

# Comparison of Electronic and Magnetic Properties of 4d Transition Metals Based NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> Spinels

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Keywords	Abstract
Spinel	Half-metallic properties of NbAl <sub>2</sub> F <sub>4</sub> spinel and semiconductor characteristics of TcAl <sub>2</sub> F <sub>4</sub> spinel were
NbAl <sub>2</sub> F <sub>4</sub>	investigated with the help of the WIEN2k program. NbAl <sub>2</sub> F <sub>4</sub> spinel shows a metallic character in the up- electron states, while it has a semiconductor nature in the down-electron states. In NbAl <sub>2</sub> F <sub>4</sub> spinel, the
GGA+mBJ	$E_g$ bandgaps were calculated in GGA and GGA+mBJ 1.551 eV and 1.622 eV, respectively. The $E_{HM}$
Half-metallic TcAl <sub>2</sub> F <sub>4</sub>	half-metallic bandgaps were obtained 0.410 eV and 0.422 eV, respectively. In the up-spin states of TcAl <sub>2</sub> F <sub>4</sub> spinel, $E_g$ values were obtained 1.199 eV and 1.447 eV for the GGA and GGA+mBJ methods, respectively, while they were obtained 1.281 eV and 1.519 eV in the down-spin states, respectively. When GGA+mBJ is used, it is easily observed that the semiconductor characters increase. Total
	magnetic moments of NbAl <sub>2</sub> F <sub>4</sub> and 1CAl <sub>2</sub> F <sub>4</sub> spinels were calculated 6.00 $\mu_B$ /cell and 10.0 $\mu_B$ /cell, respectively. When both electronic and magnetic moment values are carefully examined, NbAl <sub>2</sub> F <sub>4</sub> and TcAl <sub>2</sub> F <sub>4</sub> spinels can be used as alternative compounds in spintronic applications.

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## **1. INTRODUCTION**

In recent years, the characteristics of magnetic materials, especially with the effect of their spin orientation, have begun to be used in many areas of technological applications. This branch of science, which forms the basis of spin orientations, is called spintronics. In short, spintronics is the science that examines the technological properties of nano-sized devices depending on their spin orientation. Hirohata et al. (2020) have recently carried out a very detailed study involving spintronic applications. They showed similar and different properties of magnetic and semiconductor devices. Since spin generation and spin-polarized electrons form the basis of spintronic applications, they mentioned the methods by which spins can be polarized even in nonmagnetic materials. These methods are spin injection from a magnetic field, an electric field or a ferromagnetic material. Magnetic random-access memories (MRAM) are one of the application products of spintronic technology. Shin et al. (2006) investigated the effects of Cl<sub>2</sub> and O<sub>2</sub> gases with varying density on the Co<sub>2</sub>MnSi magnetic film, which was thought to be used for MRAM devices. They experimentally proved that the etch rate of the  $Co_2MnSi$  magnetic film decreased with increasing  $Cl_2$  and  $O_2$  gas densities. Quiroz et al. (2020) investigated the properties of Co/TiO<sub>2</sub> bilayer and TiO<sub>2</sub>:Co thin films. It was observed that these materials, obtained experimentally at 293 K and 473 K temperatures, had low energy consumption and expanding storage capacities in the applied magnetic field. Just like MRAM, magnetic sensors (MS) are one of the products obtained as a result of spintronic applications. Zhang et al. (2020) investigated the magnetic sensor properties of boron-doped FeGa film. As a result of their experimental measurements, the magnetic sensitivity of the FeGaB magnetic sensor was obtained as 152.1 Hz/mT. This value was found to be almost four times the FeGa film obtained without boron doping. The sensitivity characteristics of the BaTiO<sub>3</sub>-Fe-Ga&PDMS magnetic sensor were reported by Xu et al. (2022). The best value of the sensitivity of the magnetic sensor was experimentally obtained as 3.4 V/T when the BaTiO<sub>3</sub> concentration was 20%. In addition, experimental investigations of silicon-based Bi<sub>2</sub>FeNiO<sub>6</sub> double perovskite-based and MgAl<sub>2</sub>O<sub>4</sub> barrier magnetic tunnel junctions (MTJ) were performed by Ravi (2020) and Khanal et al. (2022), respectively. When Bi<sub>2</sub>FeNiO<sub>6</sub> double perovskite was used, tunnelling magnetoresistance varying from 0.2% to 1.8% was obtained at different temperatures. While the magnetic anisotropy energy density of the MgAl<sub>2</sub>O<sub>4</sub> barrier magnetic tunnel joint annealed at 400  $^{\circ}$ C degree was obtained as 2.25 mJ/m<sup>2</sup>, 60% success was achieved in the magnetoresistance. As can be seen from the examples, there are many technological devices that spintronic applications are transformed into.

