Characterization of Hydroxyapatite Coating on Ti6Al4V by Sol-gel Method

Eyüp Can BİÇER¹, Atilla EVCİN², Gür Emre GÜRAKSIN¹*

¹Afyon Kocatepe University, Engineering Faculty, Biomedical Engineering Department, Afyonkarahisar-Turkey
²Afyon Kocatepe University, Engineering Faculty, Material Science and Engineering Depart, Afyonkarahisar-Turkey

* Corresponding Author : emreguraksin@aku.edu.tr
ORCID: 0000-0002-1935-278

Abstract:
Biomaterials are natural or synthetic materials that are used in human body and are materials to evaluate, treat, support or replace any tissue, organ, or function of the body. Due to its favorable biocompatibility and mechanical properties, HA-coated metal-based implants are used in the orthopedic surgery operations. For this purpose, sol-gel synthesized nano sized HA powders were produced. Ti6Al4V alloy was coated with hydroxyapatite (HA) by using the sol-gel method. Surface morphologies, thermal, mineralogical properties and bioactivity of the coatings were examined and the characterization of the coatings was performed by using DTA-TG, XRD, SEM, and EDX.

1. Introduction

Bone is a natural composite which mainly consists of an inorganic (Hydroxyapatite-HA) and an organic (collagen). Bones have different pore shapes according to their locations and functions throughout the body. The skeleton consists of two main structural types of bone as shown in Fig 1. Compact (cortical) bone is dense with a low surface area. Spongy (trabecular or cancellous) bone has a lower density and larger surface area than compact bone. The mineral part of the bone is constituted by calcium phosphates, in the forms of crystallized HA and/or amorphous calcium phosphates (ACP), which contains various ions like Na+, Mg²⁺, K⁺, CO₃²⁻, Cl⁻, and F⁻ [1,4].

Sol-gel process is an attractive alternative to other methods for synthesis of ceramics and glasses for many reasons: for example, low temperature synthesis, simple equipment to be used, thin film formability and so on [9-11].
Many researchers have used different methods for the fabrication of hydroxyapatite coatings such as plasma spray, HVOF, molecular beam epitaxy (MBE) vapor-liquid-solid mechanism (VLS), solution-phase, shadow sputtering, and sol-gel methods [7].

In this work, we developed Hydroxyapatite nano particles and coating by sol-gel on Ti6Al4V metallic implant. Crystal structure, surface morphology, composition of samples was characterized by means of XRD, SEM, and EDX.

2. Materials and Methods

To synthesize powders chemically, Analytic grade Ca(NO$_3$)$_2$.4H$_2$O, (NH$_4$)$_2$HPO$_4$, NH$_4$OH were used as precursors. All chemicals were purchased from Merck and used as received. Ti6Al4V (Grade 5) was used as metallic implant to coat. 1.00 M stock solutions of calcium nitrate 4-hydrate and di-ammonium hydrogen phosphate were prepared using distilled water. A precipitate was formed by the drop-wise addition of the (NH$_4$)$_2$HPO$_4$ solution to the Ca(NO$_3$)$_2$ solution, with continuous stirring, to obtain a Ca/P ratio of 1.67 in the mixed solution. The solution was brought to pH 11 by addition of concentrated ammonium hydroxide solution (Fig 2). The obtained precipitation was HA. The precipitate was aged overnight at room temperature. The resulting powder was dried in an oven for 10 h. Finally, the dried powder was calcined in an electrical furnace at 1000 °C for 2 h after heating at the rate of 5 °C/min. in air.

Then these films were air dried after deposition and heated and dried at 100 °C in an oven for 15 min. For hydroxyapatite phase formation, thin films were annealed at 500 °C for two hours at ambient conditions. Structural and thermal properties of calcined powders were characterized by XRD and DTA/TG respectively. Morphological and surface properties of annealed thin films were studied. In vitro bioactivity test of thin films were performed in stimulated body fluid (SBF) at 37 °C under static condition for 1-21 days.

2. Results and Discussion

The thermal properties of the HA powders were characterized by TG-DTA (Netzsch STA449F3) (Fig 4). There are one endothermic and two exothermic peak. An endothermic peak was found at 175 °C, due to removal of water and organic elements from powder. Thermal decomposition of the powder began at 400 °C. Second exothermic peak was observed at 800 °C. After this reaction, the crystal structure of the system becomes stable. Total weight loss was %11.44.

Crystal structures of the powders were characterized via powder X-ray diffraction (XRD, Bruker D 8 Advance) using Cu-K$\alpha$ X-rays of
wavelength = 1.5406 Å and data were taken in the
2θ range of 20° to 70° with a step of 0.1972°. As
can be seen in Fig. 4, all spectra showed the HA
reflections.

