

Research Article

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## Fabrication and characterization of Ni-based electrodes for improved NiZn battery performance

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### Highlights

- $\beta$ -phase nickel hydroxide as a cathode material is synthesized via a traditional hydrothermal method under various conditions.
- The optimized Ni-based electrode has a high current density of  $220 \text{ mA cm}^{-2}$ .
- NiCoZn composite material is developed for Ni-based electrodes in NiZn batteries.
- NiCoZn electrode in NiZn battery shows excellent capacity of  $192.7 \text{ mAh g}_{\text{active}}^{-1}$ .

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### ABSTRACT

In this study, a hydrothermal method was used to synthesize nickel hydroxide ( $\text{Ni}(\text{OH})_2$ ) powders, which are active materials for use in nickel (Ni) electrodes located in nickel-zinc (NiZn) batteries. X-ray diffraction (XRD), scanning electron microscopy (SEM), zeta potential, Brunauer-Emmett-Teller (BET), and electrochemical characterization were used to characterize the cathode material in the prepared Ni electrodes. These results showed a  $\beta$ -phase  $\text{Ni}(\text{OH})_2$  nanosphere with a well-crystalline structure. The electrochemical test results indicated the Ni electrode has a stable cyclic cycle in the half-cell. Based on the electrochemical performance results, the Ni electrode with  $\text{Ni}(\text{OH})_2$ , which was synthesized at  $70^\circ\text{C}$  for 3h of aging time (Ni\_pH 12\_3h\_70°C), was the best-performing metal oxide. Compared with nickel electrodes, the NiCoZn electrode exhibited high OER (oxygen evolution reaction) and ORR (oxygen reduction reaction) activities because the combination of cobalt and zinc oxides with nickel provides excellent electrolyte access capability and promotes effective ion transfer through the active material. The NiZn battery with NiCoZn electrode showed a high capacity of  $192.7 \text{ mAh g}_{\text{active}}^{-1}$  at  $10 \text{ mA cm}^{-2}$  and cycling durability (after cycling at  $10 \text{ mA cm}^{-2}$  for 70 cycles). Benefiting from the excellent interaction between Ni, Co, and Zn, NiCoZn exhibited high onset potential and current density, suggesting that the NiCoZn electrode is a promising candidate as a high-performance configuration for Ni-based electrodes in NiZn batteries.

**Keywords:** Nickel hydroxide synthesis, Nickel electrode, NiZn batteries, Battery performance

## 1. INTRODUCTION

Recently, interest in alkaline batteries has gained great attention due to their higher specific energy and the high demand for electric vehicles and portable electronic devices [1]. Among alkaline storage batteries, nickel-zinc (NiZn) batteries have been regarded as a promising candidate for energy storage systems due to their theoretical energy density ( $372 \text{ Wh kg}^{-1}$ ), high open circuit voltage (1.75 V), low cost, and availability of materials in nature [2, 3]. NiZn batteries exhibit moderate performance with a power density of  $140\text{-}200 \text{ W kg}^{-1}$  and an energy density of  $55\text{-}85 \text{ Wh kg}^{-1}$  when compared to that of other rechargeable Ni-based batteries (NiCd, NiMH, NiFe batteries, etc.) [2, 4].

Nickel hydroxide ( $\text{Ni(OH)}_2$ ) has gained much interest in Ni-based alkaline batteries due to its structure, high theoretical capacity, and excellent durability in alkaline electrolytes [5].  $\text{Ni(OH)}_2$  possesses a hexagonal layered structure with  $\alpha$  and  $\beta$  phases. While  $\alpha\text{-Ni(OH)}_2$  with an isostructural structure has stacked  $\text{Ni(OH)}_{2-x}$  layers intercalated with different anions or water molecules,  $\beta\text{-Ni(OH)}_2$  with a brucite-like structure contains stacked layers without intercalated species [6, 7].  $\alpha\text{-Ni(OH)}_2$  has higher theoretical specific capacity ( $433 \text{ mAh g}^{-1}$ ) than that of  $\beta\text{-Ni(OH)}_2$  ( $289 \text{ mAh g}^{-1}$ ). However,  $\beta\text{-Ni(OH)}_2$  is widely used in rechargeable batteries since  $\alpha$ -phase is very unstable in water and alkaline media.

In a charge/discharge cycle,  $\beta\text{-Ni(OH)}_2$  is converted to  $\beta\text{-NiOOH}$  and the  $\alpha$ -phase immediately changes to  $\gamma\text{-NiOOH}$ , resulting in the swelling of electrode volume by absorbing the alkaline electrolyte due to the high expansion of the  $\gamma$ -phase, leading to reduce discharge potential [8-12].

NiZn batteries have the main challenges for large-scale battery applications such as Zn electrode shape change during the charge/discharge cycles and irreversibility of Ni electrodes [7, 13, 14]. An available approach is to eliminate the swelling of Ni electrodes during the charge/discharge cycles. The swelling phenomenon occurs when  $\beta\text{-Ni(OH)}_2$  phase converts into  $\gamma\text{-NiOOH}$ , leading to high volume change and rapid capacity loss during charge/discharge cycles [15]. Many studies have been carried out to minimize the swelling phenomenon when increasing the number of charge/discharge cycles. Another effective one is to construct well-defined Ni-based nanostructures as electrodes. It is established that  $\text{Ni(OH)}_2$  particles and  $\text{Ni(OH)}_2$  composites with various morphologies such as flowerlike [16], nanotubes [17], nanorods [18], mesoporous films [19], nanosheets [20], spheres [21-23], nanoribbons [24], and stacks of pancakes [25] can be

synthesized by using different methods such as chemical precipitation [26, 27], hydrothermal [28], solvothermal [29, 30], sonochemical [31] and electrodeposition methods [32].

There are several studies in the literature on the synthesis of  $\text{Ni}(\text{OH})_2$  with good crystallinity, uniformity, high surface area, and uniform particle size distribution to aid the higher electrochemical performance of Ni-based electrodes. Liu and co-workers examined the electrochemical performance of Ni-based electrodes with prepared  $\text{Ni}(\text{OH})_2$ . It was reported that the nickel electrode with 10% nano-sized  $\text{Ni}(\text{OH})_2$  addition showed higher electrochemical performance compared to the common micro-sized Ni-based electrode [33].

In addition to the change in the morphology and crystallinity of  $\text{Ni}(\text{OH})_2$ , by using active materials, it would be possible to improve the performance and stability of the capacity of Ni electrode during charge/discharge cycles due to providing improved mass transfer kinetics in an alkaline electrolyte and reactions of transition metal ions including  $\text{OH}^-$  [34]. It is widely known that  $\text{Ni}(\text{OH})_2$  has poor conductivity because of its semi-conductive nature [35]. The effective approaches to improve these issues have been studied such as the use of various additives, such as bismuth [36], ytterbium [37], yttrium [38], cobalt [33, 35, 39-41], zinc oxide [42], cadmium oxide [43, 44], etc., in Ni-based electrode due to inhibiting the  $\gamma$ - $\text{NiOOH}$  formation by altering interlayer bonding forces of the nickel crystal lattice. Cobalt is considered a promising candidate as an additive in the area of Ni-based batteries, to provide an enhancement of nickel electrode conductivity, a better rechargeable by increasing the potential for OER, and/or reducing the Ni electrode working electrode potential, an enhancement of nickel electrode mechanical resistance, and a reduction of  $\gamma$ - $\text{NiOOH}$  growth at the Ni electrode during charging that leads to swell and blister the Ni electrode [35, 39, 45]. There are various studies to focus on the inclusion of cobalt as an additive to Ni-based electrodes. For example, Yuan et al. examined the effect of cobalt powder on the electrochemical characteristics of Ni electrodes with co-precipitated  $\text{Ni}_{1-x}\text{Zn}_x(\text{OH})_2$  [35]. It was reported that the discharge capacity and stability were improved when compared to the absence of cobalt powder. The use of cobalt in the Ni-based electrode was investigated by Armstrong et al. [46]. The results showed that the cobalt addition increases the oxygen over potential and the effect of iron poisoning and promotes the utilization of the nickel species. Mao and coworkers examined the influence of cobalt powders with different diameters on the inner pressure of Ni/MH batteries during the charging process [47]. It was pointed out that the cell with the finest cobalt indicated a decline in inner pressure and a long cycle life. Cobalt was used in the study by Sood and groups to examine

the effect on the electrochemical performance of Ni-based electrodes in lineated electrolytes [48]. It was reported that the addition of cobalt in Ni-based electrodes played an effective role in its performance and lifetime.

Herein, we present an investigation of the synthesis, characterization, and electrochemical properties of Ni(OH)<sub>2</sub>. We attempt to apply different methods to synthesize Ni(OH)<sub>2</sub> and to compare the electrodes based on their particle size and crystallinity. The purpose of this study is to determine the optimum synthesis conditions (temperature, reaction pH, and aging time) of Ni(OH)<sub>2</sub> powder to improve the electrochemical performance of Ni-based electrodes. We also compare the electrochemical performances of Ni electrodes fabricated by the synthesized Ni(OH)<sub>2</sub> powders. We synthesized NiM oxide (M: Metal; cobalt (Co), or cobalt-zinc (Co-Zn)) powders by the hydrothermal method to investigate the activity of the combination of Co and Zn with Ni. We also report the electrochemical performance of Ni electrodes by combining Co and Zn with Ni in a NiZn battery. Excellent behaviors in terms of high current density at constant voltage and good cycle stability make the nanosphere NiCoZn (Ni<sub>0.925</sub>Co<sub>0.05</sub>Zn<sub>0.025</sub>(OH)<sub>2</sub>) powders a member of the most attractive and promising candidates for Ni-based electrode materials in NiZn batteries.

