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# **Catalytic Effect of Ni and Cu Embedded Graphene Surface on Sulfur Dioxide (SO2) Decomposition Reaction**

Aykan AKÇA\* 1

# **Abstract**

Sulfur dioxide  $(SO<sub>2</sub>)$  decomposition reaction on Ni and Cu embedded graphene surfaces were investigated using density functional theory. Grimme-D2 correction was used for Van der Waals interactions that could be induced by the interactions between adsorbed structures and surface. Metal-embedded graphene surfaces are expected to be much cheaper than bulk system because they only use a few metal atoms contrary to bulk. Firstly, the charge density on metal embedded systems and SO2 adsorbed on both surface was displayed with the electron density difference map and investigated with the Bader charge analysis. Then, the sequential dissociation of  $SO_2$  were systematically investigated. Finally,  $SO_x$  molecules and their decomposed geometries were obtained and CI-NEB calculation was performed to find activation barriers related to  $SO<sub>x</sub>+yO$ . It is concluded that Cu embedded graphene surface is more favorable than Ni embedded graphene surface in terms of activation energetics. Cu-based graphene materials can be used as catalyst an efficient and cheap in  $SO<sub>2</sub>$  decomposition.

**Keywords:** Nickel-embedded graphene, copper-embedded graphene, SO<sub>2</sub> decomposition, single-atom catalysis

# **1. INTRODUCTION**

A significant part of energy needs in transportation, power plants, factories and houses is supplied by fossil fuels [1,2]. Sulfur dioxide  $(SO<sub>2</sub>)$ , which is an important air pollutant formed during the combustion reactions of fossil fuels, causes negative health and environmental effects known as acid rain by releasing into the atmosphere. Moreover, sulfur dioxide is an important environmental pollutant containing sulfur in its molecular structure. Environmental regulations emphasize the development of highly efficient catalysts to reduce  $SO<sub>2</sub>$  emissions, control automobile emissions, remove sulfur from fossil fuels, and reduce the  $SO<sub>2</sub>$  ratio generated during combustion reactions [3]. Adsorption and decomposition of  $SO<sub>2</sub>$  have been extensively studied before experimentally and theoretically on various metal surfaces including Ni [4,5]Cu [5,6], Pt [7,8], Pd [9], Rh [10]. Yokoyama et al. observed the adsorption mechanism of  $SO<sub>2</sub>$  on Ni (111) and Ni (100) surfaces by using near-edgeadsorption fine structure (NEXAFS) and X-ray photoelectron spectroscopy (XPS) [11]. They found that  $SO<sub>2</sub>$  bound to both surfaces molecularly and S-O intramolecular bonds elongated. Wilde et al. investigated the  $SO<sub>2</sub>$ decomposition depending on the temperature on the Ni  $(110)$  surface and they observed that  $SO<sub>2</sub>$ lost its molecular structure under 160 K temperature and  $SO<sub>3</sub>$  and S structures occurred on both surfaces [12]. Moreover, they found the average bond length of all S containing species

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(S-Ni) as 2.30 Å Yokoyama et al. studied the adsorption mechanism of the SO<sup>2</sup> molecule on the Ni (110) surface and they reported that the S-O bond lengths elongated compared to the gas phase [13]. In another study by Jackson et al., the adsorption mechanism of  $SO<sub>2</sub>$  molecule on Cu (111) surface was investigated by using Chemical-shift normal-incidence X-ray standing waves (CS-NIXSW) technique [14].At that study, it was found that while  $SO<sub>2</sub>$  bound to the surface molecularly at low temperatures, it disintegrated as it approached to room temperature Terada et al. examined the adsorption mechanism of the  $SO<sub>2</sub>$ molecule on the Pd (111) surface using X-ray absorption fine structure measurements [15]. In the study, it was reported that the asymmetrical structure of  $SO<sub>2</sub>$  bonds played an important role in the elongation of S-O bonds Ungerer et al. investigated Platinum surface morphology and SO<sup>2</sup> decomposition reaction on different surfaces depending on the temperature parameter, and they stated that the percentage surface ratios expressed in Pt surface morphology changed depending on the temperature [16]. Moreover, they stated that while two surfaces  $((001)$  and  $(111)$ ) were dominant at 0 and 800 K, three surfaces ((001), (011) and (111)) were dominant at 298 and 400 K. Besides, they indicated that the Pt (111) surface was more active than other surfaces on the reaction Rodriguez et al. studied  $SO<sub>2</sub>$ decomposition on the surface of TiC (001) experimentally and theoretically. In their study, the adsorption heat of  $SO<sub>2</sub>$  was determined as 13-16 kcal/mol. Furthermore, they reported that while a significant part of the molecular structures was moving away from the surface below room temperature, some of them underwent complete decomposition into S-O atoms [17]. Rodriguez et al. investigated the catalytic activity of the Au/TiO<sub>2</sub> (110) surface, which was doped with metal to the metal oxide surface, on the  $SO<sub>2</sub>$ decomposition reaction [18]. Moreover, they reported that the  $Au/TiO<sub>2</sub>$  surface was more active compared to stoichiometric gold Graphene-based materials have highly superior mechanical and electrical capabilities. Because of these properties, they are used extensively in chemical sensors and catalyst researches [19-22].

