The architecture of a typical joint ML and IoT model, as a IoE-based healthcare application [6].

A typical IoE environment in healthcare is shown in Figure 3. Sensors are widely used in hospital applications to collect data from the environment. For example, medical sensors can monitor a patient's heart rate, oximetry, pressure, glucose, temperature, ECG, EEG, EMG, etc. In some set-ups, a Raspberry Pi board harbors blood pressure, temperature, and heart rate sensors. The accelerometer, gyroscope, proximity, GPS, and smartphone camera sensors are also used in healthcare applications. Radiofrequency identification (RFID), chemical and Infrared (IR) sensors, and wearable IoT devices are the other tools used in expert hospital systems [6]. Additionally, custom-made software organizes sensor data recording, storing to Cloud Computing Servers, and analysis of the stored information to determine health abnormalities. For example, continuous patient monitoring using a Pulse Sensor can predict the presence of a coronary disease. An example work on the subject uses the Support Vector Machine (SVM), Naive Bayes, Decision Tree, K-NN, and Random Forest algorithms [7]. Another example is the monitoring of patients by sensors placed in the room. Furthermore, wearable fall or bedside carpet sensors can detect a patient's fall and issue an alarm for assistance [8, 9].

Hence, IoT has enabled professionals to connect with patients proactively. For example, RFID-based systems help professionals to apply the right dose to the right patient at the right time [10, 11]. The proposed system integrated base stations to remotely regulate patients' pulse and body temperature. The patient data is also transferred to the doctor's phone [11]. Expert IoT systems can send SMS to patient relatives and medical experts during emergencies.

ML comes into play for formal evaluation of the patient's health status. For example, ML is used in various cancer classification applications to diagnose cancer types correctly. ML algorithms, widely used in healthcare applications, are also applied to many clinical decision support systems to create advanced ML models. ML finds further application in medical imaging in the processes used for creating body-part images to help with treatment and diagnosis. ML is also applied to studying environmental and physiological factors to diagnose diseases effectively. In other words, ML is used to identify signs, symptoms, and risk factors associated with a particular disease to increase diagnostic efficiency and accuracy. The most popular supervised ML algorithms used in healthcare are the K-Nearest Neighbor, Naive Bayes, Decision Trees, SVM, Neural Networks, Gradient Boosting, Regression Tree, and Random Forest [11].

In the health and life science sector, ambulance services can be improved by tracking personnel and assets in the ambulance with automatic data collection technology. The technique includes automatic tracking of mobile assetsin the ambulance with RF technology supported by a database [12]. In an ambulance automation system, an IoE control

center can organize the hospitals in a smart city to arrange the correct hospital with the appropriate capacity tosend the nearest available ambulance to an emergency[8].

"Networked contact lenses" are also a type of wearable device in health technology. In 2014, Google and Novartis began developing a connected contact lens to monitor blood sugar levels by analyzing a patient's tear fluid [8].

Further examples in the literature that synergize innovation in healthcare are listed in work [8], as below:

- The EU I-PROGNOSIS project is a smartphone application that enables early detection of Parkinson's disease.
- Projects in the treatment of acutely ill patients.
- The "Artificial Intelligence in Pathology" project (November 2018 – October 2020) is an AI tool that supports diagnosing and treating colon cancer.
- A computerized project to detect skin tumors, 136 accurately diagnosed cases.
- A project in Boston, an autonomously operating AI robotic catheter in humans and animals.
- Optical touch sensors and image processing algorithms for determining the exact location in the human body.
- AI projects to diagnose and treat psychological problems at an early stage.
- An AI model that can recognize depressive changes based on speech patterns.
- Automated medical chatbots, emotion recognition, cognitive and baby health monitoring systems [13].
- Computer-based wheezing sound detection model for improved diagnosis of respiratory disorders [14]

Many other studies have been carried out on the diagnosis of heart diseases, nutritional monitoring, clinical decision support systems, dementia monitoring, diabetes-related eye problems, children with Autism Spectrum Disorder, etc. The long list has helped the term Internet of Medical Things (IoMT) emerge from networking sensor-based medical devices to improve healthcare services. In summary, the background of IoMT includes a significant amount of IoT research and ML work in healthcare that makes the healthcare IoE environment.