## 2. EXPERIMENTAL

### 2.1. Synthesis of Ni(OH)<sub>2</sub> Powder

Nickel (II) hydroxide powder synthesis was carried out via hydrothermal method under various conditions (pH, temperature, and aging time). 2 M NiSO<sub>4</sub>·6H<sub>2</sub>O (≥99%, Sigma Aldrich) solution and 4 M NaOH (≥98%, Sigma Aldrich) solution were mixed when adjusting pH (9.5 or 12). To control reaction pH, 10 M NH<sub>4</sub>OH (≥99%, Sigma Aldrich) buffer solution was added during the experiment. The reaction temperature was adjusted from 50 to 80°C for optimum precipitation. The solution was aged at a desired reaction temperature from 3h to 24h. The liquid phase was vacuum-filtered and separated from the precipitate. To remove excess ions such as SO<sub>4</sub><sup>2-</sup>, OH<sup>-</sup> and NH<sub>4</sub><sup>+</sup>, the precipitate was washed three times with deionised water. The precipitate was dried at 70°C for 15 hours in a vacuum. To observe the effect of additives, cobalt sulfate (CoSO<sub>4</sub>·7H<sub>2</sub>O, ≥99%, Sigma Aldrich) and zinc sulfate (ZnSO<sub>4</sub>·7H<sub>2</sub>O, ≥99%, Sigma Aldrich) were used during the Ni(OH)<sub>2</sub> powder synthesis. Additionally, 2 M NiSO<sub>4</sub>·6H<sub>2</sub>O solution, 0.03 M CoSO<sub>4</sub>·7H<sub>2</sub>O solution and/or 0.06 M ZnSO<sub>4</sub>·7H<sub>2</sub>O solution, and 4 M NaOH (≥98%, Sigma Aldrich) solution were mixed at pH 12 and 70°C. Two different solutions were aged for 3h. The same experimental procedure was carried out for synthesis of NiCo and/or NiCoZn powders. Three different kinds

of mixture powders were prepared which are Ni/Co:95/5 (wt.%) and Ni/Co/Zn:92.5/5/2.5 (wt.%). Synthesized Ni(OH)<sub>2</sub> powder samples are denoted as shown in Table 1.

**Table 1.** Identification of Ni(OH)<sub>2</sub> samples synthesized in this study

Sample Code	Temperature (°C)	Aging time (h)	pH
Ni_pH 9.5_3h_50°C	50	3	9.5
Ni_pH12_3h_50°C	50	3	12
Ni_pH12_3h_60°C	60	3	12
Ni_pH12_3h_70°C (Ni <sup>a</sup> )	70	3	12
Ni_pH12_3h_80°C	80	3	12
Ni_pH12_6h_70°C	70	6	12
Ni_pH12_12h_70°C	70	12	12
Ni_pH12_24h_70°C	70	24	12
NiCo <sup>b</sup>	70	3	12
NiCoZn <sup>c</sup>	70	3	12

a: Ni:100 (wt.%), b: Ni/Co:95/5 (wt.%), c: Ni/Co/Zn:92.5/5/2.5 (wt.%)

## 2.2. Nickel Electrode Preparation

A 1 cm x 1 cm piece of nickel foam (Nanografi A.S.) as a substrate was used as a current collector. To prepare the pasted nickel electrodes, 85 wt.% as-prepared Ni(OH)<sub>2</sub> or NiM (M: Metal, Co or CoZn) hydroxide, 10 wt.% carbon black (Ogan Ocali, A.S.) were used and mixed with a certain amount of 5 wt.% carboxymethyl cellulose (Ogan Ocali, A.S.) solution to obtain a homogeneous paste. The electrode paste was mixed mechanically for about 5 min. The paste was poured into a nickel foam. It was dried at 80 °C for 4 h. Finally, the electrodes were pressed at 10 MPa for 5 min using a hydraulic press (Erdes HP2 hydraulic press) to provide high electrical contact between the nickel foam and the synthesized active materials.

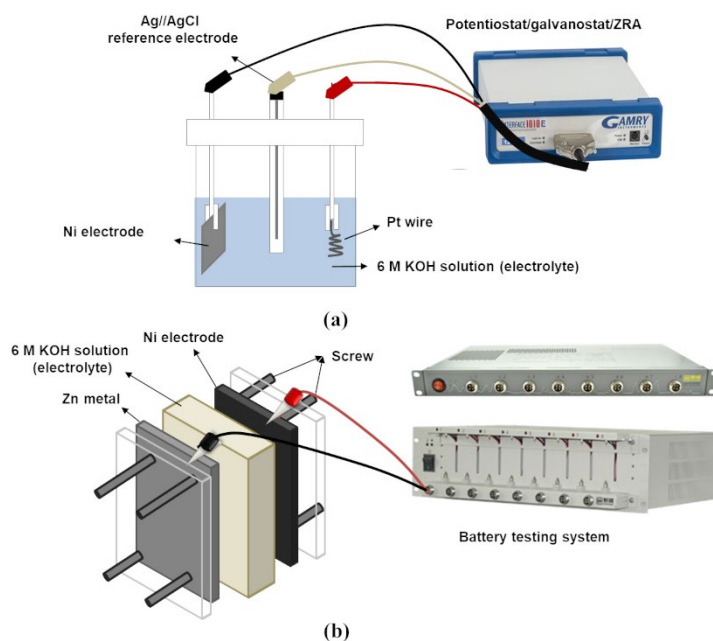
## 2.3. Characterization

The structures of synthesized Ni(OH)<sub>2</sub> powders were determined by X-ray diffraction (XRD) analysis using a Phillips<sup>TM</sup> Xpert diffractometer. Data was collected from 5-90° in the 2θ range at variable speed (0.336 sec<sup>-1</sup>). The surface morphology and element distribution of powders in Ni electrodes was studied with Scanning Electron Microscopy (SEM, FEI Quanta250) with energy-dispersive X-ray spectroscopy (EDS). Brunauer-Emmett-Teller (BET) analysis (Quantachrome Autosorb) using a nitrogen adsorption/desorption process at 67 K was performed to determine the

surface area and pore size distribution of each sample. A NanoPlus zeta/nanoparticle analyzer was used to investigate the zeta potential and size distribution of each sample which was dispersed in deionized water. To measure the tap density of Ni(OH)<sub>2</sub> samples, samples were loaded into a cylinder and tapped 10 times. The weight of the empty cylinder, the powder-filled cylinder, and the volume of Ni(OH)<sub>2</sub> were recorded. Therefore, the tap density was determined as follows: powder mass divided by the volume of the cylinder occupied by the powder. Elemental analysis of the amount of Ni, Co and Zn in NiCo and NiCoZn powders was determined from inductively coupled plasma-optical emission spectroscopy (ICP-OES, Agilent 5110).

## 2.4. Electrochemical Measurements

Electrochemical performance was measured by using a three-electrode potentiostat/galvanostat/ZRA (Gamry model 22162). During electrochemical measurements, a 3-electrode cell with an as-prepared Ni electrode as a working electrode, a Pt wire as the counter electrode, and Ag//AgCl as a reference electrode were used, as shown in Figure 1(a).



**Figure 1.** Schematic illustration for (a) half cell (b) full cell experimental set up.

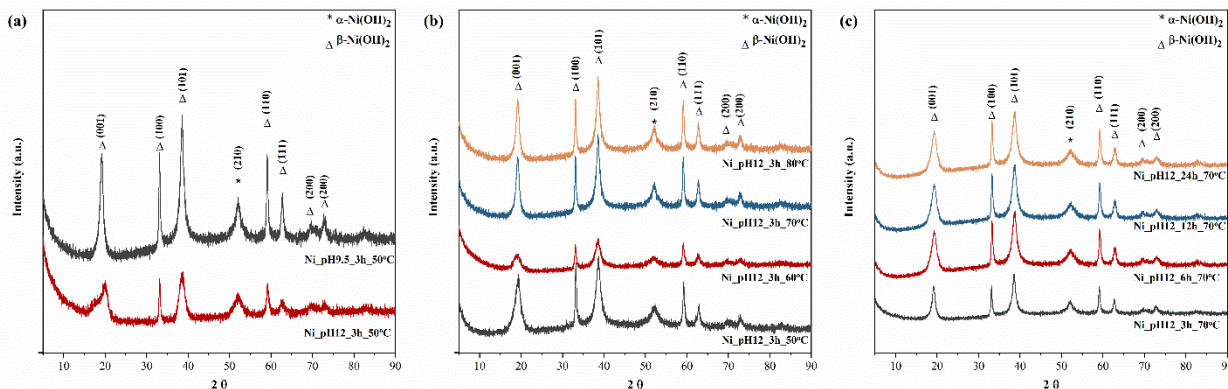
The electrochemical behavior of Ni(OH)<sub>2</sub> was studied by cycling voltammetry (CV) and linear sweep voltammetry (LSV) between 0 and 0.8 V at a scan rate of 10 mVs<sup>-1</sup> in 6M KOH as an electrolyte at 25 ± 2 °C. The NiZn batteries were constructed using a zinc metal plate and a 6 M KOH electrolyte with an excess weight of zinc utilized at the anode. Long-term cycling of the

NiZn battery was accomplished on a battery testing system (NEWARE, CT-4008Tn-5V6A-S1, China), as seen in Figure 1 (b). The NiZn battery was cycled using a constant current/constant voltage (CCCV) charging process at a charge/discharge current of 10/20 mA cm<sup>-2</sup>.