When de Groot et al. (1983) discovered the half-metallic (HM) features of NiMnSb ferromagnetic material, half-metallic materials have also been widely used in spintronic applications. Half-metallic materials attract attention with their 100% spin polarization and different electronic properties under magnetic fields. One of the spin orientations of HM materials shows metallic character around the Fermi energy level, while the other shows semiconductor character. The spin orientation, which shows semiconductor character, adds 100% spin polarization to the materials. Therefore, these materials are true half-metallic. In cases where the Fermi energy level is cut by electrons and still has a band gap, the spin polarization ratio decreases. In such cases, materials are called nearly half-metallic. Birsan and Kuncser (2022) and Patel et al. (2022) investigated the half-metallic character of  $Zr_2CrAl$  and  $Mn_2SiRh$  Heusler compounds. The HM band gap ( $E_{HM}$ ) and total magnetic moment of Zr<sub>2</sub>CrAl ferrimagnetic compound were calculated as 0.35 eV and 1.00  $\mu_B/f.u.$ , respectively. In Mn<sub>2</sub>SiRh compound, the E<sub>HM</sub> value was obtained as 0.36 eV. The half-metallic characters of the Cd<sub>1-x</sub>Zn<sub>x</sub>Cr<sub>2</sub>Se<sub>4</sub>, GeGa<sub>2</sub>O<sub>4</sub> and MnTi<sub>2</sub>O<sub>4</sub> spinels were studied by Bouhbou et al. (2019), Rafiq et al. (2022) and Nadeem et al. (2022), respectively. Differences in half-metallic band gaps were observed according to varying x-amounts of  $Cd_{1-x}Zn_xCr_2Se_4$  (x = 0, 0.125, 0.375 and 0.5) compounds. In x=0, the theoretical band gap values were obtained as 0.43 eV and 1.11 eV by GGA and GGA+mBJ methods, respectively. Here, an experimental band gap of 1.3 eV was obtained and the closest theoretical calculation to this value was with GGA+mBJ. It was observed that the theoretical band gaps obtained with increasing x values decreased. GeGa<sub>2</sub>O<sub>4</sub> oxide spinel showed semiconductor character with band gaps of 1.605 eV in the spin-up state and 1.578 eV in the spin-down state. The MnTi<sub>2</sub>O<sub>4-x</sub> compound was obtained with a direct band gap of about 5 eV in the spin-up states. Here x represents the vanadium concentration. The band gap rises above 5 eV when the compound is changed to  $MnTi_{2-x}O_4$ . As can be seen from the examples given, spintronic application areas involved different types of materials. One of the most widely used among these types is half-metallic characters. The main purpose of this study is to examine in detail the electronic, magnetic and HM characters of 4d-transition metal-based NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels. The magnetic moments of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were calculated as 6.00  $\mu_B$ /cell and 10.0  $\mu_{\rm B}$ /cell, respectively. NbAl<sub>2</sub>F<sub>4</sub> spinel has a half-metallic ferromagnetic structure while TcAl<sub>2</sub>F<sub>4</sub> spinel has a semiconductor nature. Therefore, two different compounds NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels will be alternative compounds for spintronics applications.

## 2. MATERIAL AND METHOD

Electronic and magnetic properties of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were performed using WIEN2k developed by Blaha et al. (2020, 2022). Both GGA-PBE (Perdew et al., 1996) and GGA+mBJ (Tran & Blaha, 2009) methods were used with the help of WIEN2k, which is based on Density Functional Theory. Some research on Density Functional Theory and its methods was carried out by Singh (1994). While forming NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels, 227-space number and Fd-3m symmetry group were chosen.