2.2 Manufacturing Sector

Figure 4 includes various use cases for smart manufacturing 4.0, such as Digital Twins, quality sensing, augmented workforce, etc. [15].

IoT and ML towards IoE are used together to automate the production process, optimize products, reduce cost and energy waste, and provide useful information by analyzing data collected from different aspects of the manufacturing business, including production equipment.

In the manufacturing sector, joint IoT and ML projects provide important transformation by combining the power of

automation and data analysis. Today, machines connect and collaborate to make better decisions without human intervention. AI improves and automates decision-making in complex manufacturing environments, increasing production efficiency and reducing human error. Combining these technologies enables the creation of smarter and more efficient production processes in the manufacturing sector. Therefore, ML is one of the key elements of smart manufacturing. ML algorithms are used in many areas, such as making predictions by analyzing data flow in production processes, increasing quality control, predicting failures, and optimizing maintenance processes. These technologies give manufacturers a competitive advantage by enabling more efficient and flexible production processes and effectively utilizing the abundance of data in their production facilities. Example applications include monitoring, fault prediction, and bearing models in electricalmachines [16].

Advancements in IoE technology are promising improved sustainable smart production. Some examples of the benefits provided to smart production by big data and applications are as follows. For example, fault detection is a critical component of predictive maintenance. ML helps to detect errors at a very early stage. Hence, ML applications reduce maintenance costs, repair stoppages, machine failure, and inventory, increasing spare parts life span, operator safety, production,and overall profit [17].

2.3 Agriculture Sector

One of the focuses of smart agriculture is increasing crop productivity and reducing irrigation waste. Large crop production is a critical issue as it is a basic human need. Therefore, more products are obtained by field-based sensors, drones, and advanced technologies. A future water scarcity forecast is also provided [18].

Crops must be constantly monitored for moisture, temperature, soil, light, etc. Figure 5 shows a smart agriculture IoT architecture for monitoring the factors supported by IoT technology. The farmer gets data for analysis from sensors such as Ultra Violet, Temperature, Air Humidity, Soil Moisture, Soil Ingredient, Color, etc. Drones also take part in data collection. Hence, abundant data is collected to extract information and obtain smart agriculture applications [19]. ML is the tool for making sense of the collected IoTdata.

Drone-supported IoT technology and ML jointly play an important role in the agricultural sector's productivity, sustainability, and decision-making. ML algorithms help farmers detect plant diseases, increase crop productivity, and optimize irrigation. Moreover, farmers can use more efficient and effective farming methods with the automation of agricultural machinery, drones, and the integration of forward-thinking technologies. AI uses weather forecasts, harvest timing, and market demands to help farmers make

decisions. The combined use of the technologies enables a significant transformation in agricultural production. A simple example is the smart irrigation system that predicts crop water needs. The results obtained with an ML algorithm are sent to farmers via e-mail to make in advance water supply decisions [18].

Figure 4. Smart Manufacturing 4.0 use-case diagram [15].

Figure 5. The architecture of a smart agriculture application [19].

Figure 6. The diagram of a smart grid [20].

2.4 Energy Sector

Smart grid is a term used to describe various new databased services in the supply, marketing, storage, and use of renewable energy. The smart grid that provides transmission and distribution of power in smart cities is shown in Figure 6. Smart grids are considered as the future of electrical energy management systems [20]. Smart grids emerged due to the digitalization of energy systems, making processing and interpreting digital data possible. Merged IoT and ML models play an important role in the fast and efficient processing of energy distribution data. ML models perform data storage, visualization, analysis, capture, and update tasks.

Smart grid systems play a key role in power line load forecasting. The ML techniques use to forecast power line loads helped design load balancing in power grids. Predictive maintenance (PM), supply on demand, and energy source discovery are all possible with the IoT-ML merged applications. PM isindispensable not only for the vitality and longevity of machines but also for reducing human error in energy distribution. Condition-based maintenance often saves costs and optimizes production. There are also multilevel neural ML networks that learn consumption patterns by analyzing large amounts of data from different energy sources, devices, and power system infrastructure [21,22]. Many companies were able to implement Industry 4.0 and Society 5.0 due to the IoE environment. Examples of IoE environment technologies include the following subjects:

- energy distribution and storage systems,
- advanced energy materials,
- response to energy demand and efficiency,
- strategic energy planning under uncertainty,
- large-scale integration of renewable energy,
- Big data analytics in a smart grid environment.