### 3. RESULTS AND DISCUSSION

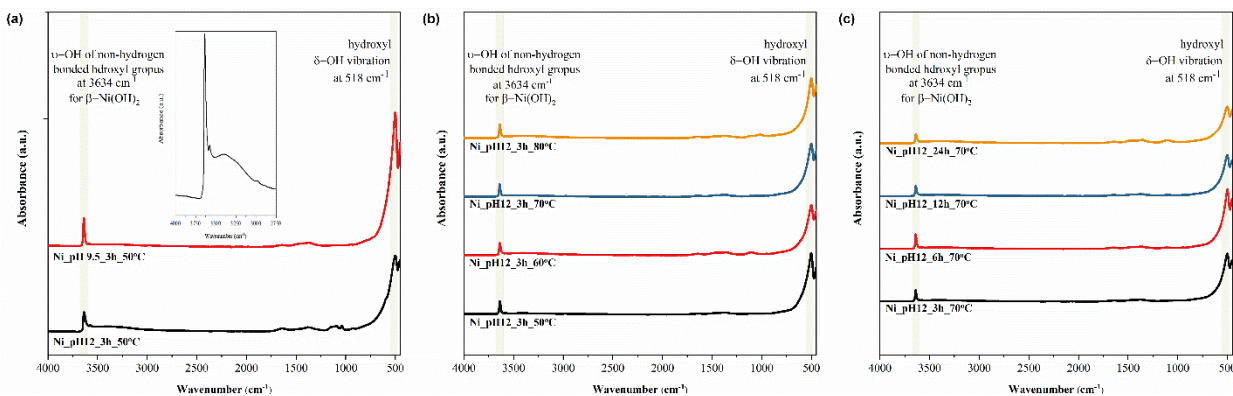
#### 3.1. Characterization of Synthesized Ni(OH)<sub>2</sub>

Figure 2 shows the influence of synthesis conditions (reaction pH, temperature, and aging time) on the crystallinity of the Ni(OH)<sub>2</sub>, as revealed by the XRD patterns. The characteristic peaks were consistent with both two known phases, β-Ni(OH)<sub>2</sub> and α-Ni(OH)<sub>2</sub>. This observation indicated that these samples had typically non pure phases [11, 33, 49, 50]. XRD peaks corresponding to (001), (100), (101), (110), (111), (200), (201), (202), and (210) planes of Ni(OH)<sub>2</sub> were given at 2θ values of 19.3°, 33.1°, 38.6°, 59.1°, 62.8°, 69.4°, 72.8°, 82.7° and 52.8°, respectively, confirming with the standard values (JCPDS file no.14-0117 and 380715). As shown in Figures 2 (a-c), the difference in diffraction intensity between the samples is ascribed to the presence of their difference in microstructure and morphology. The peaks of (001) and (101) were broadened in samples of Ni\_pH 9.5\_3h\_50°C and Ni\_pH12\_3h\_50°C, meaning that there is microdistortion in their structures and the reduction of their crystallite sizes. The control of the crystallinity and chemical structure (morphology, degree of crystallinity, crystalline disorder, crystallite size and crystal growth orientation, etc.) by variation of the preparation conditions such as pH and temperature etc., has been of great interest in the improvement of the electrochemical performance of nickel hydroxide [51-54]. This has a profound and often unpredictable effect on the reversible charge storage capacity of Ni-based batteries. As seen in Figure 2(a), the peaks corresponding to the XRD pattern of Ni\_pH 9.5\_3h\_50°C are broader than those of Ni\_pH 12\_3h\_50°C. This indicates that Ni\_pH 9.5\_3h\_50°C is poorly crystallized. Increasing the precipitation pH leads to a decrease in the crystallite size and an increase in the degree of ordering and crystallinity. Figure 2(b) shows that the XRD peak of the (101) plane for Ni\_pH 12\_3h\_70°C was higher intense than the (001) plane, confirming that the growth of microstructures were observed along the (001) plane. The small size of the crystallites and crystal growth orientation can result in the high chemical proton diffusion coefficient, which leads to the excellent electrochemical performance of the Ni-based electrodes.



**Figure 2.** XRD patterns of Ni(OH)<sub>2</sub> prepared under different conditions (a) pH (b) temperature (c) aging time.

The synthesis of Ni(OH)<sub>2</sub> was confirmed by FTIR results showing the presence of non-hydrogen bonded νO–H vibration groups for β-Ni(OH)<sub>2</sub>, as seen in Figure 3. High intense stretch at 3634 cm<sup>-1</sup> shows the presence of non-hydrogen bonded νO–H vibration groups for β-Ni(OH)<sub>2</sub> [51]. The broad peak at 3456 cm<sup>-1</sup> was found to correspond to the hydroxyl groups in α-Ni(OH)<sub>2</sub> only for the Ni\_pH 9.5\_3h\_50°C sample, confirming that they were intercalated water molecules in the interlamellar spaces of Ni(OH)<sub>2</sub> in α-phase nickel hydroxide. The sharp and narrow band at 512 cm<sup>-1</sup> corresponds to the νNi–OH stretching vibration [55-58].

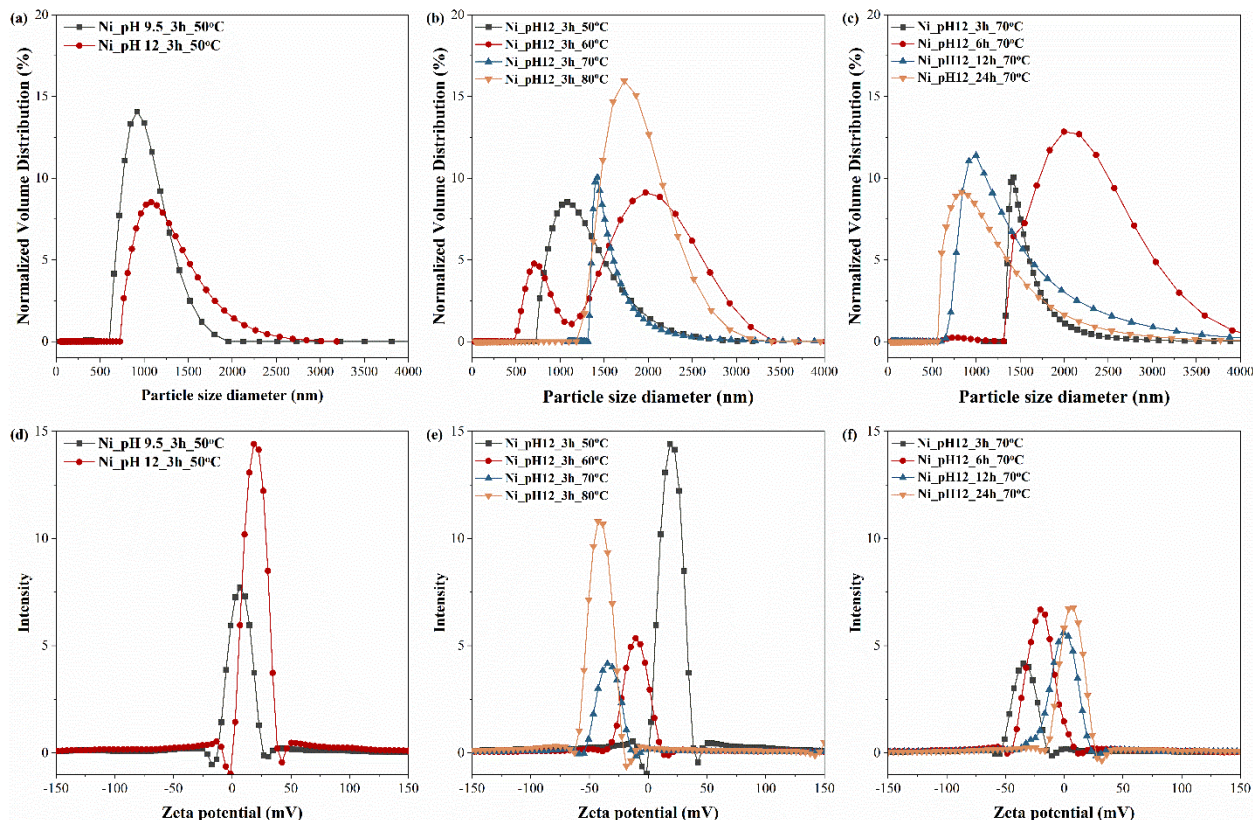


**Figure 3.** FTIR analysis of all Ni(OH)<sub>2</sub> samples.

Figure 4 represents the particle size distribution and zeta potential of all Ni(OH)<sub>2</sub> samples. Figures 4(a-c) show the particle size distribution of Ni(OH)<sub>2</sub> to understand how operating parameters (reaction pH, temperature, and aging time) affect the particle size of samples. It was found that particle size distribution comparisons were best made for groups of samples derived from different temperatures and aging times. There was a smaller variation in particle size (550-2000 nm) for the Ni\_pH9.5\_3h\_50°C sample when compared to Ni\_pH 12\_3h\_50°C. The particle size range of the