Here, the transition metals (Nb, Tc), Al and F atoms are placed at the space coordinates of 0.125/0.125/0.125, 0.50/0.50/0.50 and 0.25/0.25/0.25, respectively. The cubic structures obtained with the help of all these space numbers, symmetry groups and atomic coordinates were given in Figure 1. For the initial state calculations, the spin cutoff values ( $R_{MT}K_{max}$ ) were set to 7 giving the minimum energy values. While performing the potential calculations of the atoms using the GGA method, all atoms of the formed structure are examined separately. Therefore, potential calculations of Nb, Tc, Al, and F atoms examined in this study are performed in Muffin-Tin spheres, an imaginary sphere surrounding these atoms. The radii of these Muffin-Tin spheres must be larger than the radii of the atoms they surround. If they are small, potential leaks from atoms can occur. Therefore, inaccurate results may occur in the calculations performed. In this study, the radii of Muffin-Tin

spheres surrounding Nb, Tc, Al, and F atoms were selected as 2.00 a.u., 1.88 a.u., 2.22 a.u., and 2.02 a.u., respectively. Electron convergence values were chosen as 0.0001 e within 1728 special k-points in the first Brillouin zone.



*Figure 1.* The molecular schematic representations of  $NbAl_2F_4$  and  $TcAl_2F_4$  spinels

#### **3. RESULTS AND DISCUSSION**

The energies obtained around the values corresponding to -10%, 8%, 6%, 4%, 2%, 0%, 2%, 4%, 6%, 8% and 10% changes of the lattice parameters, and volume-energy curves of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were obtained. According to these curves, ferromagnetic phases were obtained as more stable for both spinels. Since Murnaghan's equation of state (1944) fit these energy values, the initial state values were calculated, and optimization curves were given in Figure 2.



*Figure 2.* The calculated energy-volume curves of  $NbAl_2F_4$  and  $TcAl_2F_4$  spinels

Equilibrium lattice parameters, bulk modulus and the first derivative of bulk modulus, equilibrium volume and energy values of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were given in Table 1. The lattice parameter of NbAl<sub>2</sub>F<sub>4</sub> spinel was calculated as 10.11 Å, while it was calculated as 10.25 Å for TcAl<sub>2</sub>F<sub>4</sub> spinel. The lattice parameter obtained at the equilibrium point of the TcAl<sub>2</sub>F<sub>4</sub> spinel formed by using the Tc element, whose atomic radius is smaller than the Nb element, is larger. Therefore, the equilibrium lattice parameters of the compounds may need to be evaluated independently of the atomic radii of the elements. Similar studies were obtained by Mahmood et al. (2022). The equilibrium lattice parameters of CaTi<sub>2</sub>S<sub>4</sub>, CaV<sub>2</sub>S<sub>4</sub>, CaCr<sub>2</sub>S<sub>4</sub>, and CaFe<sub>2</sub>S<sub>4</sub> spinels obtained as 10.73 Å, 10.75 Å, 10.91 Å, and 10.88 Å when Ti, V, Cr, and Fe elements were used, respectively. Although the atomic radii of the transition metals are used to decrease, the equilibrium lattice parameters and thus the volumes of the compounds obtained tend to increase. However, it may not be right to talk about the fact that this increase or decrease can occur at certain rates. As can be seen, there was a decrease in the lattice parameter when the transition metal Fe was used. The bulk modulus of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were obtained as 51.25 GPa and 48.44 GPa, respectively. Therefore, the resistance of NbAl<sub>2</sub>F<sub>4</sub> spinel to volume change was higher than that of TcAl<sub>2</sub>F<sub>4</sub> spinel. The B'value, which shows the value of the bulk modulus against pressure change, was obtained as 4.90 in NbAl<sub>2</sub>F<sub>4</sub> spinel, while it was obtained as 5.26 in TcAl<sub>2</sub>F<sub>4</sub> spinel. This is the expected result. Since the value of the bulk modulus is lower in  $TcAl_2F_4$  compound, it indicates that there is a possibility of undergoing more volume changes with the applied pressure. Comparison of decreasing bulk modulus with increasing lattice parameter can be done with  $CaTi_2S_4$ ,  $CaV_2S_4$ ,  $CaCr_2S_4$ , and  $CaFe_2S_4$  spinels. When Ti, V, Cr, and Fe transition metals were used, the ballast modulus of the structure obtained was 65.38 GPa, 62.10 GPa, 50.00 GPa, and 54.71 GPa, respectively. As can be seen, when Ti, V, and Cr transition metals are used, decreases in bulk modulus were observed inversely with increasing lattice parameters.