To date, a few studies were available studying SO<sup>2</sup> decomposition on cost effective systems instead of expensive transition metals. In this study, the catalytic activities of single atom Ni and Cu elements on defective graphene substrate were theoretically investigated for  $SO<sub>2</sub>$ decomposition reaction.

# **2. COMPUTATIONAL METHODS**

All calculations are by Quantum Espresso package (version 6.3, [23]) based on performed based on density functional theory GGA-PBE was selected as an exchange-correlation functional [24]. PAW (the projector augmented wave) potential was used to analyze the electronic structure [25]. 80 Ry and 800 Ry were considered for kinetic energy cutoff for wavefunctions and kinetic energy cutoff for charge density and potential, respectively during all geometric calculations. Grimme-D2 term was embedded in the system to observe the weak and van der Waals interactions [26].  $1x10^{-5}$  Ry/atom and  $1x10^{-6}$ Ry/atom were used as criteria for force and energy converging during geometric optimization, respectively. Spin-polarization effect is considered for all calculations. Brillouin zone samplings of  $14x14x1$  for the electronic structure interactions and 4x4x1 Monkhorst-Pack k-point for geometric optimizations were used. Gas calculations were made in a 15  $\AA$ <sup>3</sup> length of threedimensional box. Partial atomic charges were calculated by using Bader charge analysis.

The adsorption energy of atomic and molecular structures (adsorbate) on the graphene surfaces are calculated by using the following,  $E_{ads}$ , equation:

$$
E_{ads} = E_{adsorbates/surface} - (E_{surface +} E_{adsorbate}) \tag{1}
$$

where Eadsorbates/surface is the total energy of atomic and molecular system on the graphene surface; Esurface is the total energy of the optimized graphene layer and Eadsorbate is the total energy of the optimized atomic and molecular structures in gas phase.

Transition geometry and the energy that is needed was calculated with CI-NEB (climbing imagenudged elastic band) [27,28]. In this study, the

reaction and activation energies were calculated using the equation (2) and equation (3), respectively.

$$
E_r = E_{IS} - E_{FS}
$$
 (2)

$$
E_a = E_{TS} - E_{IS} \tag{3}
$$

 $E_{IS}$  and  $E_{FS}$  represent the initial energy and final energy, respectively.  $E_{TS}$  is the transition energy between E<sub>IS</sub> and E<sub>FS</sub>. Avogadro [29] and Burai software [35] were used to visualize for all geometric configuration and the bond lengths.

# **3. RESULTS AND DISCUSSION**

# **3.1. Modeling of Ni and Cu Embedded Graphene Surfaces**

To ensure sufficient accuracy of our calculations, some important parameters obtained for Niembedded graphene (NiG) and Cu-embedded graphene (CuG) surfaces are given in Table 1. For modeling graphene systems, a  $4 \times 4$  singlegraphene layer consisting of 32 carbon atoms was constructed. The distance between the nearest C-C on the optimized graphene sheet was obtained by 1.42 Å. This value is consistent with other studies in the literature [30]. Then, the carbon at the center of graphene nanosheet were substituted with Ni or Cu. These systems are shown in Figure 1.



Figure 1 The electron density difference (EDD) maps  $(\pm 0.006$  au) of (a) Ni and (b) Cu embedded graphene systems

The binding energy of the Ni and Cu atoms to the defective graphene surface were -6.77 eV and - 3.92 eV. Moreover, the bond lengths between the closest carbon atoms and Cu and Ni were 1.87 Å and 1.88 Å, respectively.