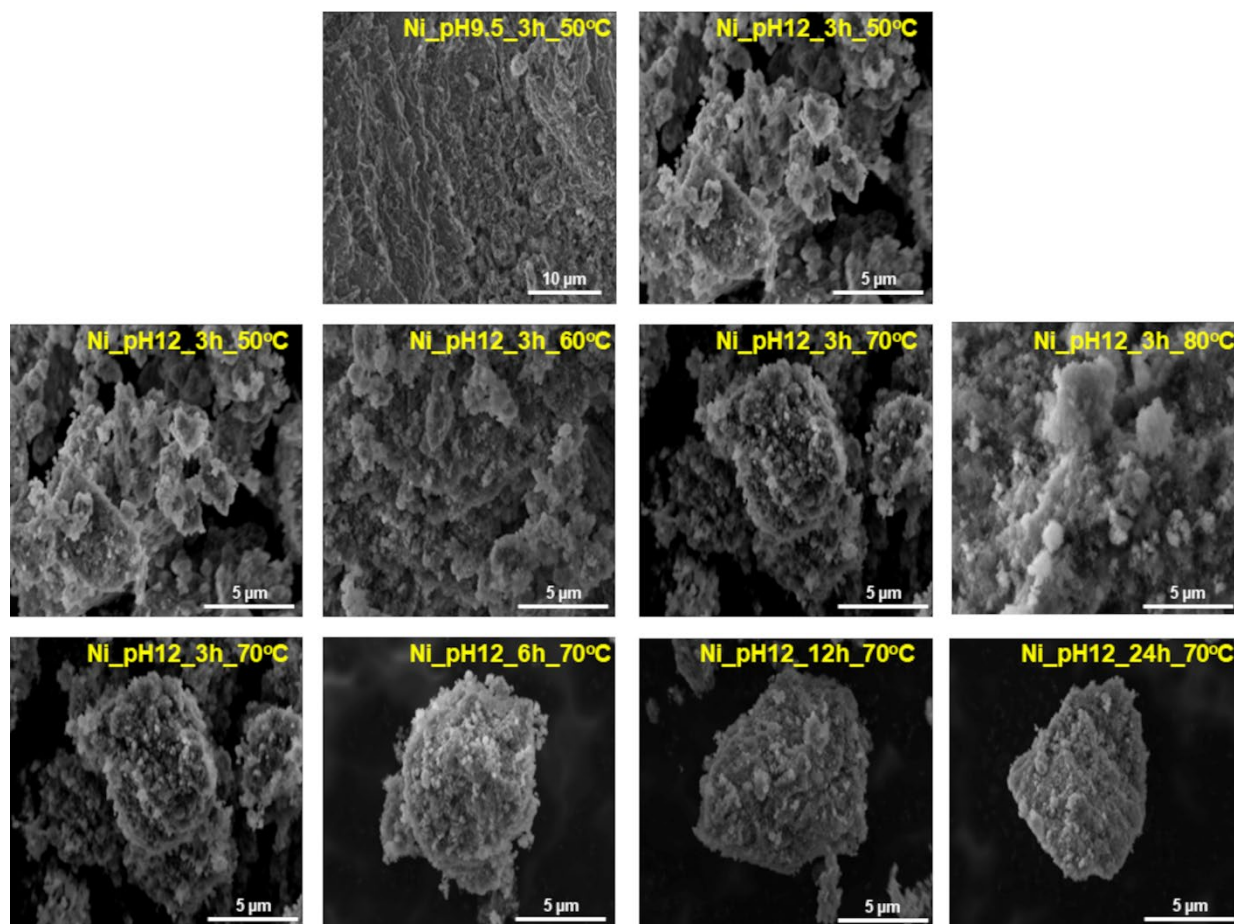
Ni\_pH12\_3h\_70°C sample was found as 1250 nm and 2000 nm. Particle size for Ni\_pH12\_3h\_60 C was distributed in two regions (500-1150 nm and 1150-3500 nm) with low percentage counts (<10%). A similar trend in particle size distribution with temperature change was observed as expected from the zeta potential results. The effect of aging time was found to be due to a difference in the population profile of Ni\_pH12\_3h\_70°C (1250-2000 nm), Ni\_pH12\_6h\_70°C (1250-4000 nm), Ni\_pH12\_12h\_70°C (650-3500 nm) and Ni\_pH12\_24h\_70°C (550-2500 nm). The reason for this is that the particle size of Ni(OH)<sub>2</sub> decreases in case of an increment of aging time. During the aging time as ripening, unstable  $\alpha$ -Ni(OH)<sub>2</sub> particles gradually dissolve in ammonium solution, leading to the growth of larger  $\beta$ -Ni(OH)<sub>2</sub> particles since  $\alpha$ -Ni(OH)<sub>2</sub> particles with a small number of hydroxyl ions (OH<sup>-</sup>) compared to  $\alpha$ -Ni(OH)<sub>2</sub> particles [59, 60]. The sample of Ni\_pH12\_3h\_70°C tended to be more organized than the others due to the narrowest population profile. The zeta potential shows the surface charge state of particles in the slurry, implying the type of electrostatic interactions or the control of suspension stability. The traditional tendency of how zeta potential affects the load stability is that zeta potentials higher than +25 mV or lower than -25 mV exhibit well-dispersed, stable, deflocculated particles in slurry [59, 61]. The load stability depends on the region, higher than +30 mV or lower than -30 mV [62]. The results shows that the trends of the zeta potential of Ni(OH)<sub>2</sub> changed with process parameters (reaction pH, temperature, and aging time) in ultrapure water at the same pH (8-8.5 of pH), as shown in Figures 4 (d-f). The increase in reaction pH led to an enhancement of the electrostatic charges on the surface of particles. Thus, the sample of Ni\_pH12\_3h\_50°C was more stable than the sample of Ni\_pH9.5\_3h\_50°C. Among the Ni(OH)<sub>2</sub> samples synthesized at different temperatures, the Ni\_pH\_12\_3h\_70°C showed the maximum surface charge, and its zeta potential was close to -35 mV, as seen in Figure 4 (e). The stability of the load was considered to be excellent. The samples synthesized with various aging times might be considered as moderately stable since the zeta potentials were in the range of -40 to -4 mV (Figure 4 (f)).



**Figure 4.** Particle size distribution (a-c) and zeta potential distribution (d-f) of all Ni(OH)<sub>2</sub> samples.

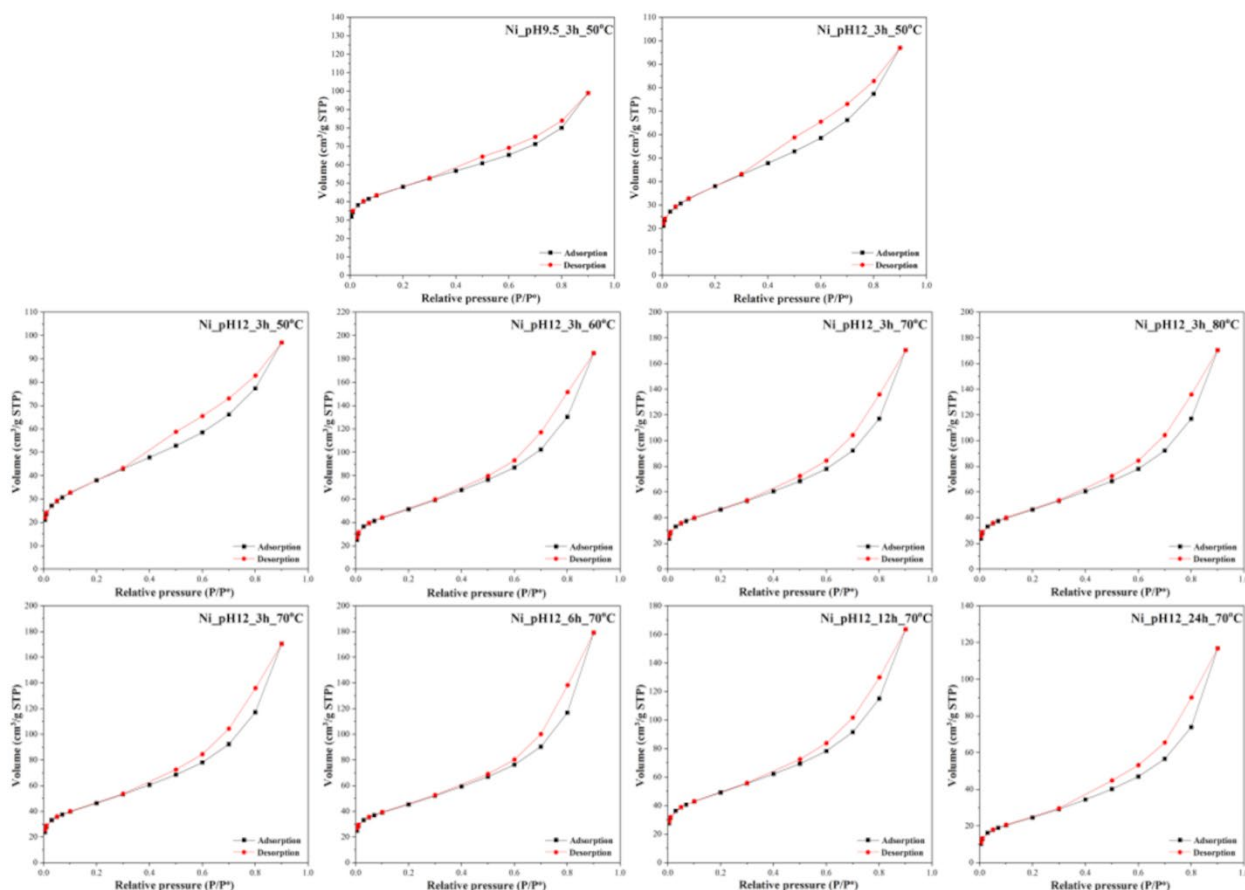
We expect that the morphology and surface microstructure of Ni(OH)<sub>2</sub> samples play a significant role on the electrochemical performance of NiZn batteries. The nano-particle morphologies of Ni(OH)<sub>2</sub> samples synthesized at different conditions (pH, temperature, and aging time) were investigated by SEM, as seen in Figure 5. The Ni\_pH9.5\_3h\_50°C showed agglomerated nanoparticles with nanosheet stacks. Figure 5 shows that the lower pH led to densely packed layers. The change in reaction pH could directly be related to the crystallinity of Ni(OH)<sub>2</sub> polymorphs rather than morphology [25]. The increase in reaction pH led to weaker agglomeration and consisted of lots of finer particles as well as the formation of less organized crystallites. Like the previous study in the literature, nano Ni(OH)<sub>2</sub> samples with plate-like shapes were observed when increasing reaction pH (pH>10) [25, 63]. The formation of Ni(OH)<sub>2</sub> with a larger spherical shape was obtained at higher reaction temperatures (>50°C). The grain sizes of samples range from 5 to 35 μm. This observation might be due to the as-synthesized Ni(OH)<sub>2</sub> is prevented from growing as the reaction temperature increases [64]. By comparison, the Ni\_pH12\_3h\_70°C sample exhibited many spherical nanoparticles with well-distributed size having a diameter of 2.5-5 μm (Figure 5 (e)). The effect of aging time on the synthesized Ni(OH)<sub>2</sub> morphology was reported in the literature [65]. From Figures 5 (f-i), the products consisted of around 5 μm spherical particles.