Compounds	a (Å)	B (GPa)	B	V <sub>0</sub> (a.u.) <sup>3</sup>	<b>E</b> <sub>0</sub> ( <b>Ry</b> )
NbAl <sub>2</sub> F <sub>4</sub>	10.11	51.25	4.90	1740.807	-18822.638
TcAl <sub>2</sub> F <sub>4</sub>	10.25	48.44	5.26	1815.716	-20687.184

*Table 1.* The initial state values of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels

After the obtained initial state values, the electronic properties of  $NbAl_2F_4$  and  $TcAl_2F_4$  spinels were investigated. First, the total densities of states (TDOS) were plotted for the -10 eV and +8 eV energy regions and were given in Figure 3.



Figure 3. TDOS of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels

Positive values of the total densities of states represented up-spins (majority), while negative values represent down-spins (minority). In the far negative energy densities of NbAl<sub>2</sub>F<sub>4</sub> spinel, sharp peaks were observed at the GGA method. The 100% symmetry between the up and down spins indicated that electrons were bounded in these energy regions. These conditions were the same both in the value obtained by the GGA+mBJ of NbAl<sub>2</sub>F<sub>4</sub> spinel and in the calculation results of TcAl<sub>2</sub>F<sub>4</sub> spinel. The vertical dotted lines at total densities of states indicated Fermi energy levels. As the Fermi energy levels were approached, there were separations in the electron densities. At each method, the up-spin electrons of NbAl<sub>2</sub>F<sub>4</sub> spinel contacted Fermi energy levels. This showed that the valence electrons can pass to the conduction band, that is, the metallic character revealed.

In down-spins, band gaps existed. In down-spin states, NbAl<sub>2</sub>F<sub>4</sub> spinel had a semiconductor character. Therefore, NbAl<sub>2</sub>F<sub>4</sub> spinel was a half-metallic ferromagnetic material. It is possible to see band gaps in both the up and down spins of TcAl<sub>2</sub>F<sub>4</sub> spinel when using the GGA and GGA+mBJ. In this case, TcAl<sub>2</sub>F<sub>4</sub> spinel had a semiconducting nature. The differences in the symmetry between the up and down spins of NbAl<sub>2</sub>F<sub>4</sub> spinel around the Fermi energy level allow for the formation of a net magnetic character. Likewise, the presence of these asymmetric electron distributions in energy ranges near the Fermi energy level of TcAl<sub>2</sub>F<sub>4</sub> spinel made the magnetic characters appear. It is possible to talk about studies similar to the electronic structure of spinels in this study. Nazar et al. (2022) investigated the magnetic and electronic calculations of Gd<sub>2</sub>MgS<sub>4</sub> and Tm<sub>2</sub>MgS<sub>4</sub> spinel were obtained as 1.44 eV and 1.65 eV, respectively. Thus, Gd<sub>2</sub>MgS<sub>4</sub> spinel showed a semiconductor character. Therefore, Tm<sub>2</sub>MgS<sub>4</sub> spinel was obtained as half-metallic ferromagnetic by Nazar et al. (2022). The electronic properties of the semiconductor and half-metallic spinels Gd<sub>2</sub>MgS<sub>4</sub> and Tm<sub>2</sub>MgS<sub>4</sub> obtained by Nazar et al. (2022) show similar characteristics to the electronic properties of TcAl<sub>2</sub>F<sub>4</sub> and NbAl<sub>2</sub>F<sub>4</sub> appendent of the semiconductor and half-metallic spinels for the semiconductor for the semiconductor spinels for the semiconductor an

The atomic contributions of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were shown in Figure 4 and Figure 5 to investigate the partial electron densities to the total electron distributions. The electron contributions of the 4d transition metals Nb and Tc atoms were shown in the energy range of -6 eV to +6 eV. Their changes clearly and unequivocally come from these transition metals. Nb atoms have band gaps in their down-spins while their upspin cuts the Fermi energy levels. This supports TDOS. Most electron contributions come from d-orbitals in transition metals. In the -9 eV and -6 eV energy regions, contributions belong to the p-orbitals of fluorine atoms, while the main contributions come from the s-orbitals of aluminum atoms. This is the expected result because Nb atom has 5 electrons in 4d orbital, Tc atom has 4 electrons in 4d orbital and 1 electron in 5s orbital. The main electron contributions of 4d orbitals of Nb and Tc atoms have also been shown by Saberi et al. (2014) in transition metals doped to GaN nanotubes, Zhang et al. (2022) in Tc, Ru, Rh, and Cd doped to the blue P monolayer, Ali et al. (2021) in K<sub>2</sub>NbCl<sub>6</sub> and Rb<sub>2</sub>NbCl<sub>6</sub> variant perovskite compounds. In short, the electronic properties obtained in this study gave similar results with many studies in the literature.