AI also plays an important role in complex issues such as estimating renewable energy sources used in power generation, managing energy storage systems, and predicting energy market prices. Thus, AI contributes to creating more efficient, reliable, and sustainable energy systems. In summary, the IoE environment of merged IoT and ML is widely used in smart energy production, power grid management, optimum power flow, voltage, and reactive power control [23].

2.5 Transportation Sector

The use of IoE environments in smart transportation has attracted the attention of many researchers. The IoT-ML merged projects have been used in areas such as traffic management, route optimization, parking, street light illumination, accident prevention/detection, road and infrastructure anomaly detection, telematics, and autonomous driving services[24]. Some examples of IoT-ML projects in the transportation sector are [25]:

• forecasting travel demand,

- smart card data analysis,
- smart route,
- sustainable urban mobility practices and development.

The Canadian Intelligent Transportation System (ITS) is a warning system that prevents traffic accidents and congestion by offering a smart route and controlling traffic lights. Therefore, ITS plays a vital role in improving road safety. A typical ITS system can be seen in Figure 7. Technically, an ITS forms a network to provide commuters with all the functions and information they need to get safely from their starting point to their destination [26].

IoE projects help the transportation sector by providing environmental-aware vehicles. ML algorithms help cars to perceive the environment and move safely. Learning drivers' preferences offer a more personalized driving experience through ML analysis. The creation of driverless vehicles has also facilitated our lives in some cities. In summary, AI has helped us to predict traffic density, optimize travel times, prevent traffic accidents, and create safer, more efficient, and environmentally friendly smart cities.

2.6 Construction Sector

The hype of the construction sector is the smart house. Figure 8 presents the layered architecture, modules, and their interrelationships of the smart house design. The smart house application design comprises presentation, security, control, communication, data, and device layers. Various remotely controlled and monitored IoT devices, such motion/presence detectors, temperature sensors, and air conditioner-TV-oven consoles, serve in a smart house. The smart house automation system that provides control and communication is obtained by analyzing the collected Big Data from the IoT devices [27].

As a result of the widespread smart house applications, AI is now capable of managing assets, implementing energyefficient automation, ensuring security, and monitoring (e.g., Building Information Management). Various automation systems can automate a house's security, energy management (e.g., Home Energy Management System), and well-being. Integrated AI and smart IoT home systems optimize energy consumption, improve security systems, and increase occupant comfort [28].

Figure 7. The schema of the Canadian ITS system [26].

Furthermore, ML and AI are jointly used in many areas, such as building design, construction process, and management (e.g., Home Management System). ML algorithms enable innovation in civil engineering and architecture in complex tasks such as building materials selection, structural analysis, and project planning. AIpowered design software offers architects and designers faster and more efficient design processes.

Meanwhile, AI-controlled robots automate construction processes and increase efficiency.

3. Related Transformative Projects

The related work merging IoT and ML has been studied. Sectors in Table 1 present examples of the joint IoT and ML projects that propel IoE technologies. As a fast-progressing area, new literature comes out every day.