The Ni(OH)<sub>2</sub> morphologies did not remarkably change with the increase in aging time from 3 h to 24 h, confirming the XRD analysis and particle size distribution results.



**Figure 5.** SEM images of all Ni(OH)<sub>2</sub> samples synthesized under various conditions.

Nitrogen adsorption-desorption isotherms were performed to examine the structural characteristics, pore size distribution (insets), and surface area of all Ni(OH)<sub>2</sub> powders, as shown in Figure 6. All Ni(OH)<sub>2</sub> synthesized under different conditions showed the type IV isotherms with H3 hysteresis in the relative pressure ( $P/P^0$ ) range of 0.005-0.9 according to IUPAC classification, associated with non-rigid aggregates of mesoporous particles that produce slit pores. The behavior of this profile was consistent with capillary condensation in the structures due to the type of pore of solid material [66].



**Figure 6.** Nitrogen adsorption-desorption isotherms and pore size distribution curve (insets) of all Ni(OH)<sub>2</sub> samples.

Table 2 shows pore volume, the BET surface area, and tap density of all Ni(OH)<sub>2</sub> samples. The tap density of sample Ni\_pH9.5\_3h\_50°C was measured to be 0.80 g/cm<sup>3</sup>, which was higher than that of Ni\_pH12\_3h\_50°C (0.56 g/cm<sup>3</sup>). The reaction temperature improved the tap density varied from 0.56 to 0.8 g/cm<sup>3</sup>, which might be attributed to the structure of Ni(OH)<sub>2</sub> and the formation of agglomeration of particles. Moreover, with increasing aging time from 3 to 24h, the tap density of samples increased from 0.68 to 1.02 g/cm<sup>3</sup>. The tap densities of the Ni(OH)<sub>2</sub> samples in this study were lower when compared with those reported in the literature (2.82 g/cm<sup>3</sup>) due to their irregular particle morphology and broad particle size distribution [24, 34]. The average pore diameter of these samples was measured to be around 65 Å. All Ni(OH)<sub>2</sub> samples presented large BET surface areas and high pore volumes, which might be due to the size of the precursor cations. The results showed that the pore volumes of all samples were in a range of 0.125-0.289 cm<sup>3</sup>g<sup>-1</sup>, attributed to surface areas between 132.94 and 178.32 cm<sup>2</sup>g<sup>-1</sup> (in Table 2). The Ni\_pH12\_3h\_70°C sample had the highest surface area (178.32 cm<sup>2</sup>g<sup>-1</sup>) and pore volume (0.289 cm<sup>3</sup>g<sup>-1</sup>). The effective material

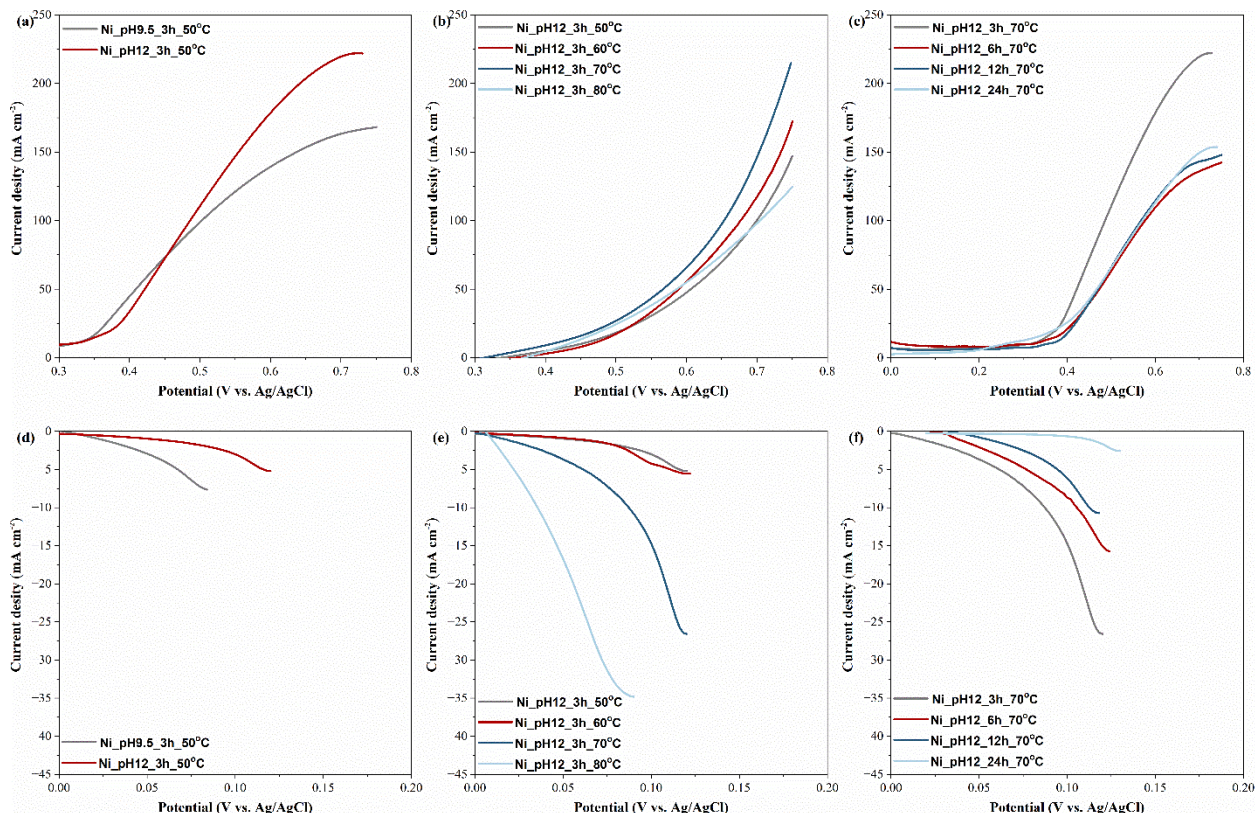
and higher current density have been realized by improving conductivity and reducing polarization via surface modification with Ni(OH)<sub>2</sub> samples. Therefore, this observation showed that the Ni\_pH12\_3h\_70°C sample could be a promising candidate as a material in Ni electrodes for the NiZn battery system due to its better BET surface area and tap density.

**Table 2.** Tap density, BET surface area, pore diameter, and pore volume and particle size for all Ni(OH)<sub>2</sub> samples in different conditions.

	Tap density, g cm <sup>-3</sup>	BET specific surface areas, m <sup>2</sup> g <sup>-1</sup>	Average pore diameters, Å	Pore volume, cm <sup>3</sup> g <sup>-1</sup>	Particle size, nm
Ni_pH 9.5	0.80	160.14	45.11	0.131	779.70
Ni_3h_50°C	0.56	132.94	51.67	0.125	911.23
Ni_3h_60°C	0.61	183.38	60.26	0.257	1111.50
Ni_3h_70°C	0.68	184.60	62.31	0.279	1468.62
Ni_3h_80°C	0.80	133.83	62.24	0.211	1602.85
Ni_3h_70°C	0.68	184.60	62.31	0.279	1468.62
Ni_6h_70°C	0.71	160.79	66.28	0.259	1427.16
Ni_12h_70°C	0.81	178.32	66.19	0.289	844.31
Ni_24h_70°C	1.02	133.83	62.24	0.211	659.99

### 3.2. Electrochemical Performance Ni Electrode with Synthesized Ni(OH)<sub>2</sub>

The electrochemical performance of the as-prepared Ni electrodes was investigated as a function of pH, temperature and aging time. Half-cell oxygen reduction reaction (ORR) and oxygen evolution reaction (OER) CV and LSV comparisons between electrodes with Ni(OH)<sub>2</sub> are shown in Figure 7. During the ORR electrochemical activity measurement, the Ni\_pH9.5\_3h\_50°C electrode exhibited a slightly lower onset potential ( $E_{\text{onset}}$ ) of 31 mV compared with the Ni\_pH12\_3h\_50°C (32 mV), as seen in Figure 6. Ni\_pH12\_3h\_70°C electrode showed a high onset potential compared to other samples prepared at different temperatures. This demonstrated that the electrochemical activity of ORR affected the Ni(OH)<sub>2</sub> structure properties. Additionally, there was no influence of aging time for Ni(OH)<sub>2</sub> synthesis on the electrochemical performance except for 3h aging time.