It can be said that the electron distributions of aluminum and fluorine atoms of both  $NbAl_2F_4$  and  $TcAl_2F_4$  spinels have almost 100% symmetry. That is, the electrons of the aluminum and fluorine atoms are almost completely bounded. Both total and partial electron (PDOS) states show that  $NbAl_2F_4$  spinel shows a half-metallic character, while  $TcAl_2F_4$  spinel shows a semiconductor character.



Figure 4. The obtained PDOS of (a) NbAl<sub>2</sub>F<sub>4</sub> and (b) TcAl<sub>2</sub>F<sub>4</sub> spinels in GGA method



Figure 5. The obtained PDOS of (c)NbAl<sub>2</sub>F<sub>4</sub> and (d) TcAl<sub>2</sub>F<sub>4</sub> spinels in GGA+mBJ method

In order to support all these densities of states, the band structures were obtained and given in Figure 6 for the up and down spins of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels. It can be clearly seen in Figure 6 that the up-spins of NbAl<sub>2</sub>F<sub>4</sub> spinel show metallic character by cutting the Fermi energy levels in each method. In the GGA, electron densities are available in the -0.5 eV energy region. However, in the GGA+mBJ, all the electrons are destroyed in the range of 0 to -1 eV. Electron densities obtained in the GGA shifted to the range of -2 eV in the GGA+mBJ. In other words, the GGA+mBJ directly affected the energy gaps of the valence electrons.



Figure 6. The calculated majority (up-spin) and minority (down-spin) band structures of  $NbAl_2F_4$  and  $TcAl_2F_4$  spinels

In down-spin states, there are direct band gaps in each method. Valence band maximum (VBM) and conduction band minimum (CBM) values were at  $\Gamma$ -points. VBM, CBM, band gap (E<sub>g</sub>) and HM band gap (E<sub>HM</sub>) values of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were given in Table 2 and Table 3 for each method.

Compounds	VBM (eV)	CBM (eV)	E <sub>g</sub> (eV)	E <sub>HM</sub> (eV)
NbAl <sub>2</sub> F <sub>4</sub> <sup>GGA</sup>	-1.141	0.410	1.551	0.410
$NbAl_2F_4{}^{GGA+mBJ}$	-1.200	0.422	1.622	0.422

*Table 2.* The calculated VBM, CBM,  $E_g$  and  $E_{HM}$  of down-spin of NbAl<sub>2</sub>F<sub>4</sub> spinel

Table 3. The calculated VBM, 0	<i>CBM</i> , and $E_g$ of both up and	down-spins of TcAl <sub>2</sub> F <sub>4</sub> spinel
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Compounds	VBM <sup>Majority</sup> (eV)	CBM <sup>Majority</sup> (eV)	VBM <sup>Minority</sup> (eV)	CBM <sup>Minority</sup> (eV)	Eg <sup>Majority</sup> (eV)	Eg <sup>Minority</sup> (eV)
$TcAl_2F_4{}^{GGA}$	0	1.199	-0.067	1.214	1.199	1.281
$TcAl_2F_4{}^{GGA+mBJ}$	-0.047	1.400	0	1.519	1.447	1.519