All values obtained in Table1 are consistent with the study reported by Krasheninnikov et. [31]. According to Bader charge analysis, electron transfer occurs from metal atoms to neighboring carbon atoms. After this charge transfer, positive charge accumulation is  $+0.67|e|$  and  $+0.65|e|$  at Ni and Cu atoms, respectively.

<b>Parameters</b>	NiG	CuG
Elevation-h $(\AA)$	1.4	1.51
Bond Lengths (Å) (Ni- $C$ ), $Cu-C$ )	1.87	1.88
Magnetic Moment $(\mu_R)$		1
Binding Energy $(E_R)$ (eV)	$-6.77$	$-3.92$
Metal (Ni, Cu), $C_1$ , $C_2$ ,	$+0.67, 0.27, -$	$+0.65, 0.25, -$
$C_3$ charges, $ e $	$0.22 - 0.27$	$0.24 - 0.20$

Table 1 Physical parameters obtained for the Ni and Cu embedded graphene surfaces

The electron density difference (EDD) maps is plotted, where the electron density accumulation and depletion sites are displayed in yellow and blue, respectively. According to EDD, electron density accumulation was occurred around Ni-C and Cu-C bonds, which verifies strong binding between metal atom and neighboring carbon atoms. Moreover, the blue regions are showed the positive charge density around both Ni and Cu. Positive charge centers create a strong potential attraction region. Therefore,  $SO<sub>2</sub>$  and other intermediate state are expected to interact on these regions

# **3.2. The Adsorption of SO<sup>2</sup> on The Ni and Cu Embedded Graphene Surfaces**

Two different adsorption geometry were considered for SO<sup>2</sup> molecule on the Cu and Ni embedded graphene surfaces.  $SO<sub>2</sub>$  can adsorb to graphene surfaces by both sulfur and oxygen, namely through its S-end or O-end. These configurations were showed in Figure 2 and the results obtained were given in Table 2.

In the complex\_1, the adsorption energy of  $SO<sub>2</sub>$ was -0.9 eV on NiG surface. The bond lengths between S and O atoms were calculated to be 1.47 Å. Also, the Bader analysis reveals that about 0.34  $|e|$  are transferred from the surface to the  $SO<sub>2</sub>$ 

molecule. According to EDD maps, a great electron density accumulation is found around the Ni-O bond, which confirms the chemical binding between the O atom and Ni. Therefore, the total electron density around the Ni Atom has been seen to decrease. In other words, very low a charge transfer takes place from the surface to the S-O bonds. This implies that the surface has low activity over the S-O bond lengths.



Figure 2 The different adsorption geometries of  $SO_2$  molecules on both surface and EDD maps ( $\pm 0.003$  au)





In the Complex 2, the adsorption energy of the molecule  $SO_2$ , which binds to the surface by oxygen-end, is calculated as -1.17 eV. The bond lengths between S and O atoms were calculated to be 1.48 Å and 1.54 Å. Bader analysis results of these geometry showed that 0.43|e| was transferred from the surface to the  $SO<sub>2</sub>$  molecule. Moreover, the adsorption of the  $SO<sub>2</sub>$  molecule by O-end tends to cause a significant electron density on molecule, which shows elongation in the S-O bonds of the molecule.

In the Complex 3, the adsorption energy of  $SO<sub>2</sub>$ molecule was determined to be of -1.02 eV. Bader analysis results 0.35 |e| charge transfer was realized from the surface to the SO<sub>2</sub> molecule. Similar to the Bader analysis results of the "Complex 1", a great amount of electron density on Cu atom was observed to be consumed and this density was accumulated on between Cu and S. The surface had a low effect on the S-O bond length.

In the complex 4, the adsorption energy of  $SO<sub>2</sub>$ , which is adsorbed to the surface by the O-end, is -1.36 eV. From Bader charge analysis results, a transfer of  $0.47$ |e| from surface to  $SO<sub>2</sub>$  molecule was found. Most of the transferred electron was observed to accumulated on the S-O bonds. This result could be the reason for the elongation in S-O bonds that was given in Table 2.

It is found that, the adsorption geometries of  $SO<sub>2</sub>$ , which is adsorbed to the surface by the O-end have significant catalytic activity in the elongation of S-O bonds.