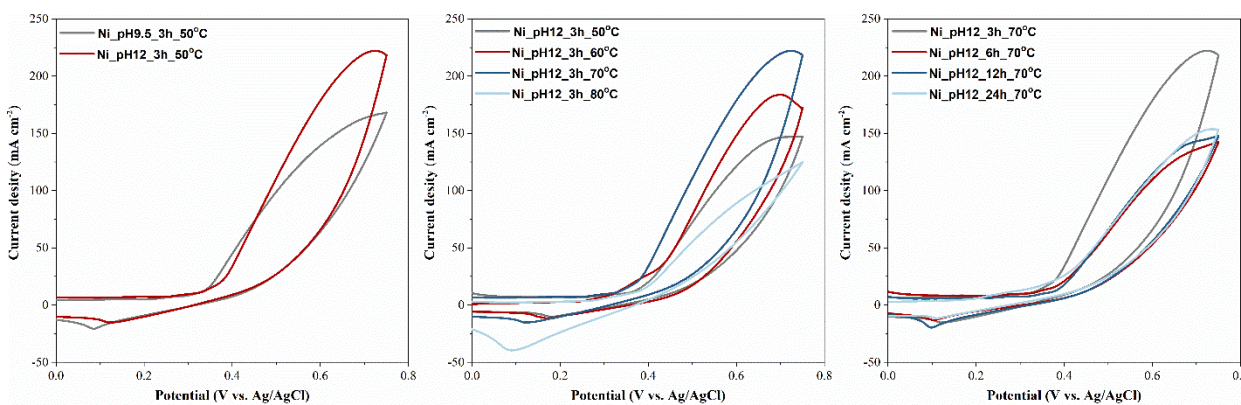
**Figure 7.** LSV polarization curves measured in 6M KOH of all Ni electrodes for OER and ORR.

During ORR measurements, the Ni\_pH12\_3h\_70°C electrode exhibited an onset potential of 37 mV which was more positive compared to other Ni electrodes, (see Table 3). Therefore, Ni\_pH12\_3h\_70°C electrode showed high electrocatalytic activity for ORR. An improvement of electrochemical activity on the Ni\_pH12\_3h\_70°C electrode was indicated by its slightly smaller  $E_0$  (31 mV) than other Ni electrodes. This might suggest that the spherical nanoparticles of Ni(OH)<sub>2</sub> for Ni\_pH\_12\_3h\_70°C showed more electrochemical activity. Therefore, the Ni\_pH12\_3h\_70°C electrode with high onset potentials and low  $E_0$  is a promising candidate as an electrode material for the NiZn battery system.

**Table 3.** OER and ORR onset potentials of all Ni electrodes for ORR and OER and E<sub>0</sub>

	Onset potential, mV		E <sub>0</sub> , mV
	OER	ORR	
Ni_pH 9.5_3h_50°C	32	31	35
Ni_pH 12_3h_50°C	38	32	33
Ni_pH 12_3h_50°C	35	31	32
Ni_pH 12_3h_60°C	28	36	36
Ni_pH 12_3h_70°C	37	37	31
Ni_pH 12_3h_80°C	34	35	34
Ni_pH 12_3h_70°C	37	37	31
Ni_pH 12_6h_70°C	25	30	33
Ni_pH 12_12h_70°C	38	32	34
Ni_pH 12_24h_70°C	16	27	33

The oxidation and reduction peaks of Ni electrodes curves at various reaction conditions were observed in cyclic voltammetry (CV), as shown in Figure 8. This observation showed that the CV area changed with the reaction conditions during Ni(OH)<sub>2</sub> synthesis, where the maximum value occurs at 70°C during 3h of aging time. This suggests that the excellent electrochemical activity of Ni(OH)<sub>2</sub> was found at this condition. The equivalent catalytic behavior in KOH electrolyte for NiOH oxidation on the Ni electrode (Eq. 1) may be as follows:

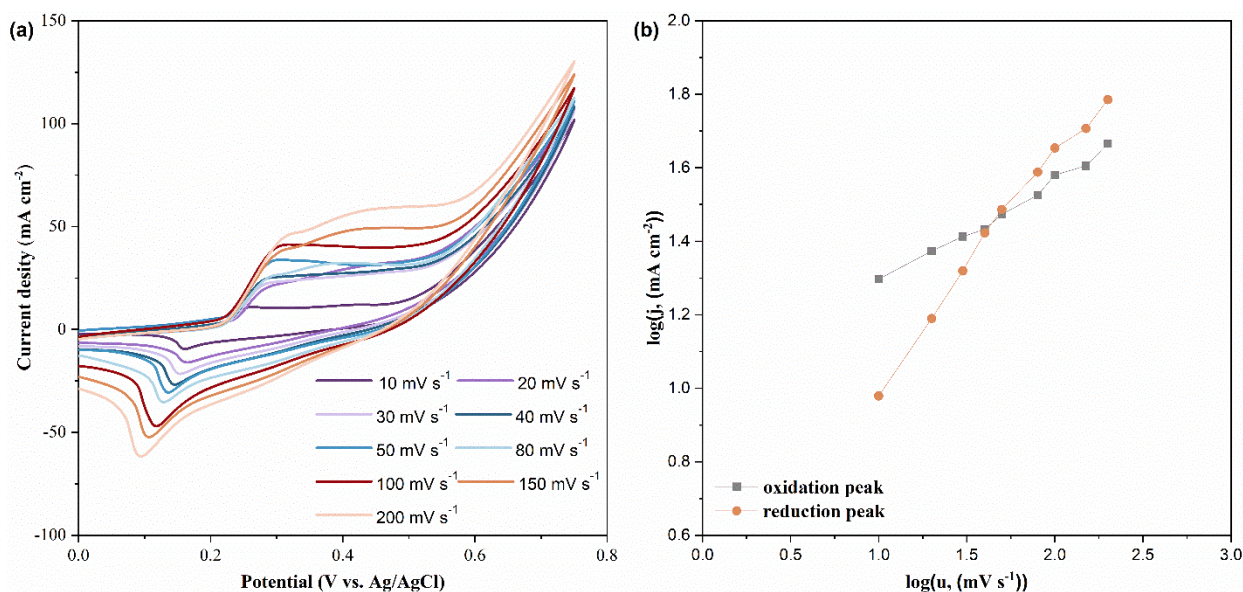


**Figure 8.** CV curves for measured in 6M KOH of all Ni electrodes with as-fabricated Ni(OH)<sub>2</sub>.

To understand the electrochemical kinetics of the Ni\_pH12\_3h\_70°C electrode, its performance parameters were investigated, as seen in Figure 9. The CV curve showed the increase in current densities with the increase in the sweep rate, implying excellent electrochemical performance of the Ni\_pH12\_3h\_70°C electrode. The peak associated with OER and ORR could be observed over a wide potential range from 0 to 0.8 V. The relationship between current density and scan rate can be calculated using the equation (Eq. 2);

$$\log(j) = b \log(v) + \log(a) \quad (2)$$

where  $j$  represents current density,  $v$  is the scan rate. The constant value is represented as  $a$  and  $b$  is a slope for charge-transfer coefficient. The oxidation and reduction peak current densities for Ni\_pH12\_3h\_70°C electrodes were found to be  $v^{0.28}$  and  $v^{0.62}$ , respectively. This observation showed that the reduction and oxidation reactions at the Ni\_3h\_70°C electrode are diffusion-controlled processes due to near 0.5 of the “ $b$ ” value [63].



**Figure 9.** (a) Effect of the voltammetric scan rate on current density for Ni\_pH12\_3h\_70°C (b) corresponding relation of the logarithm of peak current and the logarithm of the scan rate for OER and ORR.