In GGA calculations, the VBM and CBM values were calculated as -1.414 eV and 0.410 eV, respectively. In the GGA+mBJ, the VBM and CBM values were -1.200 eV and 0.422 eV, respectively. According to these results, the band gap values were obtained 1.551 eV and 1.622 eV in GGA and GGA+mBJ, respectively. The band gap values in the down spins of NbAl<sub>2</sub>F<sub>4</sub> spinel, a HM ferromagnetic material, can be calculated as different values in different applied methods. The band gaps of both the up and down spins of  $TcAl_2F_4$  spinel with semiconductor character can be clearly seen. There are direct band gaps in each method. In Figure 6, the band gaps increase when the GGA+mBJ is used. In addition, the VBM values in the up-spins are 0 and -0.047 eV in the GGA and GGA+mBJ, respectively, while the CBM values are 1.199 eV and 1.400 eV, respectively. Therefore, band gaps in the up-spin of TcAl<sub>2</sub>F<sub>4</sub> spinel were obtained as 1.199 eV and 1.447 eV in GGA and GGA+mBJ, respectively. When using the GGA method in down-spin conditions, the VBM value was calculated as -0.067 eV, while the CBM value was obtained as 1.214 eV. The band gap in the down-spin of the NbAl<sub>2</sub>F<sub>4</sub> spinel using the GGA method was 1.281 eV. When the GGA+mBJ was used, it was observed that the valence band electrons approached the Fermi energy level. The CBM value was 1.519 eV while the VBM value was 0. In other words, the Eg was obtained as 1.519 eV in the GGA+mBJ. The Eg values obtained in GGA+mBJ methods increased in both spinels. In short, while NbAl<sub>2</sub>F<sub>4</sub> spinel is a HM ferromagnetic material with its semiconductor character in its down-spins, TcAl<sub>2</sub>F<sub>4</sub> spinel was obtained as a semiconductor ferromagnetic material with its semiconductor character in both up and down spins.

Total magnetic moment values of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were obtained as 6.00  $\mu_B$ /cell and 10.0  $\mu_B$ /cell, respectively. Most contributions to the total magnetic moments came from the Nb and Tc transition metals. The partial magnetic moments of the NbAl<sub>2</sub>F<sub>4</sub> spinel are 1.561  $\mu_B$ /f.u., 0.003  $\mu_B$ /f.u. and 0.031  $\mu_B$ /f.u. for Nb, Al and F, respectively. The partial magnetic moments of TcAl<sub>2</sub>F<sub>4</sub> spinel are 3.149  $\mu_B$ /f.u., 0.003  $\mu_B$ /f.u. and 0.031  $\mu_B$ /f.u. and 0.092  $\mu_B$ /f.u. for Tc, Al and F, respectively. It was mentioned that the unbounded electrons at the total state densities are the 4d transition metals Nb and Tc. Therefore, these unbounded electrons did not create magnetic moments. It is clearly seen how much these electron contributions in Fermi energy levels affect the magnetic moments. As a result, NbAl<sub>2</sub>F<sub>4</sub> spinel is a HM ferromagnetic material with 6.00  $\mu_B$ /cell magnetic moment.

## 4. CONCLUSION

In this study, the electronic and magnetic properties of NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels were investigated. Ferromagnetic phases of spinels were found to be more stable at ground state values. NbAl<sub>2</sub>F<sub>4</sub> spinel was obtained as a true half-metallic ferromagnetic in each method. In the down-spins of NbAl<sub>2</sub>F<sub>4</sub> spinel, the  $E_g$ =1.551 eV and  $E_g$ =1.622 eV values were found for the GGA and GGA+mBJ, respectively. However, since the value of electron densities closest to the Fermi energy level was CBM, the half-metallic band gaps were obtained as  $E_{HM}$ =0.410 eV and  $E_{HM}$ =0.422 eV in each method. TcAl<sub>2</sub>F<sub>4</sub> spinel showed semiconductor character

in both methods. The band gaps in the up-spins were obtained as  $E_g=1.199 \text{ eV}$  and  $E_g=1.447 \text{ eV}$ , respectively, while the band gaps in the down-spins were obtained as  $E_g=1.281 \text{ eV}$  and  $E_g=1.519 \text{ eV}$ , respectively. NbAl<sub>2</sub>F<sub>4</sub> and TcAl<sub>2</sub>F<sub>4</sub> spinels have high magnetic moment values. The magnetic moment of NbAl<sub>2</sub>F<sub>4</sub> spinel was obtained as  $6.00 \mu_B$ /cell, while it is  $10.0 \mu_B$ /cell in TcAl<sub>2</sub>F<sub>4</sub> spinel. In both spinels, the most contributions belonged to the 4d transition metals Nb and Tc. As a result, NbAl<sub>2</sub>F<sub>4</sub> spinel is a true half-metallic ferromagnetic and TcAl<sub>2</sub>F<sub>4</sub> spinel is obtained as a semiconductor ferromagnetic material, and they are good alternative materials in spintronic applications for experimental works.