Health	Manufacturing	Agriculture	Energy	Transportation	Construction	Others
$[29]$ AI- Based Drug Diagnosis	[37] Smart Manufacturing	[46] Device to Prevent False Spring in Apricot Agriculture	[51] Fault Detectio n in Wind Turbine	[55] Smart Parking	[61] Smarthome	$[66]$ Fire Detection Using UAV
$[30]$ Gait Disorder Detection	[38] Supply Chain Management	[47] Smart Garden Irrigation	$[52]$ ML for Modelin g of Energy Consum ption	[56] Parking Lot Finder	$[62]$ Smart Building Evacuation System	[67] Robotic Systems and AI App in Livestock
$[31]$ AI- Based Diagnosis of Skin Diseases	[39] PM Decision-Making in Smart Manufacturing	[48] Detecting animals damaging fields	[53] Short- Term Solar Power Forecast ing	[57] AI Issues Fines for Honking Ban	[63] Energy Management in Buildings	[68] Stray Animal Control System with AI
$[32]$ Diagnosis of Skeletal Malocclusi on with ML	[40] Trends in Predictive Manufacturing Systems in Big Data Environment	[49] Agricultural Crop Monitoring	$[54]$ Smart Grids Energy Manage ment	[58] DL-based Multi- Functional Recognition for Autonomous Vehicles	[64] Smart mirror with image processing support	[69] Attendance System with Face Recognition
[33] Image Captioning Supported System	[41] Digital twins of manufacturing equipment	[50] Giyilebilir duyargalar		[59] Smart Cities for Traffic Management	[65] Data Mining Techniques in Smart Home	[70] Detection of Harmful Fishing Nets in Surface Waters
$[34]$ Stuttering individual developmen t tracking	[42] Supply Chain Optimization			[60] Foreign Object Detection on the Track		[71] Live Bomb Detection
[35] Classificatio n of brain waves	[43] Smart Factories					[72] Smart Water Dispenser
[36] Patient Monitoring (Philips) Healthcare)	[44] Connected Supply Chain (Industry 4.0)					[73] Fire Smoke Detection
	[45] Digital Twins for Equipment Simulation					[74-79] Works on Real-Time Human Tracking, Image Generation, Resume Classification, HydroFlow, and Emotion Analyses.

Table 1. Example joint ML and IoT projects in various sectors.

Figure 8. The architecture of a smart house application [27].

4. Results and Discussion

Integrating IoT and ML towards forming a strong IoE environment is poised to play an increasingly important role in shaping the future of various sectors. As the technologies evolve, their collective impact is expected to unlock new possibilities and opportunities. As industries become increasingly dependent on real-time data for critical decision-making processes, the future roles of IoT and ML are going to be instrumental in driving innovation, process optimization, and smart systems design. The future role of the IoT and ML merger will likely extend beyond current applications to delve deeper into more complex decisionmaking and prediction systems. The advances in wireless technologies will further strengthen the capabilities of IoE applications by providing faster and more reliable connections across all expanding sectors. Thus, faster and more reliable wireless communications will contribute to advanced automation, personalized user experiences, and sustainable applications. As a solid IoE example, the Scientific and Technological Research Council of Turkey (TÜBİTAK, Türkiye Bilimsel ve Teknolojik Araştırma Kurumu) includes Big Data, IoT, Embedded Systems, Semiconductor, Broadband and display technologies among its top innovative research and development project topics. Meanwhile, the AI technology roadmap is in the works [80].

At this point, there are certain issues when open and transparent dialogues are required among sector leaders, academics, and regulators. First, a multidisciplinary approach is needed to evaluate the joint project's social and economic impact and the principles of sustainability and equity. In this context, in-depth discussions and cooperation are inevitable in determining the ethical framework of future technological developments and social benefits.

It is true that the bridge between IoT and ML that forms the IoE world is changing the business world and daily life radically. But, the ethical concerns of rapid technological advances should not be ignored.

Also, the encountered security and privacy challenges of the growing complex IoE networks, where billions of M2M communications take place daily, should be met by regularly updated equivalent rules, regulations and laws.

5. Conclusions

This study focuses on the innovative merger of IoT and ML resulting in the present IoE world. The interaction among the people, processes, data, and things of the IoE world ismaking a big bang. The effect of the IoT-ML merger on the world sector map has been studied through state-ofthe-art transformative technologies. Our study shows that the transformation of our world by rapidly evolving sectors is enormous. Our study provides abundant solid project examples and academic works. According to our future role analyses, the transformative technologies have huge opportunities in all sectors.

Our work encourages new initiatives and multidisciplinary work on the world sector map by identifying the benefits of the joint IoT and ML initiatives. We clearly present the potential of the evolving IoE environment to create a strong incentive for a technology-driven future, in all sectors.

Declaration

The author(s) declared no potential conflicts of interest concerning this article's research, authorship, and publication. The author(s) also stated that this article is original and was prepared by international publication and research ethics, and ethical committee permission or any special permission is not required.

Author Contributions

Author Uygur performed the literature search, and prepared the manuscript. Author Özcanhan supervised and improved the study, proofread the article.

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