Figure 5. XRD patterns of HA powders

Thickness of coating layer and surface structure
were obtained from the cross-sectional and surface
images using a SEM-EDX (LEO 1430 VP). Fig 6
shows the morphology of HA powders. HA
powders are approximately under 1 μm. Fig. 7
shows the surface morphology of the HA coating
on Ti6Al4V. SEM image of coating surface is
shown in Fig 7 to illustrate crack free, smooth and
porous surface. SEM image of cross section is
shown in Fig 8 to illustrate homogenous coating
layer. Thickness of coating for HA was about 98
μm.

Figure 6. SEM image of HA powders

Figure 7. SEM image of HA coating surface.

Figure 8. SEM image of HA coating cross section.

Table 1. SEM-EDX analysis of HA coating surface.

<table>
<thead>
<tr>
<th>Element</th>
<th>Series</th>
<th>Net C</th>
<th>Unm.C</th>
<th>Nor. C</th>
<th>Atom C</th>
</tr>
</thead>
<tbody>
<tr>
<td>Carbon</td>
<td>K</td>
<td>19643</td>
<td>11.77</td>
<td>9.75</td>
<td>19.24</td>
</tr>
<tr>
<td>Oxygen</td>
<td>K</td>
<td>16174</td>
<td>41.41</td>
<td>34.28</td>
<td>50.79</td>
</tr>
<tr>
<td>Sodium</td>
<td>K</td>
<td>3511</td>
<td>2.24</td>
<td>1.85</td>
<td>1.91</td>
</tr>
<tr>
<td>Aluminium</td>
<td>K</td>
<td>6561</td>
<td>2.10</td>
<td>1.74</td>
<td>1.53</td>
</tr>
<tr>
<td>Silicon</td>
<td>K</td>
<td>2129</td>
<td>0.58</td>
<td>0.48</td>
<td>0.41</td>
</tr>
<tr>
<td>Phosphorus</td>
<td>K</td>
<td>3127</td>
<td>0.81</td>
<td>0.67</td>
<td>0.51</td>
</tr>
<tr>
<td>Chlorine</td>
<td>K</td>
<td>6055</td>
<td>1.40</td>
<td>1.16</td>
<td>0.77</td>
</tr>
<tr>
<td>Calcium</td>
<td>K</td>
<td>7716</td>
<td>1.69</td>
<td>1.40</td>
<td>0.83</td>
</tr>
<tr>
<td>Titanium</td>
<td>K</td>
<td>159953</td>
<td>57.36</td>
<td>47.48</td>
<td>23.51</td>
</tr>
<tr>
<td>Iron</td>
<td>K</td>
<td>2348</td>
<td>1.44</td>
<td>1.20</td>
<td>0.51</td>
</tr>
<tr>
<td>Total</td>
<td></td>
<td>120.8</td>
<td>100.0</td>
<td>100.0</td>
<td></td>
</tr>
</tbody>
</table>
As can be seen in Table 1, Ca/P atomic ratio is 1.63. Despite the well-recognized importance of single-phase HA, however, precisely stoichiometric HA (a Ca/P ratio of exactly 1.67) is not always present.

![Figure 9. SEM images of surface (after SBF solution dipping)](image)

In order to study the weight variety of HA coating in SBF, the dry weight of coating was measured before and after the incubation, and the percentage of weight increase (Wp) can be calculated according to the following equation (1):

$$W_p = \frac{W_b - W_a}{W_a} \times 100$$  \hspace{1cm} (1)

Where $W_b$ and $W_a$ represent the weight of sample after and before soaking in SBF, respectively. The mass change of samples before and after immersing in SBF for 1, 7, 14, and 21 days was measured by using an analytical balance (accuracy: 0.1 mg). All the samples were thoroughly rinsed with distilled water and dried at 65 °C before weighing. The bio resistivity of the specimens was assessed by SBF solution. This solution was prepared according to the protocol suggested by Kokubo et al. [13,14] Then samples were weighed and characterized by SEM. As can be seen in Fig 9 new HA crystals occurred on surface. The results show that the amount of particles formed on the HA coated Ti6Al4V implant increased with the increasing of soaking time and the percentages of weight increase at immersing for 1, 7, 14 and 21 days were 1,43 %, 3,18 %, 10.40 % and 17,42 %, respectively.

4. Conclusion

In this work, sol-gel synthesized (nano sized) hydroxyapatite (HA) powders were successfully produced and used for coating Ti6Al4V implant by using sol-gel method. Experiments we have demonstrated that HA can be coated on Ti6Al4V by sol-gel method at the low temperature.

Bioactivity results showed that HA coated Ti6Al4V implants are as equally successful as HA coated implants in achieving osseointegration. Finally it can be use in biomedical application as an implant. With the increase in soaking time, the amount of new hydroxyapatite seed formed on surface increased. This was confirmed by the weight assessment and SEM images.

XRD peaks of synthesized powders completely match with the standard HA peaks.

A dense and homogeneous surface layer is expected to contribute to high adhesion strength between the HA layer and the Ti6Al4V implant. The measured coating thickness of HA on Ti6Al4V is 98 μm.

Coating surface is homogeneous and smooth.

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