## **CONFLICT OF INTEREST**

The author declares no conflict of interest.

## REFERENCES

Ali, M. A., Ullah, R., Abdullah, S., Khan, M. A., Murtaza, G., Laref, A., & Kattan, N. A. (2021). An investigation of half-metallic variant perovskites  $A_2NbCl_6$  (A= K, Rb) for spintronic based applications. *Journal of Solid State Chemistry*, 293, 121823. doi:10.1016/j.jssc.2020.121823

Birsan, A., & Kuncser, V. (2022). Half-metallic properties of Zr<sub>2</sub>CrAl ferrimagnetic full-Heusler compound, investigated in tetragonal, orthorhombic and rhombohedral crystal structures. *Journal of Alloys and Compounds*, 900, 163491. doi:10.1016/j.jallcom.2021.163491

Blaha, P., Schwarz, K., Tran, F., Laskowski, R., Madsen, G. H. K., & Marks, L. D. (2020). WIEN2k: an APW+lo program for calculating the properties of solids. *The Journal of Chemical Physics*, *152*, 074101. doi:10.1063/1.5143061

Blaha, P., Schwarz, K., Madsen, G. K. H., Kvasnicka, D., Luitz, J., Laskowski, R., Tran, F., & Marks, L. D. (2022) WIEN2k An Augmented Plane Wave Local Orbitals Program for Calculating Crystal Properties. Vienna University of Technology Institute of Materials Chemistry, Vienna, Austria. ISBN 3-9501031-1-2 PDF

Bouhbou, M., Moubah, R., Bakkari, K., Zaari, H., Sabrallah, A., Khelfaoui, F., Mliki, N., Abid, M., Belayachi, A., & Lassri, H. (2019). Magnetic, half-metallicity and electronic studies of Cd<sub>1-x</sub>Zn<sub>x</sub>Cr<sub>2</sub>Se<sub>4</sub> chromium selenospinels. *Journal of Magnetism and Magnetic Materials*, 476, 86-91. doi:<u>10.1016/j.jmmm.2018.12.063</u>

de Groot, R. A., Mueller, F. M., van Engen, P. G., & Buschow, K. H. J. (1983). New class of materials: halfmetallic ferromagnets. *Physical Review Letters*, *50*(25), 2024-2027. doi:<u>10.1103/PhysRevLett.50.2024</u>

Hirohata, A., Yamada, K., Nakatani, Y., Prejbeanu, I.-L., Diény, B., Pirro, P., & Hillebrands, B. (2020). Review on spintronics: Principles and device applications. *Journal of Magnetism and Magnetic Materials*, 509, 166711. doi:10.1016/j.jmmm.2020.166711

Khanal, P., Zhou, B., Andrade, M., Mastrangelo, C., Habiboglu, A., Enriquez, A., Fox, D., Warrilow, K., & Wang, W.-G. (2022). Enhanced magnetoresistance in perpendicular magnetic tunneling junctions with MgAl<sub>2</sub>O<sub>4</sub> barrier. *Journal of Magnetism and Magnetic Materials*, *563*, 169914. doi:10.1016/j.jmmm.2022.169914

Mahmood, Q., Nazir, G., Alzahrani, J., Kattan, N. A., Al-Qaisi, S., Albalawi, H., Mera, A., Mersal, G. A. M., Ibrahim, M. M., & Amin, M. A. (2022). Room temperature ferromagnetism and thermoelectric behavior of calcium based spinel chalcogenides  $CaZ_2S_4$  (Z= Ti, V, Cr, Fe) for spintronic applications. *Journal of Physics and Chemistry of Solids*, *167*, 110742. doi:10.1016/j.jpcs.2022.110742

Murnaghan, F. D. (1944). The Compressibility of Media under Extreme Pressure. *Proceedings of the National Academy of Sciences*, 30(9) 244-247. doi:10.1073/pnas.30.9.244

Nadeem, A., Bashir, A. I, Azam, S., Rahman, A. U., & Iqbal, M. A. (2022). First-principles quantum analysis on the role of V-doping on the tuning of electronic and optical properties of spinel oxides MnTi<sub>2</sub>O<sub>4</sub>. *Materials Science & Engineering B*, 278, 115643. doi:10.1016/j.mseb.2022.115643