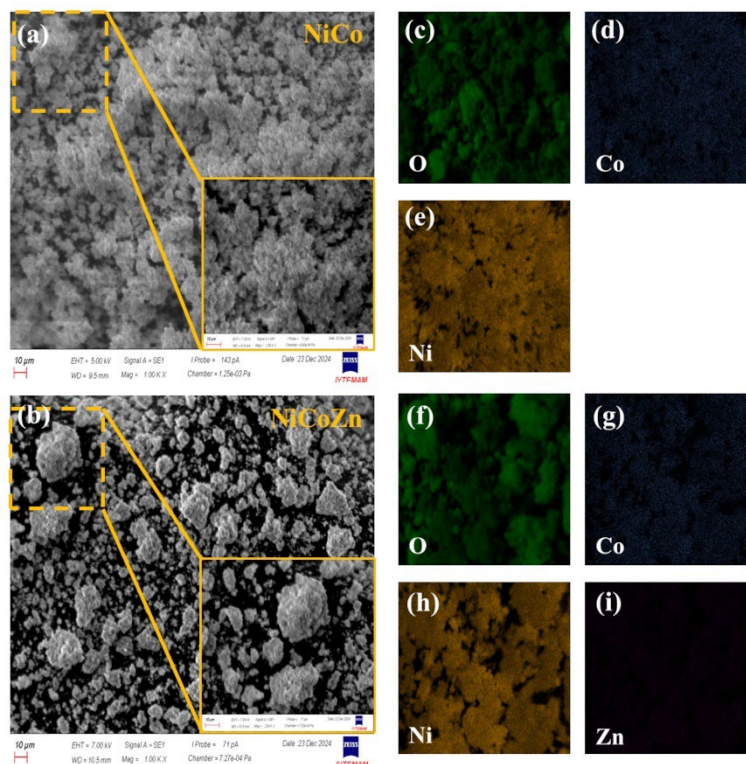
### 3.3. Addition of Transition Metals (Co And Zn) as Active Materials for Ni Electrode

There has been considerable interest in transition metals to enhance the performance of Ni-based electrodes due to their cost effectiveness and ready availability [67]. Nickel hydroxides are



employed as an active materials in Ni-based electrodes because nickel in alkaline solution is more conductive to the desorption of  $\text{OH}^-$ . However, nickel hydroxides usually face critical challenges (limited rate capacity and poor durability) due to weak electrical conductivity and insufficient active sites, leading to hindered ion diffusion [68]. Therefore, studies in the literature focused on the transition metals such as Co, Zn etc. as suitable electrode materials to enhance the electrical conductivity and its active sites [68-71]. Elements such as Al, Cu, Fe, Mn, Sn, Zn and Ca, each of which makes a unique contribution to the electrochemical activity and stability of the cathode, have been used in the search for alternatives to Co. Efforts to find alternatives to Co have led to the use of elements like Zn, Ca, and lanthanides, each contributing uniquely to the cathode's electrochemical activity and stability. The combination of transition (NiCoZn) has garnered significant attention due to their intriguing morphology and structure, indicating improved electrochemical activity. Zinc offers excellent chemical stability, provides to shift the oxygen evolution reaction's overpotential and restricts corrosion resistance, while cobalt provides electrical conductivity, catalytic active sites, and theoretical capacitance [68, 71, 72].

The morphology of NiCo and NiCoZn powders was characterized by scanning electron microscope, and also the elemental analysis was performed using the back-scattering elemental mapping. As depicted in Figure 10, the SEM images of NiCo and NiCoZn illustrate the nanoparticle morphologies. There is no significant difference between NiCo and NiCoZn powders due to the low amount of transition metals. As seen in Figure 10(a), the morphology of NiCo is nanoflower spheres with the approximately diameter of 1  $\mu\text{m}$  and particles of are agglomerated and. Figure 10(b) shows that the NiCoZn possesses a wide range of particle size distribution with mixture of spherical and elliptical particles. The EDS analysis confirmed the the synthesis of NiCo and NiCoZn. Figure 10 also shows the presence of Ni, Co and O elements in NiCo and Ni, Co, Zn and O elements in NiCoZn. Moreover, all elements were uniformly distributed in NiCo and NiCoZn. From the mapping analysis, the atomic percentage was identified and NiCo was found to contain 48% Ni, 3% Co and 49% O, while NiCoZn contained 55% Ni, 3% Co, 2% Zn and 40% O. These results suggested that addition of Co and Zn was also confirmed with the synthesis concentration ratio. For comparison to the precursor stoichiometry, the Ni:Co and Ni:Co:Zn ratios measured by EDS analysis were obtained the targeted ratios. This can be attributed to the observation of the expected synthesis of  $\text{Ni}_{0.95}\text{Co}_{0.05}(\text{OH})_2$  and  $\text{Ni}_{0.925}\text{Co}_{0.05}\text{Zn}_{0.025}(\text{OH})_2$ .



**Figure 10.** SEM images of (a) NiCo and (b) NiCoZn, EDS results of (c-e) NiCo and (f-i) NiCoZn.

In order to determine the effect of the  $\text{Co}^{2+}$  and  $\text{Zn}^{2+}$  substituents on the crystal structure of  $\text{Ni}(\text{OH})_2$ , the crystal structure of NiCo and NiCoZn powders is shown in Figure 11. The diffraction peaks of NiCo ( $\text{Ni}_{0.95}\text{Co}_{0.05}(\text{OH})_2$ ) and NiCoZn ( $\text{Ni}_{0.925}\text{Co}_{0.05}\text{Zn}_{0.025}(\text{OH})_2$ ) are consistent with the crystalline nature of the  $\beta$ - $\text{Ni}(\text{OH})_2$  in the literature, as shown in Figure 11 [68-70]. XRD peaks of  $\beta$ - $\text{Ni}(\text{OH})_2$  powder corresponding to (to (001), (100), (101), (102), (110), (111), (103), and (201) planes were obtained at  $2\theta$  values of  $19.3^\circ$ ,  $33.1^\circ$ ,  $38.6^\circ$ ,  $54.1^\circ$ ,  $59.1^\circ$ ,  $62.8^\circ$ ,  $69.4^\circ$ , and  $72.8^\circ$ , respectively planes of hexagonal  $\beta$ - $\text{Ni}(\text{OH})_2$  (JCPDS:14-0117). The XRD patterns showed that the  $\text{Ni}(\text{OH})_2$  content was dominant for NiCo and NiCoZn. After addition to Co and Zn, new diffraction peaks do not appear for  $\text{Ni}_{0.95}\text{Co}_{0.05}(\text{OH})_2$  and  $\text{Ni}_{0.925}\text{Co}_{0.05}\text{Zn}_{0.025}(\text{OH})_2$ , since high content of  $\text{Ni}(\text{OH})_2$  in the precursors was used the synthesis of NiCo and NiCoZn powders.

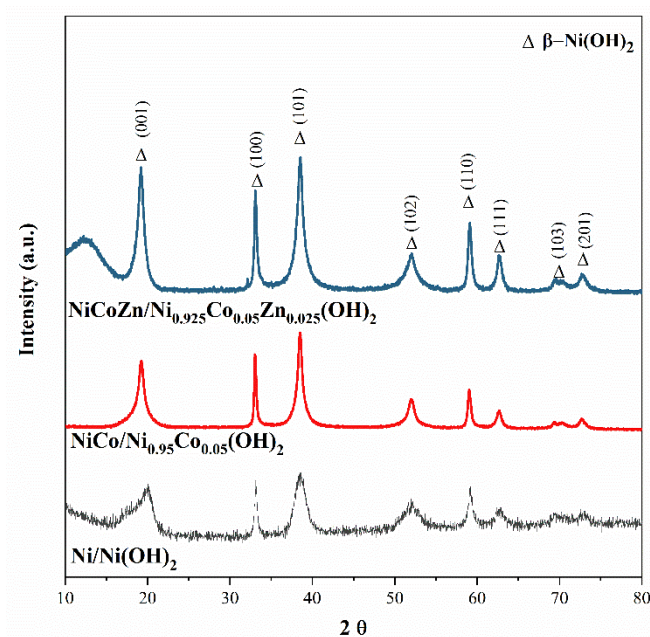


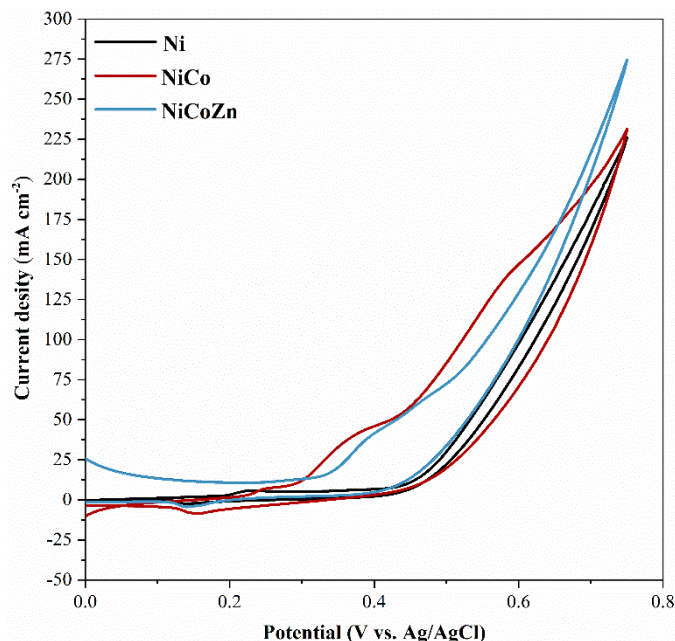
Figure 11. XRD patterns of NiCo and NiCoZn.

ICP-OES analysis is conducted on the cathode materials in Ni-based to determine the amount of metals in the synthesized powders. The Ni:Co and Ni:Co:Zn ratios, Ni<sub>0.95</sub>Co<sub>0.05</sub> and Ni<sub>0.925</sub>Co<sub>0.05</sub>Zn<sub>0.025</sub>, measured by ICP-OES are the same as the targeted ratios for Ni<sub>0.95</sub>Co<sub>0.05</sub> and Ni<sub>0.925</sub>Co<sub>0.05</sub>Zn<sub>0.025</sub> from synthesis, as listed in Table 4. This confirms the content of precursors for NiCo and NiCoZn. The results from ICP\_OES analysis confirmed the findings of mapping analysis.