# **3.3. Decomposition Mechanism of SO2→SO+O on Ni and Cu Embedded Graphene**

The reaction pathway for the sequential decomposition of  $SO<sub>2</sub>$  to S and O on Ni and Cu embedded graphene have been systematically studied using density functional theory. The adsorption geometries of  $SO<sub>2</sub>$ , which binds with

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oxygen-end to the surfaces were considered as initial state. SO+O geometry obtained by breaking down one of the S-O bonds was considered the final geometry of first step. The transition states have been investigated using the CI-NEB method. CI-NEB method allows to calculate the energy barrier needed between the initial and final state.

For the initial state, the distance between S and O bonds was 1.48 Å, 1.54 Å and 1.49 Å, 1.55 Å on NiG and CuG, respectively. In the transition state, SO<sup>2</sup> molecule loses one of its oxygen bound to the sulfur atom. The oxygen atom dissociated in the TS is at distances of 2.34 Å and 1.99 Å from Sulfur atom on NiG and CuG, respectively. According to our calculations, the energy barriers required for breaking the S-O bond was 0.86 eV and 1.58 eV on NiG and CuG, respectively. The positive reaction energy values, which was found as  $+0.53$  eV and  $+0.73$  eV on NiG and CuG, respectively, indicates that the reaction occurred as endothermic.



Figure 3 Side view of the initial, final, and transition states for  $SO_2 \rightarrow SO+O$  on CuG ve NiG surfaces

# **3.4. Adsorption and Decomposition of SO molecules Ni and Cu embedded graphene surfaces**

The adsorption geometry optimized of the SO molecule on both graphene surfaces was shown in Figure 4. Our calculations show that the adsorption energies of SO molecules on the Ni and Cu graphene surface were determined to be of -1.80 eV and -1.84 eV, respectively.



Figure 4 Side view of optimized adsorption geometries of SO molecule on CuG and NiG surfaces

The most stable adsorption geometries of the SO molecule on surfaces were considered to be the initial state of the second reaction step. The S+O configuration, which was obtained through the transition state, was considered as the final stage of the reaction. The initial, final and transition state geometries for the decomposition of SO  $\rightarrow$ S+O are shown in Figure 5.



Figure 5 Side view of the initial, final and transition states for  $SO \rightarrow S+O$  on CuG ve NiG surfaces.

For the initial state, the bond distances between S and O on the Ni and Cu surface was 1.51 Å and 1.52 Å. In transition state (TS), the oxygen passing over the activation energies were at distances of 2.69 Å and 2.81 Å from the sulfur atom on NiG and CuG surfaces, respectively. Moreover, the values of activation energies for NiG and CuG surfaces were calculated to be 0.70 eV and 0.50 eV, respectively. The reaction energies of  $SO \rightarrow S+O$  were 0.44 eV and 0.09 eV on NiG and CuG, respectively. This reaction step occurs endothermically on both graphene surface. It also appears that the transition state geometry on the NiG surface exhibit behavior close to the IS geometry, while it was close to the FS geometry on CuG surface. This result may explain the low activation energy calculated on the CuG surface.

### **4. CONCLUSION**

In this article, SO<sup>2</sup> decomposition studied on Ni and Cu embedded graphene surfaces through density functional theory. Bader charge analysis of the graphene surface was calculated and the map of electron density difference (EDD) was obtained. The charge analysis results show that a positive charge density region on the Ni and Cu atoms played an important role in detecting catalytic activity. The adsorption energies of  $SO_x$  $(x= 1,2)$  were calculated. It was observed that  $SO_x$ structures bound to the surface through chemical interaction. The activation energies needed for the  $SO_2 \rightarrow SO_2+O$  reaction on NiG and CuG surfaces were found as 1.58 eV and 0.86 eV, respectively. The required activation energies  $SO \rightarrow S+O$ reaction were calculated as 0.70 eV for NiG and 0.50 eV for CuG. The results show that CuG surface are the more active than NiG surface. The results of the CuG surface have comparable activation barriers in all steps for other transition metals [32-34].

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No conflict of interest or common interest has been declared by the author.

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The author of the paper declares that he complies with the scientific, ethical and quotation rules of SAUJS in all processes of the article and that he does not make any falsification on the data collected. In addition, he declares that Sakarya University Journal of Science and its editorial board have no responsibility for any ethical violations that may be encountered, and that this study has not been evaluated in any academic publication environment other than Sakarya University Journal of Science.

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