Nazar, M., Nasarullah, Aldaghfag, S. A., Yaseen, M., Ishfaq, M., Khera, R. A., Noreen, S., & Abdellattif, M. H. (2022). First-principles calculations to investigate structural, magnetic, optical, electronic and

thermoelectric properties of  $X_2MgS_4$  (X= Gd, Tm) spinel sulfides. *Journal of Physics and Chemistry of Solids*, 166, 110719. doi:<u>10.1016/j.jpcs.2022.110719</u>

Patel, P. D., Pandya, J., Shinde, S., Gupta, S. D., & Jha, P. K. (2022). Robust half metallic ferrimagnetic behavior and thermoelectric response of newly discovered Full-Heusler compound Mn<sub>2</sub>SiRh: DFT study. *Materials Today: Proceedings*, 67(6), 939-942. doi:10.1016/j.matpr.2022.07.468

Perdew, J. P., Burke, K., & Ernzerhof, M. (1996). Generalized Gradient Approximation Made Simple. *Physical Review Letters*, 77, 3865. doi:<u>10.1103/PhysRevLett.77.3865</u>

Quiroz, H. P., Calderón, J. A., & Dussan, A. (2020). Magnetic switching control in Co/TiO<sub>2</sub> bilayer and TiO<sub>2</sub>:Co thin films for Magnetic-Resistive Random Access Memories (M-RRAM). *Journal of Alloys and Compounds*, 840, 155674. doi:10.1016/j.jallcom.2020.155674

Rafiq, M. A., Javed, A., Rasul, M. N., Nadeem, M., Iqbal, F., & Hussain, A. (2022). Structural, electronic, magnetic and optical properties of  $AB_2O_4$  (A = Ge, Co and B = Ga, Co) spinel oxides. *Materials Chemistry and Physics*, 257, 123794. doi:10.1016/j.matchemphys.2020.123794

Ravi, S. (2020). Spin transport through silicon using a double perovskite-based magnetic tunnel junction. *Superlattices and Microstructures*, *147*, 106688. doi:<u>10.1016/j.spmi.2020.106688</u>

Saberi, S. H., Baizaee, S. M., & Kahnouji, H. (2014). Electronic structure and magnetic properties of transitionmetal (Y, Zr, Nb, Mo, Tc, Ru, Rh, Pd, Ag and Cd) doped in GaN nanotubes. *Superlattices and Microstructures*, 74, 52-60. doi:10.1016/j.spmi.2014.05.013

Shin, B., Park, I. H., & Chung, C. W. (2006). Inductively coupled plasma reactive ion etching of Co<sub>2</sub>MnSi magnetic films for magnetic random access memory. *Studies in Surface Science and Catalysis*, *159*, 377-380. doi:10.1016/S0167-2991(06)81612-5

Singh, D. J. (1994). *Planewaves, Pseudopotentials and the LAPW Method*. Kluwer Academic, Boston. doi:10.1007/978-1-4757-2312-0

Tran, F., & Blaha, P. (2009). Accurate band gaps of semiconductors and insulators with a semi-local exchange-correlation potential. *Physical Review Letters*, *102*(22), 226401. doi:<u>10.1103/PhysRevLett.102.226401</u>

Xu, M., Zhang, Q., Tan, Q., Zhang, W., Sang, S., Yang, K., & Ge, Y. (2022). A magnetostrictive BaTiO<sub>3</sub>-Fe-Ga&PDMS magnetic field sensor: Research on magnetic detection performance. *Sensors and Actuators: A. Physical*, *335*, 113383. doi:10.1016/j.sna.2022.113383

Zhang, J.-M., Duan, J.-P., Huang, Y.-H., & Wei, X.-M. (2022). Effects of the Tc, Ru, Rh and Cd substitution doping on the structural, electronic, magnetic and optical properties of blue P monolayer. *Thin Solid Films*, 756, 139386. doi:10.1016/j.tsf.2022.139386

Zhang, Z., Sang, L., Huang, J., Chen, W., Wang, L., Takahashi, Y., Mitani, S., Koide, Y., Koizumi, S., & Liao, M. (2020). Enhanced magnetic sensing performance of diamond MEMS magnetic sensor with boron-doped FeGa film. *Carbon*, *170*, 294-301. doi:<u>10.1016/j.carbon.2020.08.049</u>