Table 4. Physiochemical characteristics of NiCo and NiCoZn obtained from elemental analysis (ICP-OES)

	Element (wt.%)			Relative weight % from	
	Ni	Co	Zn	synthesis	ICP-OES analysis
NiCo	51.7	2.6		Ni <sub>0.95</sub> Co <sub>0.05</sub>	Ni <sub>0.95</sub> Co <sub>0.05</sub>
NiCoZn	55.7	3	1.3	Ni <sub>0.925</sub> Co <sub>0.05</sub> Zn <sub>0.025</sub>	Ni <sub>0.925</sub> Co <sub>0.05</sub> Zn <sub>0.025</sub>

The effect of transition metals (cobalt and zinc) on the electrochemical performance of the as-prepared Ni electrodes was examined by CV measurements with Ni-based electrodes in contact with the electrolyte (6 M KOH), as shown in Figure 12.

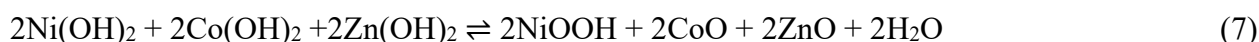


**Figure 12.** CV curves for measured in 6M KOH of all Ni electrodes with various transition metals.

The redox reaction of Ni based-electrodes takes place due to the reaction of  $M^{2+}$  ( $Ni^{2+}$ ,  $Co^{2+}$  and  $Zn^{2+}$ ) with  $OH^-$  [73, 74]. The oxygen evolution mechanism occurs first by adsorption of the hydroxyl group on the electrode. This is followed by the electrochemical desorption of  $OH^-$ . Hydrogen peroxide is formed as an intermediate, which decomposes to oxygen. At the same time, the metal and oxygen group bond are broken. The reaction mechanism occurring in alkaline media is shown in Eqs. (3)-(6) (M: metal, Ni, Co, Zn), where M represents an active site and the species adsorption site [75]



The overall reaction for the oxidation and reduction peaks (Eq. 7) in CV curves may be as follows:



The electrode prepared with NiCoZn showed a higher OER current density ( $274.7 \text{ mA cm}^{-2}$ ) when compared to other electrodes. A similar reduction in current density was measured for all Ni

electrodes (Ni, NiCo, and NiCoZn). The higher electrochemical activity measurements for both ORR and OER were observed for the NiCoZn electrode exhibits when compared to Ni and NiCo electrodes due to high onset potential ( $E_{\text{onset}}$ ), as seen in Table 5.

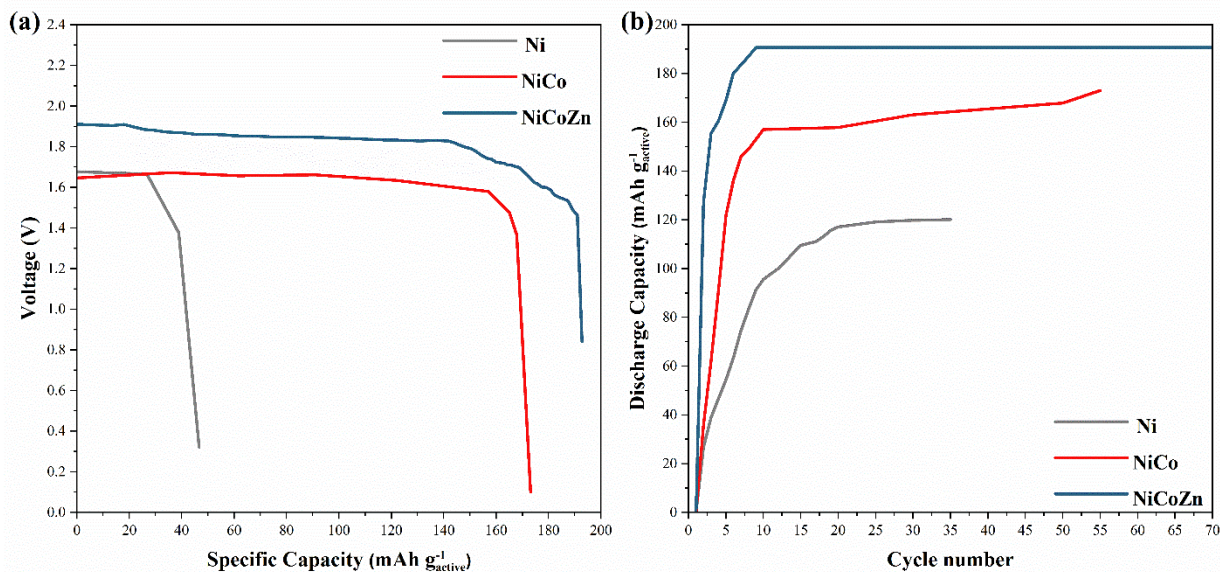
**Table 5.** OER and ORR onset potentials of Ni electrodes with various transition metals for ORR and OER

Electrodes	Onset potential, mV	
	OER	ORR
Ni*	63	167
NiCo	162	155
NiCoZn	318	206

\*Ni\_pH12\_3h\_50°C

The NiZn batteries with Ni, NiCo and NiCoZn electrodes were tested at a constant current of 10 mA cm<sup>-2</sup> to examine its electrochemical performance, as seen in Figure 13. The specific capacities of Ni-based electrodes were performed by their galvanostatic charge–discharge curves. The specific discharge capacities of 46.7 mAh g<sub>active</sub><sup>-1</sup>, 173.1 mAh g<sub>active</sub><sup>-1</sup> and 192.7 mAh g<sub>active</sub><sup>-1</sup>, were obtained for NiZn batteries with Ni, NiCo and NiCoZn electrodes, respectively. The NiZn battery with NiCoZn exhibited a two-plateau discharge in the 5th cycle (~1.85 V) and delivered the highest discharge capacity (192.7 mAh g<sub>active</sub><sup>-1</sup>) (Figure 13(a)) compared to NiZn batteries with Ni and NiCo electrodes due to potential-energy environments of distinct, reversible Ni-discharge sites. Commercially available NiZn batteries exhibit the specific discharge capacity between 3.5 Ah and 44 Ah [76, 77]. Reported specific discharge capacities of NiZn batteries with Ni-based electrodes such as NiCo<sub>2</sub>O<sub>4</sub>, NiMn/LHD, NiAlCo-LDH/CNT, NiO-CNT//Zn vary between 112 mAh g<sup>-1</sup> and 184 mAh g<sup>-1</sup> [78-82].

The charge/discharge cycling stability of the NiZn battery consisting of Ni, Ni Co and NiCoZn electrodes was tested at a current density of 10 mA cm<sup>-2</sup>, as seen in Figure 13(b). The NiZn battery with NiCoZn electrode showed excellent stability during the charge/discharge cycles. The NiZn batteries with Ni and NiCo electrodes failed after 35th and 55th charge/discharge cycles, respectively. This observation suggests that the NiCoZn electrode is a promising candidate as an electrode with the NiZn battery performance.



**Figure 13.** (a) The voltage profiles and (b) discharge capacities vs. time of NiZn batteries with a Ni, NiCo and NiCoZn electrode at 10 mA cm<sup>-2</sup>.

#### 4. CONCLUSION

This study demonstrated the successful synthesis of Ni(OH)<sub>2</sub> with nanosphere structure by a simple hydrothermal method. With the use of Co and Zn as active materials, Ni<sub>0.95</sub>Co<sub>0.05</sub>(OH)<sub>2</sub> and Ni<sub>0.925</sub>Co<sub>0.05</sub>Zn<sub>0.025</sub>(OH)<sub>2</sub> composite materials were synthesized and used as novel electrode materials. The Ni<sub>pH12\_3h\_50°C</sub> (Ni) electrode showed a higher current density of 220 mA cm<sup>-2</sup> when compared to other as-fabricated Ni electrodes. Combining the enhanced conductivity due to the addition of cobalt and zinc as active materials and the improved ORR and OER performances, the NiCoZn electrode achieved an excellent current density of 374.4 mA cm<sup>-2</sup>. In addition, we fabricated a promising battery by pairing the NiCoZn electrode with the Zn anode. The NiCoZn powder was tested for NiZn battery application, which presents high discharge capacity and long cycle life in an alkaline electrolyte. It was found that the commercial NiZn batteries exhibited the discharge capacity, varying from 3.5 to 44 Ah [76, 77]. In the literature, the maximum specific discharge capacity (192.7 mAh g<sup>-1</sup><sub>active</sub>) is much higher than those of NiZn batteries reported such as NiCo<sub>2</sub>O<sub>4</sub>//Zn (183.1 mAh g<sup>-1</sup>) [78], NiMn/LHD//Zn (180 mAh g<sup>-1</sup>) [79], NiAlCo-LDH/CNT//Zn (184 mAh g<sup>-1</sup>) [80], NiO-CNT//Zn (155 mAh g<sup>-1</sup>) [81], and NiCo<sub>2</sub>O<sub>4</sub>//Zn (112 mAh g<sup>-1</sup>) [82]. The NiZn battery with NiCoZn (Ni<sub>0.925</sub>Co<sub>0.05</sub>Zn<sub>0.025</sub>) electrode exhibited a remarkable discharge capacity and good cycling durability (70 cycles), making it a promising candidate for cathode materials in NiZn batteries.

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## DECLARATION OF ETHICAL STANDARDS

The author of the paper submitted declares that nothing which is necessary for achieving the paper requires ethical committee and/or legal-special permissions.

## CONTRIBUTION OF THE AUTHORS

**Gizem Cihanoğlu:** Performing the experiments and analyzing the results, writing the manuscript.

## CONFLICT OF INTEREST

There are no conflicts to declare.

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